Highly selective CO₂ vs. N₂ adsorption in the cavity of a molecular coordination cage†

James S. Wright,‡ Alexander J. Metherell,‡ William M. Cullen, ‡ Jerico R. Piper, Robert Dawson ‡* and Michael D. Ward ‡*

Porous solid-state materials are attractive for gas adsorption purposes, with several classes of porous material gaining increasing attention in recent years. These include metal–organic frameworks (MOFs)/coordination polymers;¹–¹⁷ covalent organic frameworks (COFs)/microporous organic polymers (MOPs);¹⁸–²⁴ molecular cages,²⁵–³⁵ and other molecular crystals.³⁶–⁴³

In the case of MOFs and MOPs, impressive gas uptake capacities have been reported, and extremely highly porous materials described.¹²,¹⁴,²¹ However, higher uptake capacity in porous materials can come at the expense of selectivity between small gaseous molecular guests, as shown in previous work comparing porous organic cages of different pore sizes with each other, and with MOFs.²³ Adsorbents which are selective for the desired adsorbate are desirable, but not necessarily at the expense of uptake capacity. For this purpose, the design of flexible adsorbents whose pores may open under the influence of an external stimulus has been demonstrated, both in MOFs¹⁷,¹³ and extrinsically porous materials;¹⁷,²⁷,⁴² this is still an emerging field.

Perhaps better developed is the functionalisation of the pore space of intrinsically porous materials, to enhance selectivity for binding of different gaseous guests. In particular, the improvement of CO₂ adsorption selectivity in MOFs has been demonstrated by the addition of hydrogen-bonding sites³,¹¹ or the fluorination of pores.⁵,⁴⁴,⁴⁵ These internal surface modifications can however come at the expense of uptake capacity by occupying some of the interior space, so an adsorbent in which a binding site is built into the ‘walls’ of the cavity is desirable.

We have previously reported the structures and guest binding properties of the cubic coordination cages [M₈L₁₂]X₁₆, in which M are transition metal dications [usually Co(II)] located at the vertices of the cage, and L are bis(pyrazolyl-pyridine) bridging ligands which connect a pair of metal ions along every edge of the assembly (Fig. 1).⁴⁶–⁵² The ligand L may be unsubstituted (L⁰: R = H in the figure) in which case the cages are soluble in polar organic solvents;⁴⁷,⁴⁸ or may be substituted (L⁵: R = CH₂OH in the figure) to make the cages water-soluble.⁴⁹–⁵² These cages have been shown to bind a wide range of organic guests in the central cavity. In organic solvents guest binding is partly driven by hydrogen-bonding of electron-rich regions of guests to H-bond donor pockets located on the interior surface of the cage, in regions of high positive electrostatic potential; this affords binding constants in the range 10²–10⁵ M⁻¹.⁴⁷ In water, the hydrophobic effect provides the dominant driving force for strong binding of hydrophobic guests with binding...
constants of up to $10^{3}$ M$^{-1}$. Here we report an investigation into the gas sorption capability of these materials, demonstrating a high selectivity for CO$_2$ uptake over N$_2$ in the solid state, which we ascribe to the presence of the same H-bond donor sites on the cage interior surface that facilitate guest binding in solution.

We used the cages [Cd$_8$(L$_8^{(a)}$)$_{12}$][BF$_4$]$_{16}$ (A) and [Co$_8$(L$_8^{(a)}$)$_{12}$][BF$_4$]$_{16}$ (B), both of which have been reported before. The compounds were prepared as methanol solvates, and then dried and thermally desolvated. Powder X-ray diffraction analysis of the dried materials (ESI†) showed that B retains the same phase that was observed in the single crystal structure of the methanol solvate, whereas A loses crystallinity and becomes amorphous. This is likely related to the fact that in B the presence of hydroxymethyl groups on the exterior surface of the cages results in a formation of an intermolecular O-H·O hydrogen-bonding network of cage molecules which allows crystallinity to be retained even when solvent molecules are lost (ESI†). In A in contrast there are no such interactions between the exterior surfaces of adjacent cages and crystallinity is lost on desolvation. However, $^1$H NMR and mass spectrometric analyses confirmed that the integrity of the molecular cages is retained even when the crystals are desolvated.

