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Thermomechanically processed chitosan:gelatin films being transparent, mechanically robust and less hygroscopic

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Abbreviations: RH, relative humidity; TGA, thermogravimetric analysis; *WI*, whitish index; SEM, scanning electron microscopy; FTIR, Fourier-transform infrared; XRD, X-ray diffraction; DMTA, dynamic mechanical thermal analysis; WCA, water contact angle.

Abstract

Chitosan and gelatin are attractive polymeric feedstocks for developing environmentally benign, bio-safe, and functional materials. However, cost-effective methods to achieve advantageous materials properties and tailor their functionality are still lacking, but interesting. Herein, we found that physically mixing chitosan and gelatin at 1:1 (w/w) ratio resulted in materials with properties (higher Young's modulus (603.8 MPa) and tensile strength (33.6 MPa), and reduced water uptake (45%) after 6 h of water soaking) better than those of the materia s ba ed on mainly chitosan or gelatin. We attribute this synergy to the ionic and hydrogen-bonding interactions between the two biopolymers enabled by high-viscosity thermomechanical processing. Despite the lowest hygroscopicity, the 1:1 chitosan:gelatin films disp'ay of the highest surface hydrophilicity. Besides, addition of gelatin to chitosan led to films bein brighter, more transparent and amorphous. Thus, this work has generated new understand² at to enhance the application of biopolymers for e.g. packaging, coating, and biomedical copilications.

Keywords

Chitosan; gelatin; biopolyme) composite; optical properties; mechanical properties; water absorption

Chemical compounds studied in this article

Chitosan (PubChem CID: 71853); Gelatin (PubChem CID: 381623137); Water (PubChem CID: 962); Glycerol (PubChem CID: 753); Acetic acid (PubChem CID: 176)

1 Introduction

Chitosan, a multifunctional polysaccharide formed by β -(1,4)-linked *N*-acetyl-D-glucosamine and D-glucosamine units, has some promising features from the application point of view, such as edibility, biodegradability, antimicrobial activity (Zhao, Wei, Xu, & Han, 2020), and biocompatibility (Pereda, Ponce, Marcovich, Ruseckaite, & Martucci, 2011). Therefore, chitosan has been widely explored as vaccine adjuvants (Li et al., 2021), dyes or heavy meta' ion adsorbents (Aramesh, Bagheri, & Bilal, 2021; Zhang et al., 2021), drug delivery carrier. (Kt rakula, Gorityala, & Moharir, 2021), tissue engineering candidates (Wu, Dong, Li, Wang, & Cao, 2017), wound healers (Torkaman, Rahmani, Ashori, & Najafi, 2021), and disposable packagu, < materials (Wang, Ding, Ma, & Zhang, 2021). However, for high-volume applications such a environmental remediation and disposable packaging, chitosan is much more expensive and has less satisfactory material properties (weak mechanical properties and high hydroph in ity) than petroleum-derived plastics (Xu, Wei, Jia, & Song, 2020). Moreover, the transparency of the chitosan film is relatively low (Meng, Xie, Zhang, Wang, & Yu, 2018), which may also restrict its application.

Gelatin, obtained fron. pa.: al hydrolysis of collagen, is one of the most widely used proteins in food (Alexandre, Lourenço, Bittante, Moraes, & Sobral, 2016; Zhang, Liang, Li, & Kang, 2020). Gelatin has advantages such as biodegradability, non-toxicity, biocompatibility, good film-forming ability and barrier properties, along with low cost (about 1/15 that of chitosan) (Ji et al., 2020). Gelatin has also been used in food packaging (Nur Hanani, Roos, & Kerry, 2014; Roy & Rhim, 2020), controlled drug release (An, Gou, Yang, Hu, & Wang, 2013), biomedical (Yang, Li, & Nie, 2007), and battery (Sun et al., 2008) applications. However, the application of gelatin may be

hindered by its high hygroscopicity, low thermal stability, and relatively poor mechanical properties (brittleness) (Ebrahimi, Fathi, & Kadivar, 2019; Guo et al., 2013).

Polymer blending can result in improvement in the physical properties of individual components (Meng et al., 2018). Interactions between different polymers may lead to improved mechanical properties (Pereda et al., 2011). Thus, the mixing of gelatin and chitosan could be a promising way to obtain new materials with enhanced physicochemical properties ard new functionality. Charged carboxylic groups (COO—) from gelatin amino acids can electros atic Illy interact with the protonated amino groups of chitosan, resulting in stable pol; electrolytic complexes and transparent and homogeneous films (Qiao, Ma, Zhang, & Yao, 2017: Rivero, García, & Pinotti, 2009; Rodrigues, Bertolo, Marangon, Martins, & Plepis, 2020).

