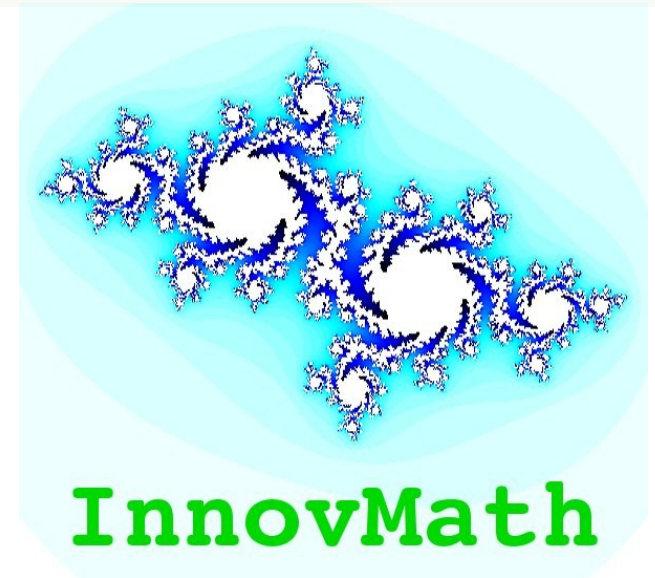


*Understanding localised corrosion phenomena
by using the synergies of thermodynamics and diffusion*

Michael Auinger

July, 3rd 2014



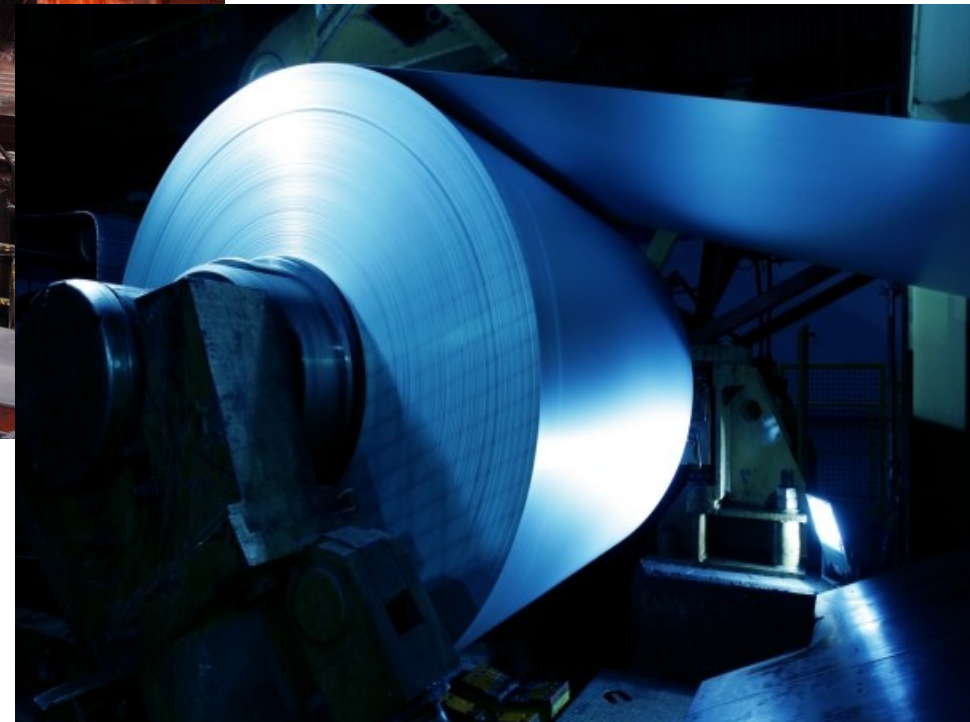
Presentation at: GTT Users Meeting 2014
Herzogenrath (Aachen, Germany)