The cages were found to be non-porous having BET surface areas <20 m$^2$ g$^{-1}$. The volumetric gas sorption isotherms were measured for uptake of CO$_2$ and N$_2$ by both cages at 298 K (Fig. 2) and also at 273 K (ESI†). Both cages demonstrate highly selective uptake for CO$_2$ vs. N$_2$ at both temperatures. The gas uptake comparisons and selectivity constants are summarised in Table 1, and Henry’s law calculation data is presented in the ESL.† The capacity for CO$_2$ uptake is very similar for both cages. Although the cages have identical internal cavities, as we mentioned above, the supramolecular structure of the cages is different because of the presence (cage B) or absence (cage A) of inter-cage hydrogen-bonding interactions between peripheral functional groups. This suggests either that CO$_2$ uptake in the interstitial spaces between cages is very low, or that the void space between cages is similar in both materials (which in the case of the desolvated cage B is known to be small due to the hydrogen bonding, and therefore uptake here would be low anyway). We have noted in previous work that when crystalline cage samples are soaked in solutions of guests, quite large guest molecules can permeate the crystals and enter the cage cavities, even when the windows are occluded in the crystal structure and when the guest dimensions are larger than the 4 Å windows in the cage faces. Thus, for guest molecules as small as N$_2$ or CO$_2$, differences in crystal packing are unlikely to prevent adsorption: the similarity in CO$_2$ uptake for both cage types therefore most plausibly relates to the similarity of the cavity inside cages A and B. Adsorption/desorption of CO$_2$ is reversible in both cases with only a slight hysteresis, as is common for porous materials, which diminishes at low pressures.

Attempts to locate CO$_2$ guests within the cages were made using X-ray crystallography on single crystals under a CO$_2$ atmosphere, but the crystals fractured rapidly into micro-crystalline powder upon desolvation. Instead, using the method that has worked with other guests, single crystals of B (still solvated to prevent cracking) were soaked in liquid CO$_2$ at 40 °C for 2 hours. This resulted in uptake of CO$_2$ into the cage cavity.

Crystallographic analysis (ESI†) showed the structure to be [Co$_8$(L$_8^{(a)}$)$_{12}$][BF$_4$]$_{16}$-CS$_2$.5H$_2$O (Fig. 3) in which a molecule of CO$_2$ is located such that it interacts with one of the hydrogen-bond donor pockets on the interior surface which are located at the two fcc tris-chelate sites at either end of the long diagonal of the approximately cubic assembly.

The site occupancy of the CS$_2$ in each of the two pockets is 0.5, i.e. in the crystal structure there is one guest molecule per cage but it is disordered equally over the two possible sites. The CS$_2$ guest is oriented such that the S atom [S(11S)] that is directed into the corner pocket is involved in several CH···S contacts (C···S distances in the range 3.5–3.7 Å) with H atoms from CH$_2$ groups and naphthyl groups that converge around the guest binding site; the (non-bonded) Co(1)···S(11S) separation is 5.65 Å. The other S atom of the guest S(13S) is also involved in a short CH···S contact (3.51 Å) with an inwardly-directed naphthyl CH proton. This set of interactions is emphasised in Fig. 3(b).

The quadrupole moment of CS$_2$ is opposite in sign to that of CO$_2$ so in terms of point charges it is denoted (δ$^+$)-(δ$^-$)-(δ$^+$).
in this respect CS$_2$ is not electronically analogous to CO$_2$ although it is a reasonable geometric model. Nonetheless the ability of the S atoms of CS$_2$ to act as hydrogen-bond acceptors, based on the local electron density at the S atoms associated with lone pairs, is well established.$^{57-60}$ We showed a while ago that the convergent array of CH donors located close to the CO$_2$ guest (50% site occupancy in each of two positions) shows space-filling; (b) a close-up of the hydrogen-bonding environment around the CS$_2$ guest, with the shortest CH...S contacts shown with red dashed lines.

Fig. 3 Crystal structure of the complex cation of [Co$_8$(Lw)$_{12}$]BF$_4$$_{16}$; CS$_2$; SH$_2$O: (a) a view of the entire cage (in wireframe) with the CS$_2$ guest (50% site occupancy in each of two positions) shown space-filling; (b) a close-up of the hydrogen-bonding environment around the CS$_2$ guest, with the shortest CH...S contacts shown with red dashed lines.

In conclusion we have demonstrated good CO$_2$ uptake by a molecular cage complex in which the gaseous guest binds in the central cage cavity even when the bulk materials are not conventionally porous. Such examples of gas sorption into the cavities of molecular cages – in contrast to porous network materials – are very rare.$^{61,62}$ On the basis of the structural model based on CS$_2$, we suggest that this arises because of favourable polar interactions between the CO$_2$ guest and charge-assisted hydrogen-bond donor sites on the interior surface of the cage host;$^{47,53,54}$ these same structural features also result in particularly high selectivity for binding of CO$_2$ compared to non-polar N$_2$.

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Notes and references

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