Researchers characterized the molecular is eractions in chitosan–collagen complexes by viscometry, wide-angle X-ray scattering (VA.(S)), and Fourier-transform infrared (FTIR) spectroscopy. It was found that chito. an and collagen can interact with each other at the molecular level (Qiao et al., 2017; Sionkowski, Wisniewskia, Skopinskaa, Kennedy, & Wess, 2004). Furthermore, it was found that a polyelectrolyte complex could be formed with pH > 4.7 (pH_{iso} of gelatin) in a chitosan–gelatin solution (Yin, Li, Sun, & Yao, 2005). Then, the composite films prepared from gelatin (or collagen) and chitosan have been found to possess improved antimicrobial activity, mechanical or barrier properties compared with those of the single-component films (Pereda et al., 2011; Rivero et al., 2009).

While solution methods have been widely used in research studies for preparing biopolymer materials (e.g. films, hydrogels, and nanofibers) (Astaneh et al., 2020; Chiono et al., 2008; Ebrahimi

et al., 2019; Pereda et al., 2011; Qiao et al., 2017; Wang et al., 2021; Wu et al., 2017; Yin et al., 2005), they may suffer from the drawbacks of low efficiency and high solvent waste generation. Thermoplastic methods are more feasible to produce films on an industrial scale, which is less solvent-demanding and can provide high shear forces and processing temperature (Wang et al., 2021). Moreover, material mixing and molecular interactions might occur in different ways under high-viscosity conditions, which has been studied to a very limited extent. It was recently reported that thermomechanical processing could lead to chitosan:silk pep ide and chitosan:carboxymethyl cellulose materials with unexpected mechanical properties (*Aeng* et al., 2018) and hydrolytic stability (Chen, Xie, Tang, & McNally, 2020a), respective, an this regard, high-viscosity mixing of polyelectrolyte biopolymers could lead to better ic and and hydrogen-bonding interactions between biopolymer chains (Chen et al., 2020a; Meng et al., 2018). However, to the best of our knowledge, there has been no study hereto on chitos and generation hybrid materials obtained by thermomechanical processing.

This work is based on the lypo hesis that molecular interactions between chitosan and gelatin, if properly realized, can lead to chitosan:gelatin materials with better properties than those of materials based on primarily chitosan or gelatin. Glycerol is one of the most widely used plasticizers for biopolymers due to its non-volatility, large availability, matching hydrophilicity, and low exudation (Chen et al., 2020a; Epure, Griffon, Pollet, & Avérous, 2011). In this work, we prepared chitosan:gelatin composite films with limited amounts of solvents using thermomechanical mixing and compression molding. The molecular interactions between gelatin and chitosan and the morphology, structure, and physical properties (mechanical properties, thermal stability, glass transition temperature, and water absorption) of the composite films were investigated. In particular, we found blending a certain content of gelatin into chitosan is instrumental to the processing of chitosan and to achieving advantageous material properties. Thus, this work could be insightful for the development of cost-effective high-performance biopolymer materials.

2 Experimental

2.1 Materials

Chitosan (CS), derived from crustaceous shells, with a specification of BR, was purchased from Jinan Xinhong Huagong Co., Ltd (Jinan, China). This chitoson has a degree of deacetylation (DD) of 90% and a viscosity of about 400 mPars (1% solution in 1% acetic acid at 25 °C). It had an original moisture content of 10.65 wt% measured by v Cight loss during drying. Gelatin (GA), in food-grade (type A, bloom index 250), was supplied by Rousselot Gelatin Co., Ltd (Wenzhou, China). The original moisture content of the gelatin is 1?.10 wt%. Glycerol (AR grade) and acetic acid (AR grade) were purchased from Sinopharm C. emical Reagent Co., Ltd (Shanghai, China). All these chemicals were used as received without inter purification.