- Introduction
 - Finding Localised Corrosion Phenomena
 - Modelling Approach

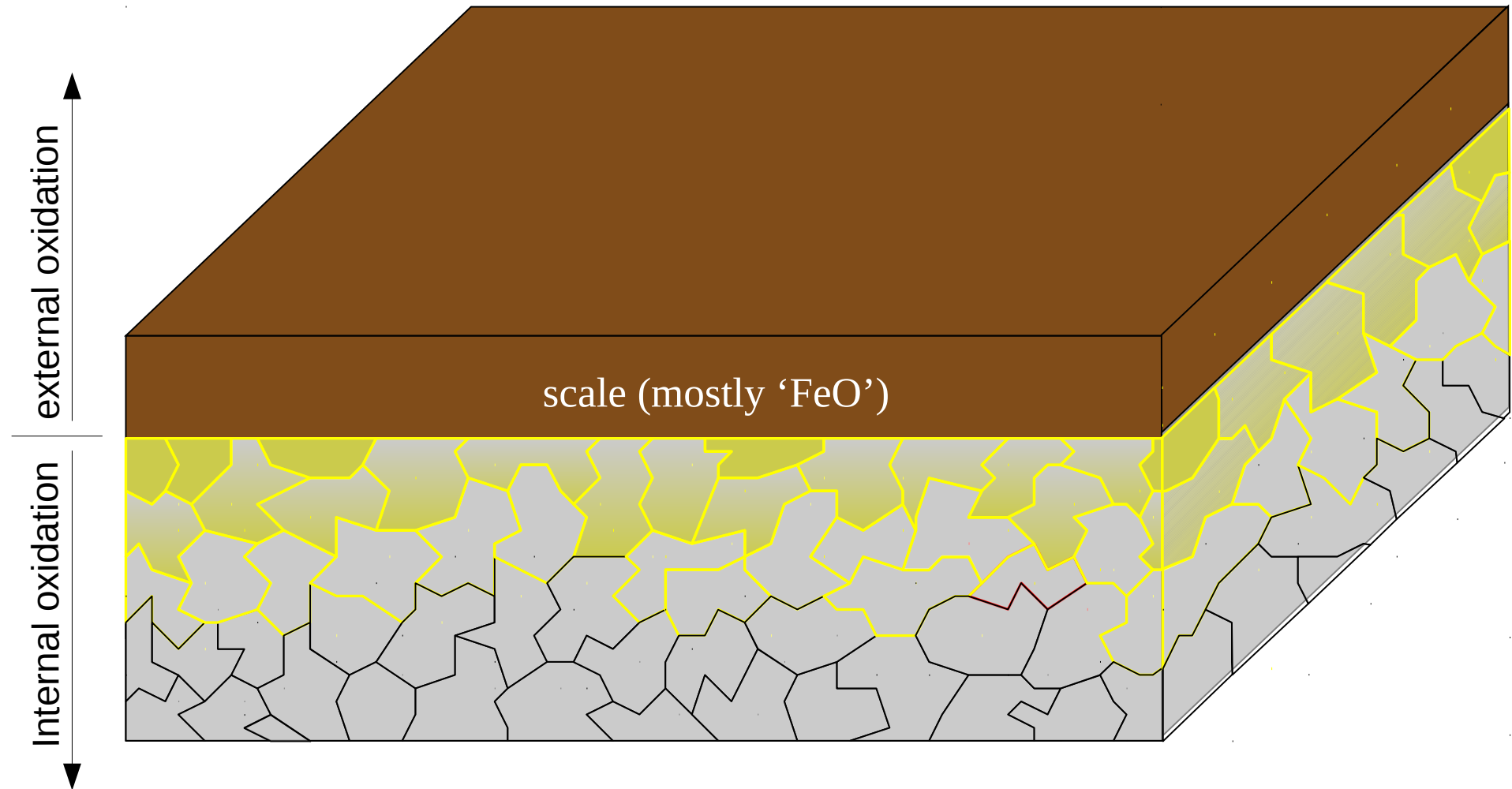
- Examples
 - Decarburation of Pearlite
 - High Temperature Corrosion
 - Hydrogen in Steels
 - Electrochemical Corrosion

- Summary

Industrial Hot-Rolling Process



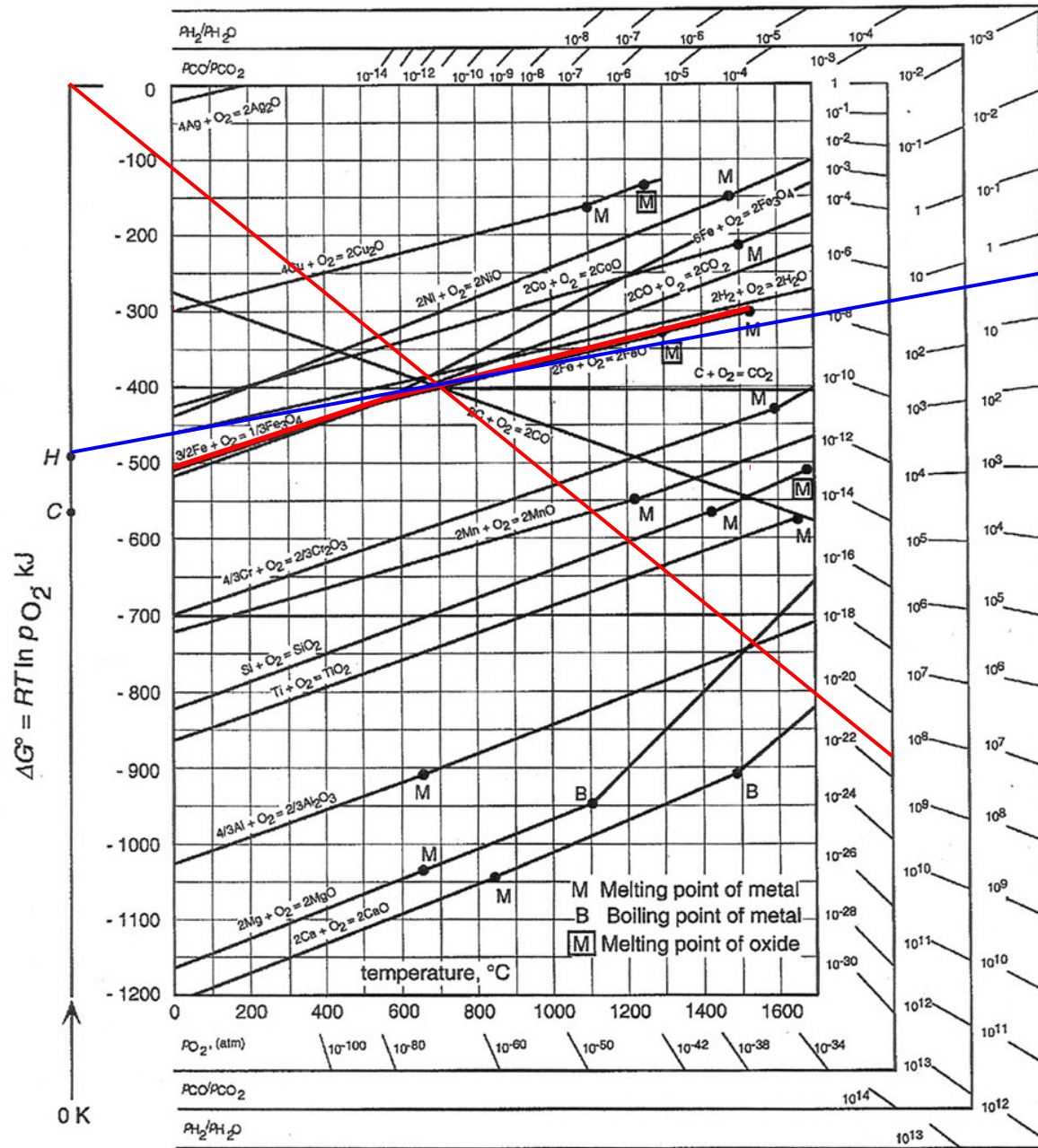
(Internal) Oxidation Scheme



Ellingham-Diagram



InnovMath



History of Diffusion

~1600 Japan: Mokume-Gane (木目金)

