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Field-dependent band-structure measurements in two-dimensional heterostructures

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KEYWORDS. Two-dimensional materials, two-dimensional heterostructures, two-dimensional semiconductors, angle resolved photoemission spectroscopy

ABSTRACT

In electronic and optoelectronic devices made from van der Waals heterostructures, electric fields can induce substantial band structure changes which are crucial to device operation but cannot usually be directly measured. Here, we use spatially resolved angle-resolved photoemission spectroscopy to monitor changes in band alignment of the component layers, corresponding to band structure changes of the composite heterostructure system, that are produced by electrostatic gating. Our devices comprise graphene on a monolayer semiconductor, WSe₂ or MoSe₂, atop a boron nitride dielectric and a graphite gate. Applying a gate voltage creates an electric field that shifts the semiconductor bands relative to those in the graphene by up to 0.2 eV. The results can be understood in simple terms by assuming that the materials do not hybridize.

Layers of van-der-Waals materials stacked in two-dimensional heterostructures (2D-HS)¹ can be used to achieve transistor operation with high on-off ratio^{2,3}, efficient light emission or absorption⁴, and switchable phase transitions^{5,6}. The electronic and opto-electronic properties of 2D-HS can be tailored by selection of the constituent layers based on their band structures and relative band alignments, as well as by control of geometry such as their relative rotation. Full working 2D electronic devices can be made by incorporating graphene or graphite as conductors, enabling application of large electric fields which modulate doping and band structure^{4,7}. For electronics and optoelectronics, graphene offers distinct advantages as the metal electrodes in 2D semiconductor heterostructures; graphene is flexible, transparent, easily stacked and integrated into the heterostructure, forms interfaces to the 2D semiconductors without dangling bonds and hence enables Fermi-level depinning, and has a work-function that can be tuned by electric field⁸. As a result, the integration of graphene contacts with 2D semiconductors has shown great promise for electrical transport^{9–20} and light-dependent / light-emitting 2D devices^{4,21–31}.

Understanding and developing such devices requires tools that probe electronic structure during device operation. For example, the band alignments at the graphene/MX₂ interface determine the the barriers to charge injection, which can change with applied electrostatic field. Although the field-dependent band structure at the graphene/MX2 interface has been calculated by ab initio modelling^{32–38}, these need testing experimentally. Gate-dependent changes in optical properties are relatively straightforward to measure, but do not directly probe the electronic spectrum. Scanning tunneling spectroscopy can be combined with gating, though it reveals only momentumintegrated densities of states in the uppermost layer.^{7,39,40} Efforts have been made to extract Schottky barrier heights from transport data^{14,18,41,42}, but these are model-dependent and indirect. On the other hand, angle-resolved photoemission spectroscopy with sub-micrometer spatial resolution (µ-ARPES) permits measurements of the full momentum-resolved electronic structure in 2D-HSs, to a depth of several layers. From this, the band alignments relative to the Fermi energy, and band offsets between layers, can be determined in mechanically exfoliated 2D heterostructures. It has recently been shown that µ-ARPES can be performed with operating gate electrodes^{43–49}, allowing investigation of in situ doping of the conduction band. Here we apply μ -ARPES to study electric-field induced changes in band alignments in back-gated graphene/monolayer transition metal dichalcogenide (MX₂) heterostructures.

Mechanical exfoliation and dry transfer were used to fabricate 2D-HS devices, which consist of top graphene, an MX₂ layer, a hexagonal BN dielectric (thickness D_{BN}), and a graphite back-gate electrode, as shown schematically in Figure 1a. The fabrication process was as described in earlier work,⁴³ using a polycarbonate film on polydimethylsiloxane stamp to pick up and align the flakes before depositing the stack on a substrate with predefined metal electrodes (Pt with Ti adhesion layer). Figure 1b is an optical microscope image of a completed device. Each device was wirebonded in a ceramic chip carrier (8-lead side-brazed package, CSB000815 from Spectrum Semiconductor Materials Inc.) and mounted on a custom-made sample plate that allowed in-situ electrical connections at the Spectromicroscopy beamline of the ELETTRA synchrotron. Samples were annealed in ultra-high vacuum (UHV) to around 400 °C for several hours immediately prior

to transfer to the analysis chamber. Further details are given in supporting information (SI) section S1. Measurements were acquired at a sample temperature of around 100 K, with a photon energy of 27 eV. A Schwarzschild mirror was used to focus the beam down to a sub-micrometer spot and the photoemitted electrons were collected by a hemispherical analyzer mounted in the UHV chamber on a 2-axis goniometer.⁵⁰ Scanning photoemission microscopy⁵¹ was used to find the regions of interest, where the monolayer graphene top contact overlapped both the monolayer MX₂ and the gate electrode. These areas were typically a few microns across, much bigger than the beam spot.