2.2 Sample preparation

Different samples were prepared according to the formulations listed in **Table 1**. The sample codes such as "75CS-7.5%" was used, where "75CS" represents the mass percentages of dry chitosan in the biopolymer matrix while "7.5%" indicates the mass content of glycerol added to the biopolymer matrix. Chitosan and gelatin were mixed by mechanical stirring for 10 min, during which 2 M acetic acid and glycerol were added dropwise. The mixtures were stored overnight at 4 °C. For each batch, 80 g of the mixed sample was thermally kneaded (80 rpm screw speed, 80 °C) for 15 min

using a twin-rotor HAAKE Rheomix mixer coupled with a Polylab RC600p system (ThermoHaake, Germany). Afterward, 35 g of the thermomechanically processed material was hot-pressed (80 °C, 2500 psi, 10 min) into a film using a flat sulfaration machine (Guangzhou Shunchuang Rubber Machinery Company, Guangzhou, China) along with a mold with a 100 mm × 100 mm × 1 mm hollow molding space. The hot-pressed films were soaked in methanol for 12 h and then washed with distilled water. All the specimens were dried in an oven at 30 °C for 24 h and then stored in a desiccator (57% relative humidity (RH)) for 4 weeks, followed by conditioning at either 57% RH or 75% RH for another week before characterization. Except for the amples conditioned at 57% RH only.

Sample	Gelatin dry mass	Clatosan dry mass	Glycerol (g) ^a	2 M Acetic acid
	(g) ^a	(g) ^a		(mL)
100CS	- 2	26 (100%)	_	61.90
100CS-7.5%	- 2	26 (100%)	1.95 (7.5%)	59.95
100CS-15%	_)	26 (100%)	3.90 (15%)	58.00
75CS	6.5 (25%)	19.5 (75%)	_	61.69
75CS-7.5%	6.5 (25%)	19.5 (75%)	1.95 (7.5%)	59.74
75CS-15%	6.5 (25%)	19.5 (75%)	3.90 (15%)	57.79
50CS	13 (50%)	13 (50%)	_	61.49
50CS-7.5%	13 (50%)	13 (50%)	1.95 (7.5%)	59.54
50CS-15%	13 (50%)	13(50%)	3.75 (15%)	57.59
25CS	19.5 (75%)	6.5 (25%)	_	61.29

 Table 1 Sam₁ e codes and formulations.

25CS-7.5%	19.5 (75%)	6.5 (25%)	1.95 (7.5%)	59.34
25CS-15%	19.5 (75%)	6.5 (25%)	3.90 (15%)	57.39

^a The numbers in the brackets are the mass percentages of either biopolymer or glycerol based on the total mass of the biopolymer matrix.

2.3 Characterization

2.3.1 Optical properties measurement

The color of the films was determined using a CIE colorin tete (X-Rite, Inc., USA). The *L* (lightness/brightness), a^* (redness/greenness) and b^* (ye low tess/blueness) values were obtained by placing the film on a white reflector standard plate. Five measurements were taken on each film surface. Then, the total difference in color (Δc) v/as calculated according to the equation as follow (Boekel, 1996):

$$\Delta E = \sqrt{(\Delta a)^2 + (\Delta b)^2 + (\Delta L)^2} \tag{1}$$

where ΔL , Δa , and Δb are the differences between the corresponding color parameters of the sample and that of the white standar 1 (L = 93.63, a = 0.95, and b = 0.46).

Whitish index (WI) was calculated according to the equation as follow (Avena-Bustillos,

Cisneros-Zevallos, Krochta, & Saltveit, 1994):

$$WI = 100 - ((100 - L)^2 + a^2 + b^2)^{1/2}$$
(2)

2.3.2 Transparency

A UV–visible spectrophotometer (Lambda 1050+, PerkinElmer, USA) was used to measure the light transmittance (T_{500}) of the films at a wavelength of 500 nm. The test was carried out in triplicate

for each sample.

2.3.3 Scanning electron microscopy (SEM)

The cryo-fractured surface morphologies of the biopolymer films were investigated using a scanning electron microscope (Phenom, Eindhoven, Netherlands) operated at a voltage of 10 kV. The samples were fractured with liquid nitrogen and were coated with gold for 90 s using a Q-150R-S sputter-coater (Quorum Technologies Ltd, UK) under vacuum before SEM imaging.

2.3.4 Fourier-Transform Infrared (FTIR) Spectroscopy

FTIR spectroscopy was performed using a PerkinElmer Frontier FTIR spectrometer (Spectrum-3, Germany) fitted with a Zn-Se attenuated total reflectance (Fr, R) accessory. FTIR spectra were collected against the air as the background over a variant method of $4000-600 \text{ cm}^{-1}$ at a resolution of 4 cm^{-1} . Each collection was based on 64 scans.

2.3.5 X-ray diffraction (XRD)

XRD analysis of the condition :d cbitosan:gelatin films was performed using an Xpert PRO diffractometer (Bruker, Germany) at 40 mA and 40 kV with Cu K α radiation (wavelength: 0.15418 nm) as the X-ray source. The scanning diffraction angle (2 θ) was from 5° to 60° with a scanning speed of 2.16 s/step and a scanning step of 0.02°.