Combination of two (or more) metals by “diffusion welding”



Knife and Sword



Rings and Jewellery

(Recent) History of Diffusion



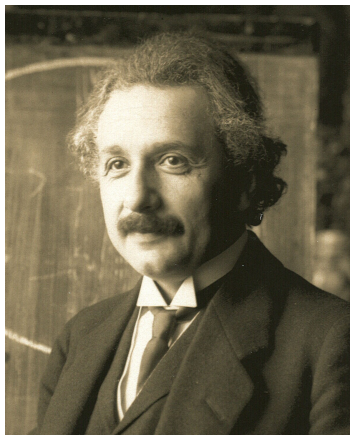
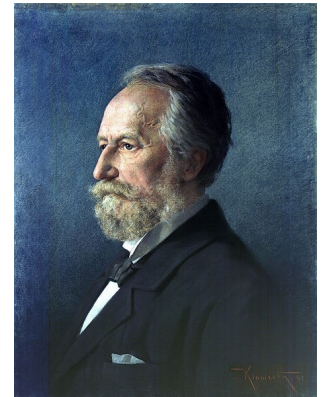
InnovMath

From the latin word *diffundere* = disperse, spread

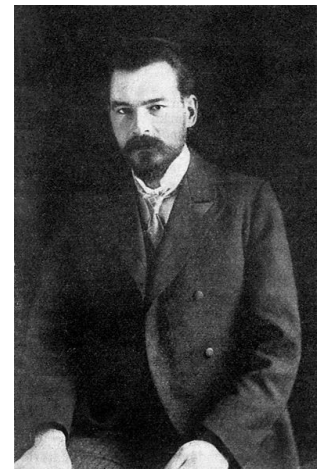


1833 *Thomas Graham*
first observations and hypotheses

1855 *Adolf Fick*
first empirical formulas



1905 *Albert Einstein*,
1906 *Marian Smoluchowski*
generalized theory



$$D \propto \frac{X^2}{t}$$

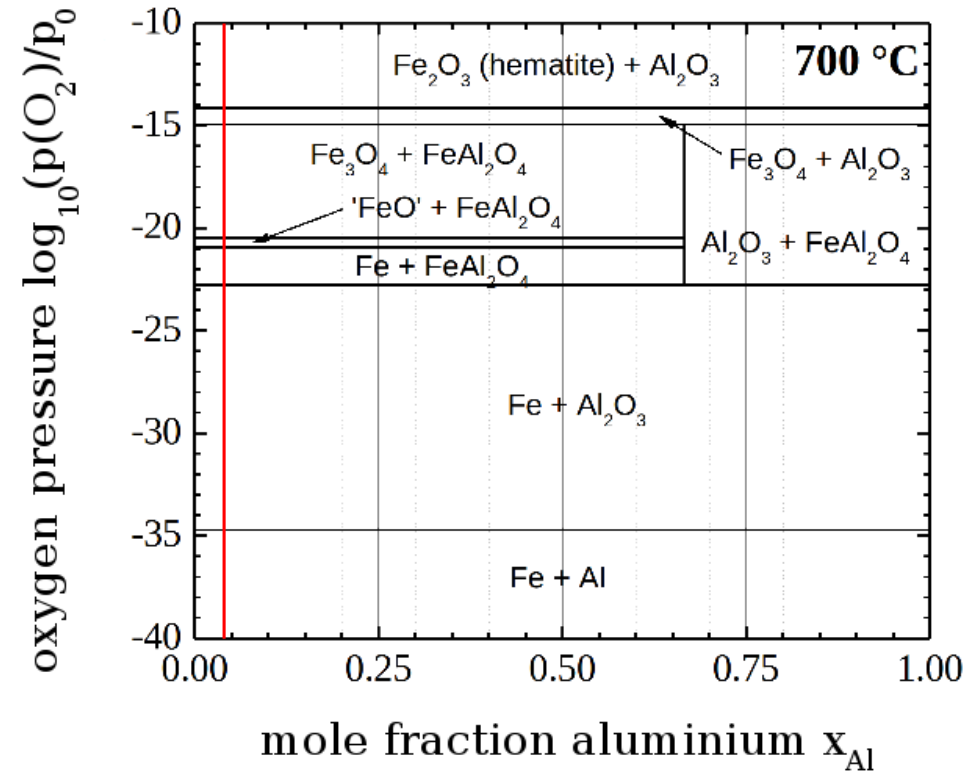
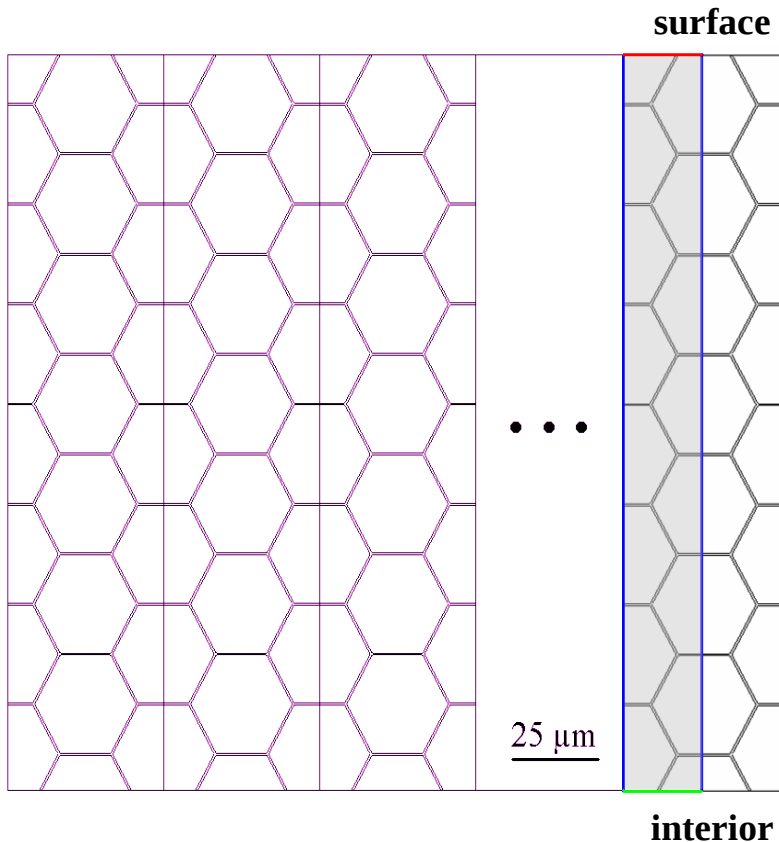
Program Structure