Angle-resolved photoemission spectra were acquired while the back-gate voltage V_G was varied *in situ* with the graphene grounded. ARPES is sensitive to the uppermost few atomic layers of the sample;^{52,53} hence the spectra include photoemitted electrons from both the graphene and the underlying MX₂ layer. The hemispherical analyzer, with two-dimensional detector, was positioned to acquire spectra near the symmetry points of the Brillouin zones of the graphene and MX₂ layers, allowing analysis of the valence band dispersion in each layer. Figure 1c shows the Brillouin zones of the graphene and monolayer WSe₂, and Fig. 1d shows sketches of the bands along the three colored lines at the zone center (Γ , red), the zone corner of WSe₂ (**K**_W, blue), and the zone corner of graphene (**K**_{gr}, green line). The behavior of the spectra along these lines, as a function of gate voltage, is shown in the energy-momentum spectra in Figures 1e-g. As the gate voltage goes from positive to negative, the bands at each position shift up in energy. For graphene, there is a clear change from electron-doping at positive gate voltage, with the Dirac point energy below the Fermi energy. Further data are shown in SI Section S2.

At each gate voltage we determined E_{Γ} , the WSe₂ valence band maximum at Γ , from a parabolic fit to the intense photoemission around Γ , and similarly $E_{\rm K}$ from the weaker photoemission around K_W . The Dirac point energy, E_D , of the graphene was found from the crossing points of linear fits to each side of the conical graphene dispersion around K_{gr} . As usual, the Fermi energy, E_F , was determined by fitting the drop in intensity across the photoemission threshold with a Fermi-Dirac distribution, and the (binding) energy is plotted relative to $E_{\rm F}$. Further details of the data processing procedures used to extract these band edge energies are described in SI section S4. We find E_{Γ} – $E_{\rm K} = 0.55 \pm 0.05$ eV, and the spin splitting at $K_{\rm W}$ (see Fig. 1d) is $\Delta_{\rm SOC} = 0.48 \pm 0.03$ eV, consistent with our previous measurements^{43,51} on ungated monolayer WSe₂. At $V_G = 0$, the valence band edge is at $E_V = E_K = -0.85 \pm 0.05 \text{ eV}$, slightly below the value of $-0.80 \pm$ 0.01 eV in our previous measurements⁴³ where there was no top graphene.⁵⁴ Since monolayer WSe₂ on hBN has a direct gap⁴³ of approximately 2.1 eV, this means that the chemical potential is deep in the gap of the WSe₂. With the WSe₂ in contact with the graphene layer, and $E_V = E_K$ measured relative to the Fermi level in graphene, E_K is a direct measure of the Schottky barrier height for hole injection at the graphene / WSe₂ contact. Although E_K could change depending on the density of interfacial traps and defects, more can be learnt from the dependence of E_K on V_G .



Figure 1. Gate-dependent valence band measurements in a 2D heterostructure. (a) Schematic, and (b) optical microscope image, of a 2D-HS device (graphene/WSe₂, $D_{BN} = 25.5 \pm 0.2$ nm) with electrical connections made in the ARPES chamber. The scale bar is 50 µm. (c) First Brillouin zones of the monolayer WSe₂ (blue hexagon) and graphene (green hexagon), with the positions of the energy-momentum slices in the lower panels marked as solid-colored lines: red, zone center; blue, zone corner of the WSe₂; and green, zone corner of the graphene. (d) Schematic of the bands along those colored lines. Gate-dependent energy-momentum slices around **Γ**, (e), **K**_W, (f), and **K**_{gr}, (g): gate voltages as labelled, the scale bars correspond to 0.2 Å⁻¹.

The variation of the graphene and WSe₂ band parameters with V_G is plotted in Fig. 2a. The graphene Dirac point E_D moves downwards in an S-shaped manner as V_G is increased, as expected due to the linearly vanishing density of states near E_D ,⁸ with an overall shift of 0.6 eV. The WSe₂ valence band edges E_{Γ} and E_K also follow an S-shape, with no detectable change in $E_{\Gamma}-E_K$ (nor of Δ_{SOC}); however, their overall shift is larger, at 0.8 V. It also appears that the inflection point occurs at different V_G for E_D and E_K , but this is most likely a result of spatial variations in the sample doping between the different locations where the measurements were made. Fig. 2b shows similar measurements made using a device with monolayer MoSe₂ instead of WSe₂. The energy-momentum slices and further device details are given in SI section S3. Spectra were only acquired from the MoSe₂ bands near Γ , because the photoemission intensity at the zone corner was too low. For the monolayer MoSe₂ device, at $V_G = 0$, the valence band edge is at $E_K = -1.24 \pm 0.05$ eV, lower in energy than for WSe₂. Combined with the smaller gap for MoSe₂ than WSe₂, this indicates slight electron-doping of the MoSe₂ compared to WSe₂. This could indicate a low concentration of gap states, but E_K is highly dependent on the energy of the gap states as well as their density⁵⁵, even at low densities of gap states, such that the doping level cannot be determined from E_K alone.