2.3.6 Dynamic mechanical thermal analysis (DMTA)

A PerkinElmer Pyris Diamond DMA8000 instrument was used to evaluate the dynamic mechanical properties of the films as rectangular tensile bars in the single cantilever tensile mode. The length of the tested tensile section was 20 mm. The tests were carried out at a frequency of 1 Hz and a strain of 0.05% with a heating ramp from -50 °C and 200 °C at 2 °C/min. The dynamic storage

modulus (E'), loss modulus (E''), and loss tangent (tan $\delta = E''/E'$) were obtained. To prevent water evaporation during the tests, the specimens were coated with silicone oil.

2.3.7 Thermogravimetric Analysis (TGA)

A PerkinElmer Diamond TGA STA 8000 facility was used to determine the thermal decomposition temperatures of the samples with a temperature ramp from 30°C to 700 °C at 10 °C/min in a nitrogen atmosphere. For each measurement, about 3 mg of the sample was weighed on a platinum pan.

2.3.8 Tensile testing

According to the ASTM D882-18 standard, the tensile properties of the biopolymer films were evaluated using a tensile testing appartus (Instron ASTM D638) with a 100 N load cell at a cross-head speed of 5 mm min⁻¹ at room tennierature. For each sample, the data were generated based on seven specimens.

2.3.9 Water absorption

The water absorption of the Sie polymer films was evaluated by measuring the weight percentages of the films at a ferent time points after immersing in distilled water. Specifically, the film specimens (4 cm \times 4 cm) were placed in a 500 mL beaker containing 300 mL of distilled water under ambient temperature (25 °C). The samples were then taken out at intervals, wiped with Whatman filter paper to remove the excess water on the surface, and weighed (with an accuracy of 0.01 g). The water absorption was expressed as:

Water absorption(%) =
$$\frac{M_t - M_0}{M_0} \times 100$$
 (3)

where M_0 is the initial mass of the specimen before water soaking and M_t the mass of the specimen

after soaking in water for a certain time (*t*).

2.3.10 Contact angle test

The contact angle of water droplets (3 μ L) on the films after 5 s and 60 s was measured using a contact angle system ZJ-7000 (Z. Jia Equipment, Shenzhen, China) at room temperature. For each sample, four different places were measured.

3 Results and Discussion

3.1 Appearance and optical parameters

Fig. S1a shows the different biopolymer films had different color shade and transparency. While the 100CS group of films were the darkest and most z_1 aque, the composite films with a higher content of gelatin were brighter and more transpirent. **Table 2** shows the *L*, *a**, and *b** values of the films. The 25CS-15% film showed the highest *W1* value and the lowest ΔE value while the 100CS group displayed the reverse. Besides, **Tat te 2** shows that the 100CS group of films had the lowest T_{500} values (2.14–3.99%) and the 2JCS group had the highest (51.49–62.56%), which confirms the highest transparency of the latte. The differences in optical characteristics among the composite films can be attributed to the different colors of chitosan and gelatin after processing.

Sample	L	a	b	ΔE	WI	T ₅₀₀
100CS	41.21±0.22	21.59±0.28	36.05±0.21	67.68±0.32	27.73±0.14	3.99±0.04
100CS-7.5%	37.43±0.30	17.00±0.34	29.05±0.07	66.14±0.32	28.95±0.33	5.66±0.03
100CS-15%	45.82±0.13	15.82±0.11	41.26±0.52	65.74±0.45	30.08±0.43	2.14±0.04
75CS	50.97±0.26	11.35±0.25	46.30±0.26	64.51±0.40	31.61±0.40	13.08±0.03

Table 2 Optical characteristic values of the different films.

75CS-7.5%	51.58±0.59	11.95±0.46	45.66±0.35	63.74±0.39	32.37±0.42	17.41±0.03
75CS-15%	56.71±0.15	13.08±0.10	44.56±0.39	59.77±0.50	36.53±0.50	12.4±0.01
50CS	53.40±0.36	8.18±0.53	42.48±0.22	59.67±0.19	36.41±0.06	21.07±0.02
50CS-7.5%	59.34±0.51	2.59±0.30	36.34±0.37	50.72±0.59	45.40±0.59	42.6±0.01
50CS-15%	60.26±0.32	0.58 ± 0.07	32.61±0.48	47.42±0.43	48.59±0.41	41.69±0.03
25CS	64.89±0.12	-1.95 ± 0.04	21.32±0.47	36.75±0.35	58.87±0.32	51.49±0.03
25CS-7.5%	64.93±0.63	-2.01±0.07	20.63±0.08	36.32±0.56	59.26±0.58	61.23±0.03
25CS-15%	65.88±0.17	-0.244 ± 0.24	18.86±0.08	34.590.25	60.94±0.27	62.56±0.05