element transport

chemical reaction

$$\frac{dc_{i(x,t)}}{dt} = \text{div}(D_{i(x,T)} \cdot \nabla c_{i(x,t)})$$



Program Output (Fe-Mn-Cr alloy)

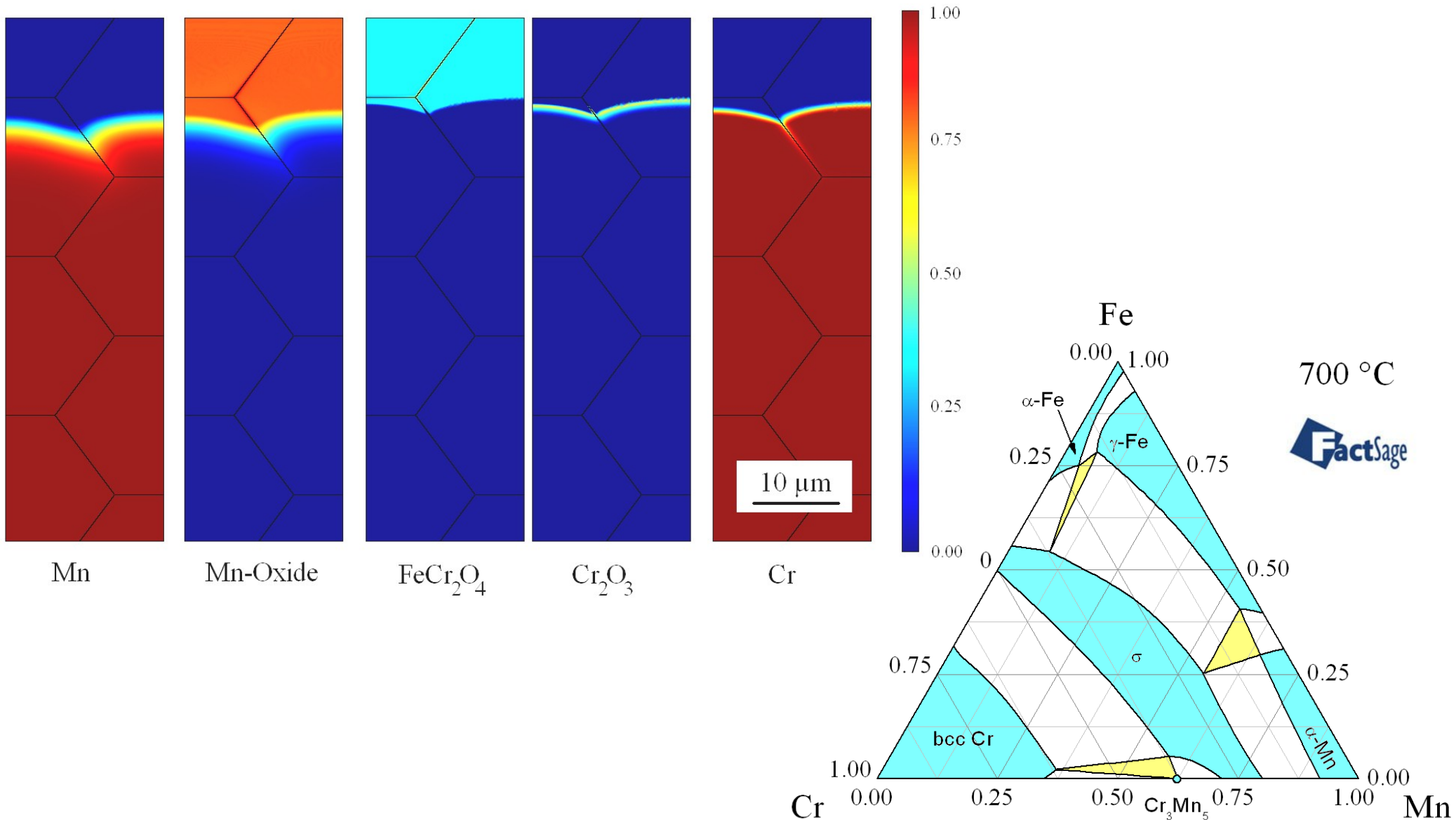


Figure: Spatial phase distribution in Fe, 2 wt-% Mn, 0.8 wt-% Cr after oxidation at $p(\text{O}_2) = 3 \cdot 10^{-22}$ bar and 700 °C for 120min (left) and ternary phase diagram (right).