Figure 2. Gate-dependence of the band energies in graphene/MX₂ heterostructures. Values of E_D (green), E_K (blue) and E_{Γ} (maroon) extracted from the data in Figure 1, plotted as a function of V_G , for (a) graphene on monolayer WSe₂ ($D_{BN} = 25.5 \pm 0.2 \text{ nm}$) and (b) graphene on monolayer MoSe₂ ($D_{BN} = 19.0 \pm 0.5 \text{ nm}$). The dashed lines are fits to the data as described in the text. In (b), E_K is calculated from the fit to E_{Γ} assuming a constant energy difference of $E_{\Gamma} - E_K = 0.44 \text{ eV}$. The data processing procedures used to extract these band edge energies are described in SI section S4.

We can interpret these band shifts using the schematic band diagrams shown in Fig. 3. No signs of hybridization are evident, so we treat the bands in the WSe₂ and graphene as separate. At $V_G = 0$ (Fig. 3a), for both WSe₂ and MoSe₂, the graphene Dirac point E_D is very close to the Fermi level (zero), implying that the graphene is undoped and the electric field in the hBN is small. The position of the valence band edge E_K in the monolayer MX₂ at that condition is defined to be E_{K0} .

Under a positive gate voltage V_G applied to the graphite (Fig. 3b) there is an electric field in the hBN and the graphene is doped with electrons such that $E_D = -\mu$, where μ is the chemical potential in the graphene. The total areal charge density in the graphene plus MX₂ is to a good approximation CV_G , where $C = \epsilon_{BN}/D$ is the geometric capacitance of the hBN with thickness D_{BN} and dielectric constant⁴³ ϵ_{BN} . Thus $e \int_0^{\mu} n(E) dE \approx CV_G$, where e is the magnitude of the electron charge and $n(E) = n_{gr} + n_{gap}$ is the sum of the density of states of the graphene, $n_{gr} = \frac{2}{\pi(hv_F)^2}E$, and of in-gap states in the MX₂, n_{gap} . For the WSe₂ device ($D_{BN} = 25.5 \pm 0.2$ nm) we obtain a good fit to the measurements of E_D vs V_G by taking $n_{gap} = 0$ (green dashed line in Fig. 2a), and we can infer that n_{gap} in the monolayer WSe₂ is less than 10¹¹ cm⁻² consistent with previous reports that have also found a low density of traps and gap states at the MX₂/BN interface^{55,56}.



Figure 3. Inferring the gate-dependent electrostatic potential drop, Δ , between the MX₂ and the graphene. (a) and (b), Schematic band diagrams across the heterostructure at zero and positive gate voltage $V_{\rm G}$, respectively. (c) and (d), The electrostatic potential difference between states in the MX₂ and the top graphene layer, determined from the band alignment data in Figure 2, plotted as a function of the electrostatic potential drop across the hBN for WSe₂ and MoSe₂ heterostructures, respectively. The dashed lines are linear fits to the data. Further details on determining Δ are described in SI section S4.

To understand the band shifts in the MX₂ layer, both the change in chemical potential in the graphene, and hence its change in work-function,⁸ and the electric field across the heterostructure must be considered. With the MX₂ layer in contact with the graphene, the MX₂ valence band edges shift with E_D and hence also follow an apparent S-shape. But their overall shift is larger than the shift in E_D : the shift of the MX₂ bands relative to the graphene Dirac point observed in Figs. 1 and 2 is a natural consequence of the gate electric field. Some of the electric flux passes through the MX₂ and terminates on the graphene, creating an electrostatic potential energy difference Δ between the electron states in the MX₂ and those in the graphene. In Fig. 3b we use a red dashed line to indicate the corresponding variation of the electrostatic potential ϕ across the stack. (The change in slope at the MX₂ layer corresponds to polarization charge in the dielectrics plus the charge in in-gap states). The total potential drop across the heterostructure is equal to the electrostatic potential drop plus the change in chemical potential in the graphene layer, sometimes