3.2 Morphology

Fig. 2 shows the SEM images of cryo-fractured surface. of the different biopolymer films. While the 100CS film still contained raw chitosar bardeles, addition of glycerol, as a plasticizer, facilitated the processing of chitosan as shown are the more cohesive structure of 100CS-7.5% and 100CS-15%. All the chitosan:gelatin comparise films displayed a cohesive structure irrespective of the presence of glycerol, which indicates that gelatin, with a low gel-sol transition temperature (around 40 °C), assisted the processing of chitosan by decreasing the overall viscosity. In this study, a film based on pure gelatine out of the obtained by thermomechanical processing as the pure gelatine became liquid due to its low gel-sol transition temperature. All the samples of different chitosan/gelatinerations exhibited a homogenous structure without apparent phase separation, suggesting the excellent compatibility and strong interactions between the two biopolymers.



Fig. 1. SEM of cryo-fracture a runaces of the different chitosan:gelatin films.

3.3 Molecular interaction:

FTIR analysis was conduced to understand the chemical interactions in the different chitosan:gelatin films (**Fig. 2**). In **Fig. 2**a, for raw gelatin, there were absorption bands at 2921 cm⁻¹ (amide B: CH₂ asymmetric stretching), 1630 cm⁻¹ (amide I: C=O stretching), 1526 cm⁻¹ (amide II: 40% C–N and N–H vibration), and 1235 cm⁻¹ (amide III: C–N and N–H vibrations and CH₂ groups) (Abuibaid, AlSenaani, Hamed, Kittiphattanabawon, & Maqsood, 2020; Chen et al., 2020a; Qiao et al., 2017). The broad peak at about 3291 cm⁻¹ and the small peak at 1446 cm⁻¹ are due to –OH groups (Ebrahimi et al., 2019; Sionkowska et al., 2004). For chitosan, the absorption band of the amino group in the region of 3292–3355 cm⁻¹ was masked by the broad absorption band from —OH groups (Sionkowska et al., 2004). The main characteristic absorption bands of chitosan at 1650 and 1259 cm⁻¹ can be assigned to amide I and amide III, respectively. These amide bands suggest chitosan is only partially deacetylated (Sionkowska et al., 2004). The intense absorption bands at 1589 cm⁻¹ are due to the amino group (Yin et al., 2005). The absorption bands at 1060, 991 and 894 cm⁻¹ (C–N stretching) and 1024 cm⁻¹ (C–O stretching vibration) γ re due to the carbohydrate ring skeleton (Kittur, Vishu Kumar, & Tharanathan, 2003; Lawrie et a ., 2t 07).



Fig. 2. FTIR spectra for new chitosan, raw gelatin, and the different chitosan:gelatin films.

Tal	ble	3	FΊ	IR	bands	and	assig	gnments	for	chitosan	and	gelatin.	
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Bands $(cm^{-1})^{a}$	Assignment	Reference	
Chitosan			
3500–3215/3293 (3292)	N-H stretching; O-H	(Lawrie et al., 2007; Qiao et	
	stretching from the saccharide	al., 2017; Sionkowska et al.,	
	structure	2004)	

2932/2925 (2869)	-CH ₂ asymmetric stretching	(Abuibaid et al., 2020;
		Sionkowska et al., 2004)
1643/1645/1648 (1650)	Amide I (C=O groups)	(Chen et al., 2020a; Chen,
		Xie, Tang, & McNally, 2020b;
		Chiono et al., 2008; Lawrie et
		al., 2007; Meng et al., 2018;
		Pereda et al., 2011;
		Sionkowska et al., 2004)
1638–1575/1594/1589 (1589)	Amide II (N–H bending;	(Chen et al., 2020a, b; Chen,
	C–H stretching); Amino	Xie, Tang, & McNally, 2021b;
	group	Chiono et al., 2008; Lawrie et
		al., 2007; Meng et al., 2018;
	.0	Yin et al., 2005)
1414 (1418)	O-H b .nd ag vibrations	(Chen et al., 2020a; Chiono et
		al., 2008; Ebrahimi et al.,
		2019; Meng et al., 2018;
		Pereda et al., 2011)
1377/1378 (1375)	-CH, bending	(Chen et al., 2021b; Chiono et
	2	al., 2008; Ebrahimi et al.,
		2019; Qiao et al., 2017)
1256 (1259)	Amide III (C-N and N-H	(Chen et al., 2020a; Chiono et
	vibrations; CH ₂ groups)	al., 2008; Ebrahimi et al.,
	2	2019)
1150 (1150)	Asymmetric C–O–C	(Chen et al., 2020a; Chiono et
	stretching of the saccharide	al., 2008; Meng et al., 2018;
	structure	Pereda et al., 2011)
1028–1022 (1024)	C–O stretching of the	(Chen et al., 2020a, b, 2021b;
	saccharide structure	Chiono et al., 2008; Meng et