Segregation

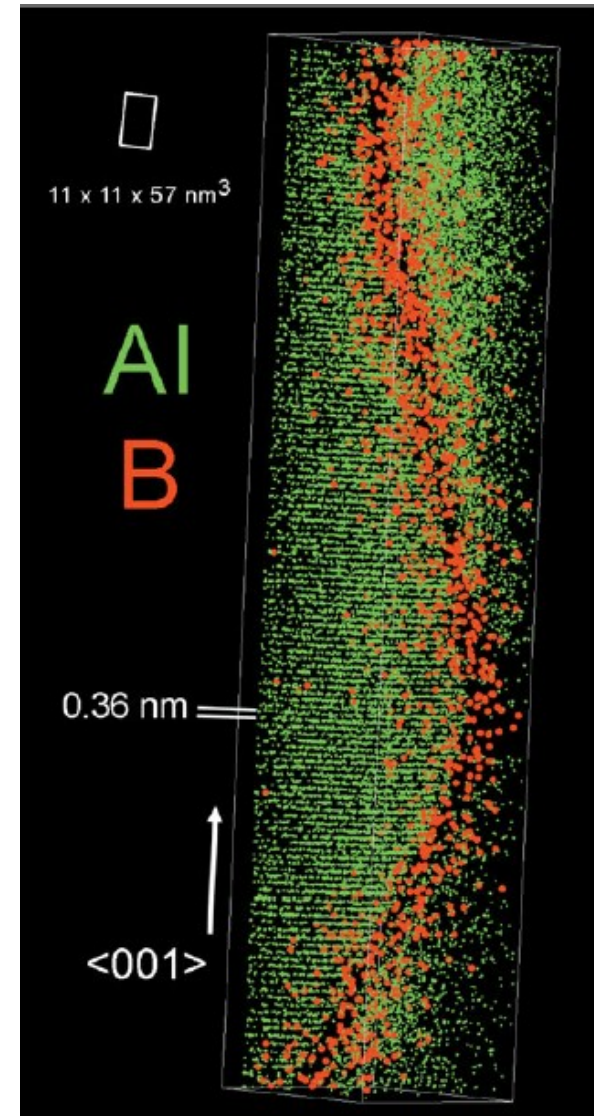
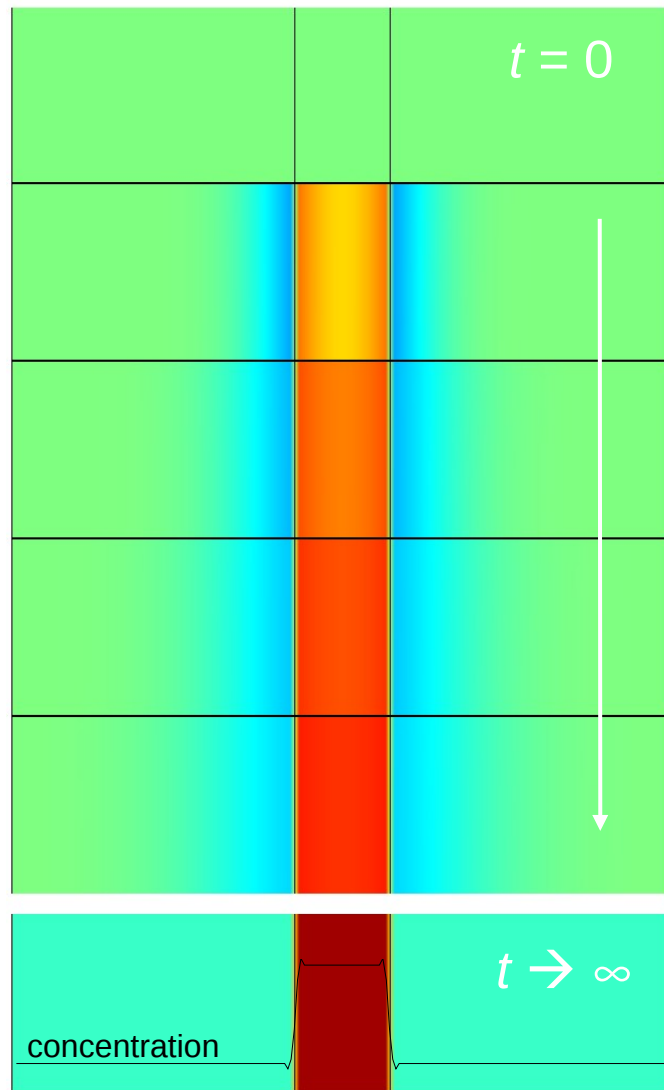


Figure: Numerical simulation of „uphill-diffusion“ (left) and 3D atom probe tomography of segregated boron atoms in a NiAl grain boundary (right).

Thermodynamics in real life



$$J_A = -D \nabla c$$

$$J_A = -L \nabla \mu$$

$$J_A = -L \nabla \mu = -L \frac{\partial \mu}{\partial c} \nabla c = \dots = -L \underbrace{\frac{RT}{c}}_D \nabla c - L \left(\nabla \mu^\circ + \frac{RT}{\gamma} \nabla \gamma \right)$$

Figure: Street crossing in Shibuya (渋谷, Japan) at red and green traffic signal.