described as adding the geometric capacitance in series with the quantum capacitance of the graphene layer⁸. Assuming the polarization of the insulators is in linear response, Δ will be proportional to the total potential drop $(V_{\rm G} - \mu/e)$ between the gate and the graphene, that is, $\Delta = \alpha (eV_{\rm G} - \mu)$, where α is a numerical constant. From Fig. 3b one can see that $E_{\rm K} = E_{\rm K0} - \mu - \Delta$ and so, using $\mu = -E_{\rm D}$,

$$\Delta = E_{\mathrm{K}0} - E_{\mathrm{K}} + E_{\mathrm{D}} = \alpha (eV_{\mathrm{G}} - E_{\mathrm{D}}) \,.$$

Therefore, a plot of $\Delta = E_{K0} - E_K + E_D$ versus $eV_G - E_D$ should yield a straight line of slope α . Such plots for both WSe₂ (Fig. 3c) and MoSe₂ (Fig. 3d) devices do indeed yield straight lines whose best fit slopes are $\alpha = 0.012 \pm 0.002$ and 0.017 ± 0.001 respectively. (In making these plots we compensated for the doping inhomogeneity; see SI section S4).

To see that these measured values of α are reasonable, let us consider a layered capacitor containing a hBN slab of thickness D_{BN} and an MX₂ slab of thickness $d \ll D_{BN}$. The fraction of the potential applied to the capacitor that drops across the MX₂ slab is $\alpha' \approx \frac{d}{D_{BN}} \frac{\epsilon_{BN}}{\epsilon_M}$, where ϵ_M is the dielectric constant of the MX₂. Taking *d* as the monolayer spacing which is 0.65 nm⁵⁷⁻⁶⁰, $\epsilon_M \approx 10$, $\epsilon_{BN} \approx 4$, and $D_{BN} \approx 25$ nm (as in the WSe₂ device) gives $\alpha' \approx 0.014$. Using $D_{BN} \approx 19$ nm (as in the MoSe₂ device) gives $\alpha' \approx 0.018$. Although this is only a very rough model of our structure, these estimates of α' are consistent with the measured values of α , implying that the dielectric-slab approach gives a practical way to estimate electric-field induced band shifts in heterostructures.

With the validation given by the results presented here, we expect this dielectric-slab approach to be generically applicable for estimating band shifts in other 2D heterostructures as long as interactions between the layers are weak. The quantitative predictions of this model will enable device design to optimize changes in Schottky barrier height and hence increase the ON/OFF ratio in graphene-contacted MX₂ field effect transistors³³, for example. The model does not consider interactions between layers such that, for example, it does not depend on the twist-angle between the graphene and MX₂ layers. With the significant difference in lattice parameter between graphene (0.25 nm) and monolayer MX₂ (0.33 nm), a moiré superlattice potential is not expected. In systems where moiré effects change the density of states near the Fermi energy, these would need to be considered and could be probed by field-dependent band structure measurements using μ -ARPES.

In conclusion, we have demonstrated that μ -ARPES can be used to observe and quantify changes induced by an applied electric field in the band structure of 2D heterostructures. The changes we observed correspond to simple shifts of the bands associated with component monolayers which do not significantly hybridize (graphene/MX₂), and illustrate the significance of the sequence of the layers³² as well as their individual properties. This being established, the technique can in the future be applied to gain insights into other heterostructures that host hybridized bands and moiré superlattices, such as the flat bands associated with diverse correlated and topological states in multilayer graphene and MX₂/M'X'₂ with appropriate twist angles.

ASSOCIATED CONTENT

Supporting Information. Sample fabrication and preparation; gate dependent spectra from graphene on monolayer WSe_2 heterostructure; sample details and gate-dependent spectra from graphene on monolayer $MoSe_2$ heterostructure; further details on determining the electrostatic potential drop across the MX_2 layer.

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The manuscript was written with contributions from all authors.

ACKNOWLEDGMENT

Research on monitoring gated electronic structure changes is supported as part of Programmable Quantum Materials, an Energy Frontier Research Center funded by the U.S. Department of Energy (DOE), Office of Science, Basic Energy Sciences (BES), under award DE-SC0019443. N.R.W. and N.D.M.H. were supported through U.K. Engineering and Physical Sciences Research Council (EPSRC) award EP/P01139X/1. N.C.T., N.Y. and A.J.G. were supported through EPSRC studentships (EP/M508184/1 and EP/R513374/1). X.Xia was supported by a University of Warwick studentship. We thank Gabriel Constantinescu and Nicholas Hine for helpful discussions.

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