		al., 2018)
1190–920 (1060/991/894)	C-N stretching; saccharide	(Chen et al., 2020a, b; Lawrie
	structure	et al., 2007; Meng et al., 2018;
		Pereda et al., 2011)
Gelatin		
3300–3400/3293 (3291)	Amide-A (N–H); O–H	(Abuibaid et al., 2020;
		Haghighi et al., 2019; Qiao et
		al., 2017)
2932/2925 (2921)	Amide-B (–CH ₂ asymmetri :	(Abuibaid et al., 2020)
	stretching)	
1632/1637 (1630)	Amide I (C=O stret ching)	(Abuibaid et al., 2020; Pereda
		et al., 2011; Qiao et al., 2017;
		Sionkowska et al., 2004)
1554/1537/1535 (1526)	Amide I (40 % C–N and	(Pereda et al., 2011; Qiao et
	60% N—H vibrations)	al., 2017; Sionkowska et al.,
		2004; Yin et al., 2005)
1440 (1446)	0–11	(Ebrahimi et al., 2019)
1200–1240/1233 (1235/1198)	Amide III (C–N and N–H	(Abuibaid et al., 2020;
	vibrations; CH ₂ groups)	Ebrahimi et al., 2019; Pereda
		et al., 2011; Qiao et al., 2017;
		Sionkowska et al., 2004)

^a The numbers in the brackets are FTIR band positions observed in this current study.

Compared with raw chitosan, the processed chitosan film (100CS) displayed the vibrational bands of amide I and amide III shifting to lower wavenumbers (from 1650 and 1259 cm^{-1} to 1638 and 1257 cm^{-1}) and the vibrational bands of the saccharide structure shifting to higher wavenumbers

(from 1060 and 1024 cm^{-1} to 1068 and 1027 cm^{-1}). These shifts indicate that the original structure of chitosan involving the interactions among these groups was disrupted.

Fig. 2b shows that with the addition of gelatin, the band signals of amide I, II and III for chitosan were increased and there was also slight red shifting of the chitosan adsorption bands (1636, 1547 and 1408 cm⁻¹) to lower wavenumbers (1632, 1543, and 1404 cm⁻¹), which indicates the formation of intermolecular hydrogen bonds between gelatin and c'hitosan. Meanwhile, the signal of carbohydrate ring skeleton (1150, 1068, and 1027 cm⁻¹) of 25CS decreased and had a blue shift compared to 100CS. However, Lawrie et al. indicated that is teraction with ionic species did not cause significant change in the band positions that reflect disc carbonyl vibration in alginate (Lawrie et al., 2007).

3.4 Crystalline structure

Fig. 3 shows the XRD patterns for the new materials and the different chitosan:gelatin films. The diffraction pattern for the raw gelatin displayed a diffuse halo centered on $2\theta = 24^\circ$, reflecting the random coiled conformation of the macromolecules (Qiao et al., 2017). The diffraction peaks for the raw chitosan at around $2\theta = 1/2^\circ$ (0 2 0 reflection) and 21° (1 0 0 reflection) can be assigned to its hydrated crystal and regular lattice, respectively (Kittur et al., 2003). Resulting from processing and conditioning, the chitosan film displayed a broad amorphous halo centered around $23^\circ 2\theta$ while the characteristic peaks at $12^\circ 2\theta$ and $21^\circ 2\theta$ of chitosan became unapparent, indicating a predominantly amorphous structure. Some new but weak peaks were observable, which could be due to the formation of new chitosan crystals during conditioning (Chen et al., 2020a). Compared with the 100CS group of films, the composite films were shown to be more amorphous, indicating that gelatin

assisted the processing of chitosan and may also hinder the recrystallization of chitosan during conditioning. This well corresponds to the SEM observation of well-processed chitosan:gelatin samples.



Fig. 3. X-ray diffractograms for a) raw chice an, raw gelatin, and b) the different chitosan:gelatin

films.