Oxide and Nitride Formation

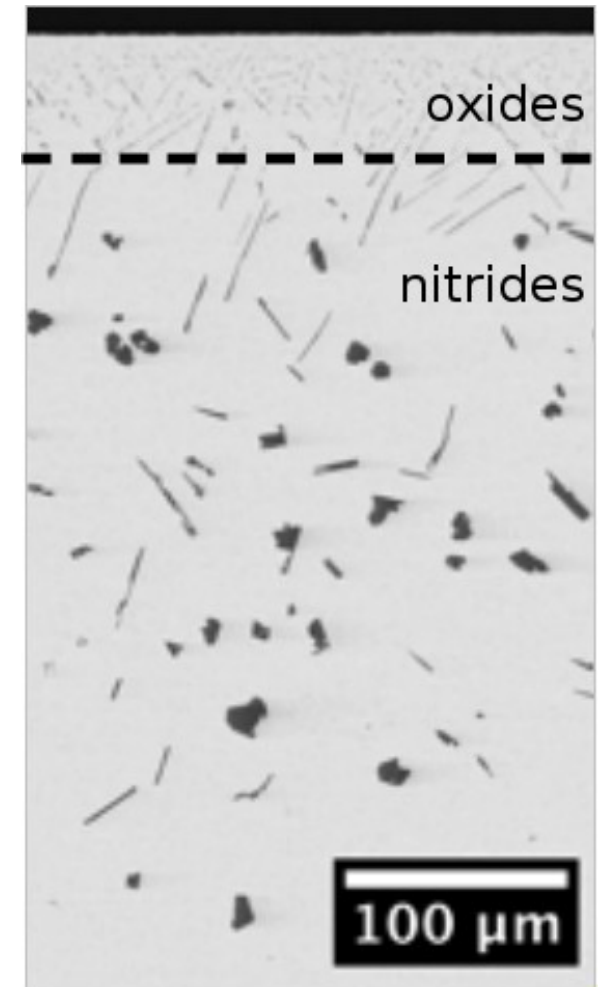
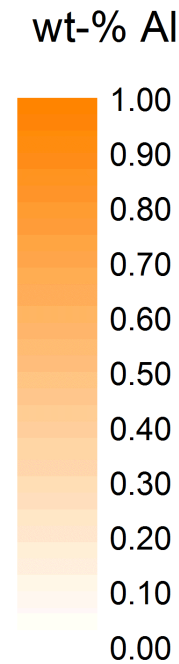
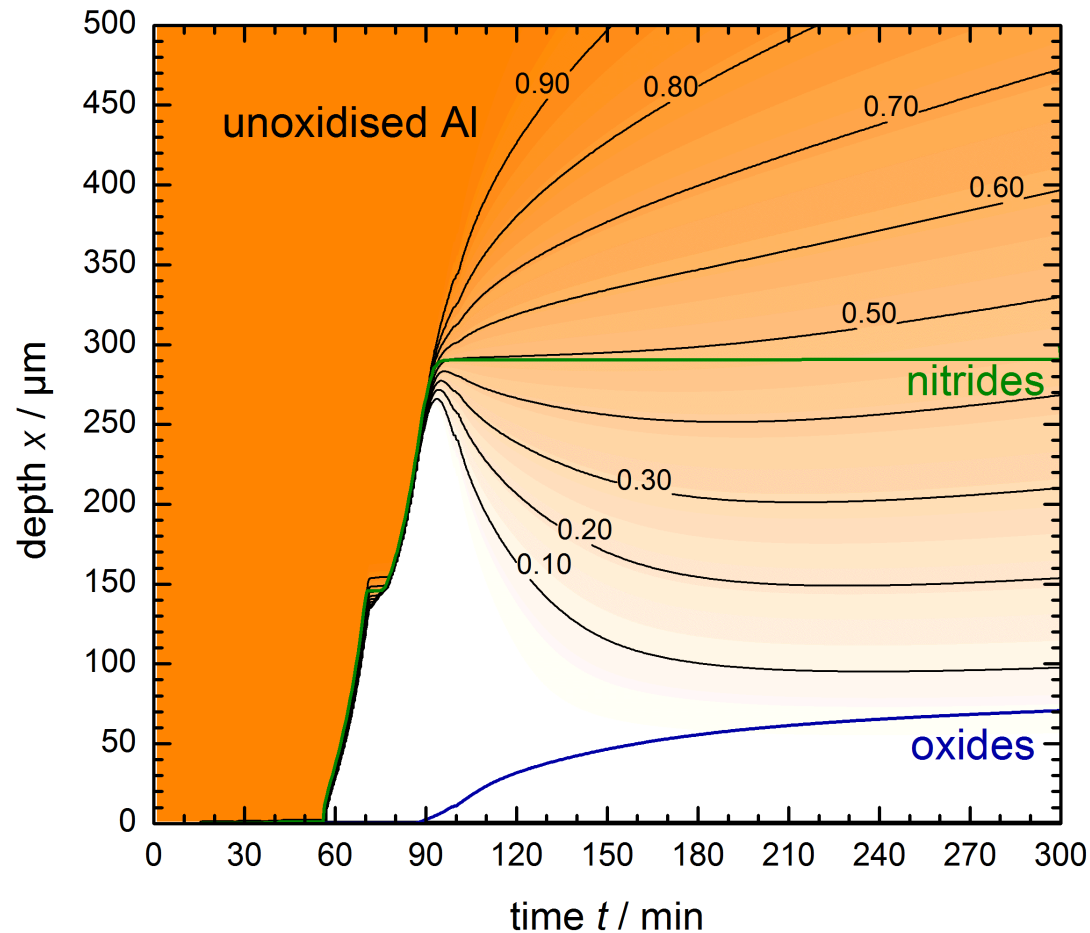


Figure: Calculated corrosion depth in Fe, 1wt-%Al, oxidised with $\text{N}_2 / 2.5\text{vol}\text{-}\% \text{H}_2 / 0.1\text{vol}\text{-}\% \text{H}_2\text{O}$ (left) and SEM-picture of the cross section after 300min (right).

Oxide and Nitride Formation



InnovMath

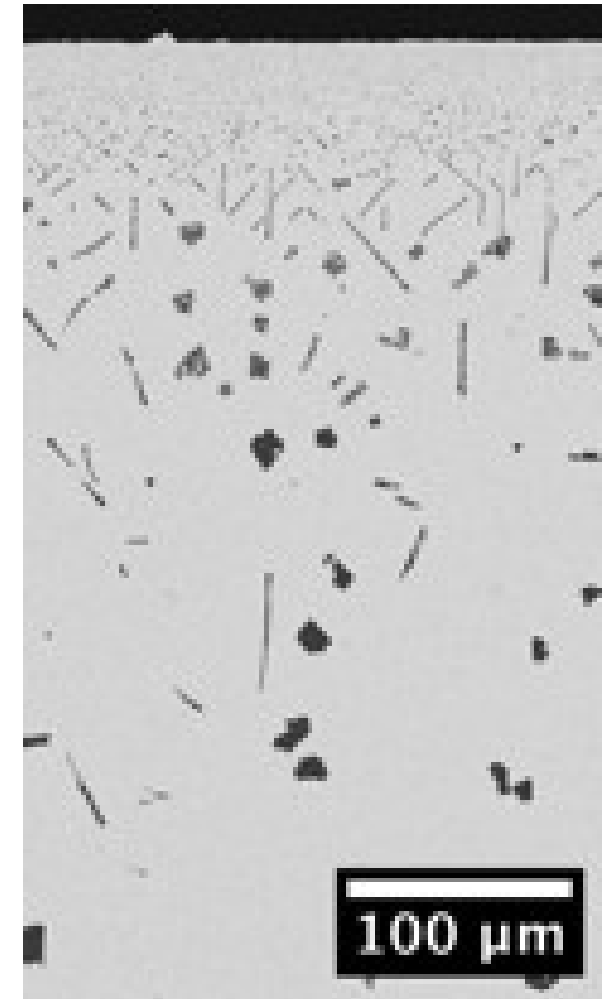
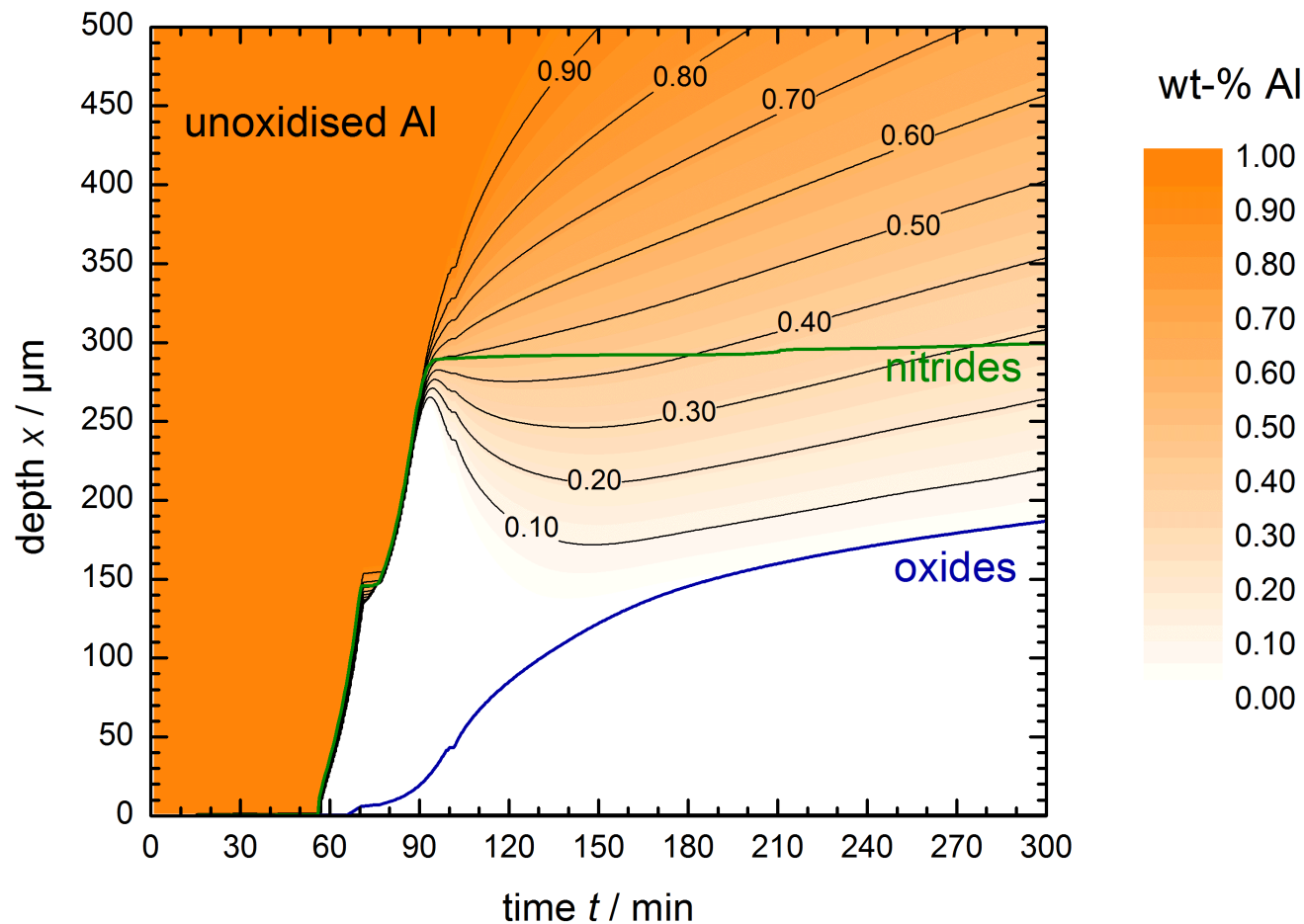