3.5 Relaxation temperatures

DMTA was used to obtain the loss tangent (tan δ) profiles as a function of temperature for the different chitosan:gelatin films (**Fig. 4**). All samples exhibited similar tan δ profiles with two obvious transitions identified. The weak transition at this lower temperature region from -50 °C to 0 °C is due to the secondary relaxation (β relaxation) attributed to the motions of side chains or lateral groups of chitosan interacting with small molecules such as water and/or glycerol by hydrogen bonding. A more prominent transition at a higher temperature corresponds to the α relaxation, which can be linked to the glass transition temperature (T_g) of the polymers (Chen et al., 2020a). It can be

seen that 100CS-15% exhibited a significantly lower T_g value than 100CS, indicating increased chain mobility of chitosan due to the plasticization by glycerol. Besides, with increasing gelatin content, T_g first increased and then decreased, with 50CS displayed the highest T_g values (116 °C for 50CS-RH57%). Regarding this, a certain amount of gelatin in the system could restrict the mobility of chitosan chains through electrostatic and hydrogen-bonding interactions to the greatest extent. Compared with 57% RH, the samples conditioned at 75% RH showed lower T_g , which can be attributed to the water-induced plasticization effect and decreased intermolecular forces between chitosan and gelatin chains. Irrespective of formulation and RH, are T_g values of all the samples were above room temperature, indicating that they were in a glacty state.



Fig. 4. DMTA curves for the different chitosan:gelatin films in a) RH 75% and b) RH 57%.

3.6 Mechanical properties

Fig. 5 shows the mechanical properties of the chitosan:gelatin films conditioned at 57% RH and 75% RH. From the stress–strain curves (**Fig. 5a** and **Fig. 5e**), we can see the sample 100CS-15% conditioned at 75% RH was the most elastomeric. Other samples conditioned at 75% RH were stiffer. The 50CS film was the stiffest and showed the most apparent strain-hardening behavior, which could be due to the strong interactions between the two biopolymers. **Fig. 5b**-d shows that the 100CS group had the lowest values of Young's modulus (100CS-15%, 22.6 MI a) a d tensile strength (100CS-15%, 9.0 MPa), while the 50CS group showed the highest (50CS, 505.5 MPa and 33.6 MPa, respectively). For each chitosan/gelatin ratio, addition of glycerol decrea. To the Young's modulus and tensile strength while increased the elongation at break, due to the plasticization effect. In particular, the 50CS film showed the lowest elongation at break, due to the plasticization effect. In particular, the 50CS group of films had the best metal properties, which could be due to the interactions between chitosan and gelatin.



Fig. 5. Mechanical properties of the chitosan:gelatin films conditioned at RH 75% (a-d) and at RH

57% (e-h)

The samples conditioned at RH 57% (**Fig. 5**e-h) were stiffer and more rigid due to a less amount of water molecules in the biopolymers for a plasticization effect, which is as expected. Among these samples, the 50CS and 25CS groups displayed the highest Young's modulus and tensile strength. In this regard, gelatin became more rigid with a lower moisture content, which also contributes to the mechanical properties of the composite system.

3.7 Thermal stability

Fig. 6 shows the TGA results for the raw gelatin, the raw chi. ssar, and the different chitosan:gelatin films. For the raw chitosan, there was a major thermal decomposition peak between about 200 °C and 400 °C with the peak temperature (T_d) being 303 °C, where the maximum decomposition rate occurred. This result for chitosan express with previous reports (Chen et al., 2020a, 2021b; Meng et al., 2018). For the raw gelatin, thermal decomposition occurred mainly between 200 °C and 500 °C with $T_d = 326$ °C. The inermal decomposition of glycerol started at about 120 °C and ending at 233 °C immediately on or the peak temperature.



Fig. 6. Derivative TGA curves for the raw chitosan, the raw gelatin, and the different chitosan:gelatin films. The reference line marks the peak temperature of the raw chitosan.

The 100CS film displayed one major thermal decomposition peak with $T_d = 275$ °C, about 28 °C lower than that of raw chitosan. Besides, there was a small peak at about 250 °C which might be attributed to the initial depolymerization of chitosan. The decreased thermal stability of 100CS could be ascribed to the reduced molecular mass and lower crystallinity resulting from processing.

Compared with 100CS, the composite samples containing gelatin displayed slightly higher T_d , which could be associated with higher thermal stability of the processed gelatin. In other words, the thermal stability of gelatin was less affected by processing than that of chitosan. For the composite films, the addition of glycerol did not change the thermal stability significantly.

3.8 Water absorption and surface hydrophilicity

Fig. 7a shows that all the samples experienced an initial rapid increase in weight during the first 40 min of water soaking and after that point, water absorption continued at a much slower rate. The maximum value of water uptake followed the sequence of 1.0Cs group > 75CS group > 25CS group > 50CS group. After 6 h of water soaking, the 100CC 11m absorbed water more than 200% as much as its original weight, while the 50CS film has where absorption of about 45%. The lowest water absorption percentage of the 50CS film ould be accounted for by the intermolecular interaction between gelatin and chitosar, a hough each biopolymer was more hygroscopic. A recent study suggested that chitosan:carbox, methyl cellulose films were more hydrolytically stable than each of the biopolymer component, which was ascribed to the polyelectrolyte complexation between the two biopolymers, stabilizing the chain network structure in water (Chen et al., 2020a). For the reduced water absorption of the 50CS film here, we similarly consider that the electrostatic interactions between chitosan and gelatin led to complexed points for the chain network, which decreased the water swellability and water absorption of the materials. Regarding the higher water absorption of 50CS-7.5% and 50CS-15%, it is proposed that glycerol weakened the electrostatic interaction between chitosan and gelatin.



Fig. 7. a) Water absorption percentage of the different biopoly ner "ilms; b) Water contact angle measured at 0 s and 60 s for the different "husan:gelatin films.

Fig. 7b shows the water contact angle (WCA) re. ulls of the different chitosan:gelatin films. A higher contact angle means higher surface hydrophilicity, which is largely determined by the surface free energy linked to the chemical groups, sposed on the material surface (Chen et al., 2021b). Since a water drop on the films greatly challed with time, WCA was measured at 0 s and 60 s. All the composite films displayed similar V CA values at 0 s (92–115°) and at 60 s (82–101°). Regarding the reduced WCA with time, a write wetting, water could disrupt biopolymer chain interactions on the material surface, leading to more free polar groups available to bind water (Chen, Xie, Tang, & McNally, 2021a; Chen et al., 2021b). Among the biopolymer films, the 50CS group had the lowest WCA values, indicating higher surface hydrophilicity. For biopolymers such as gelatin, the high chain mobility would allow for the burying of polar groups in the bulk phase, making the surface more hydrophobic (Yasuda, Sharma, & Yasuda, 1981). Thus, we propose that the intermolecular interaction between the two biopolymers in the 50CS composites could reduce the freedom of the

biopolymers to rearrange themselves to change the material surface configuration. As a result, the 50CS group of films had more polar groups exposed on the material surface and thus, higher surface energy. Moreover, the surface hydrophobicity was reduced with an increasing amount of glycerol, which could be linked to the high hydrophilicity of this plasticizer.

4 Conclusion

In summary, we prepared chitosan:gelatin films by a cost-effective thermomechanical method. We found the properties of chitosan:gelatin films were strongly a cost-effective by the chitosan/gelatin ratio, glycerol content, and the conditioning RH. We found a certain ratio of chitosan to gelatin (i.e. the 50CS group of films) led to the lowest water uptake, best mechanically properties, and the highest T_g values. This property enhancement could be accribed to the strong interactions (e.g. ionic and hydrogen-bonding) between the polysaccharide and the protein as proved by FTIR analysis.

Despite the significantly reduced hyg o copicity of the 50CS group, these formulations displayed the highest surface hydrophilicity, which could be due to the reduced material surface configuration allowing more r^{-1} are groups exposed on the material surface. This contrast shows the surface hydrophilicity and ov rall hygroscopicity are controlled by different mechanisms for biopolymer materials. Besides, addition of gelatin made chitosan more amorphous, which could be due to the restriction of chitosan chain rearrangement by gelatin. While the pure chitosan films were the darkest and most opaque, composite films with a higher content of gelatin were brighter and more transparent, and thus better visual appearance. Glycerol, as a plasticizer, assisted the processing of chitosan as shown by SEM, and gelatin was found to have a similar effect.

Thus, this study could be insightful for the design of cost-effective biopolymer materials with

tailored properties for specific applications (e.g. packaging, coating, and biomedical).

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Declaration of interests

None.

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Graphical abstract



Highlights:

- ✓ Chitosan:gelatin composite materials processed by high-viscosity thermo-mixing
- ✓ Biopolymer material properties controlled by formulation and relative humidity
- ✓ 1:1 Chitosan:gelatin had the best mechanical properties and lowest hygroscopicity
- ✓ 1:1 Chitosan:gelatin film showed the highest surface hydrophilicity
- ✓ Chitosan added with gelatin had improved processibility and higher transparency