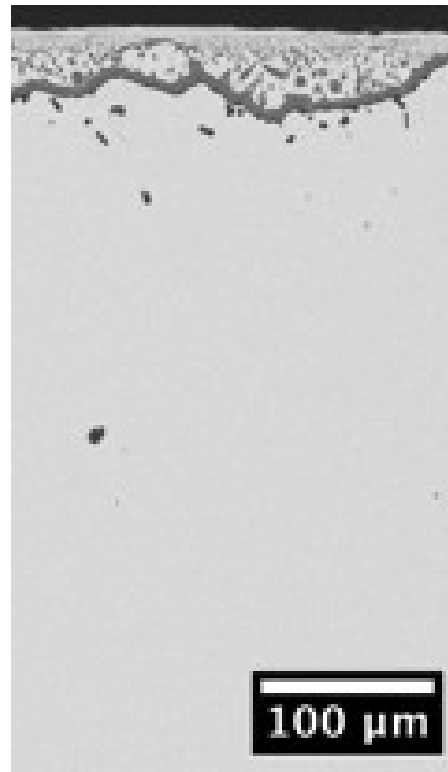
Figure: Calculated corrosion depth in Fe, 1wt-%Al, oxidised with $\text{N}_2 / 2.5\text{vol-\% H}_2 / 1.08\text{vol-\% H}_2\text{O}$ (left) and SEM-picture of the cross section after 300min (right).

Wagner's-Criterion for Internal Oxidation

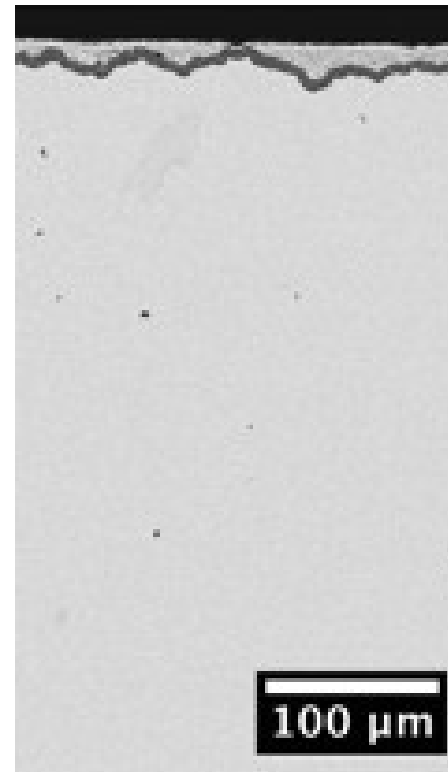
internal oxidation \longrightarrow external oxidation



Fe, 1wt-% Al



Fe-3Al



Fe-5Al



Fe-8Al

Figure: SEM-pictures of cross sectioned Fe-Al alloys after exposure to temperatures ≤ 1285 °C in N_2 / 2.5 vol-% H_2 / 1.08vol.% H_2O (DP +8°C) atmosphere.

Decarburization of Pearlite

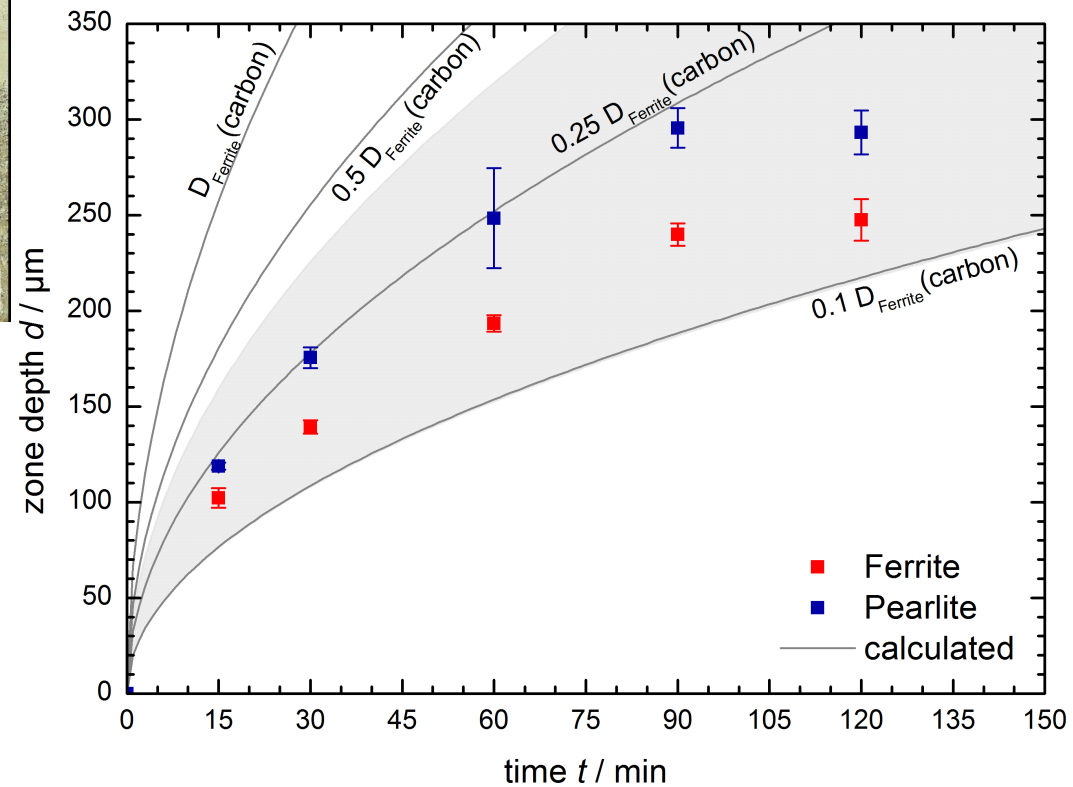
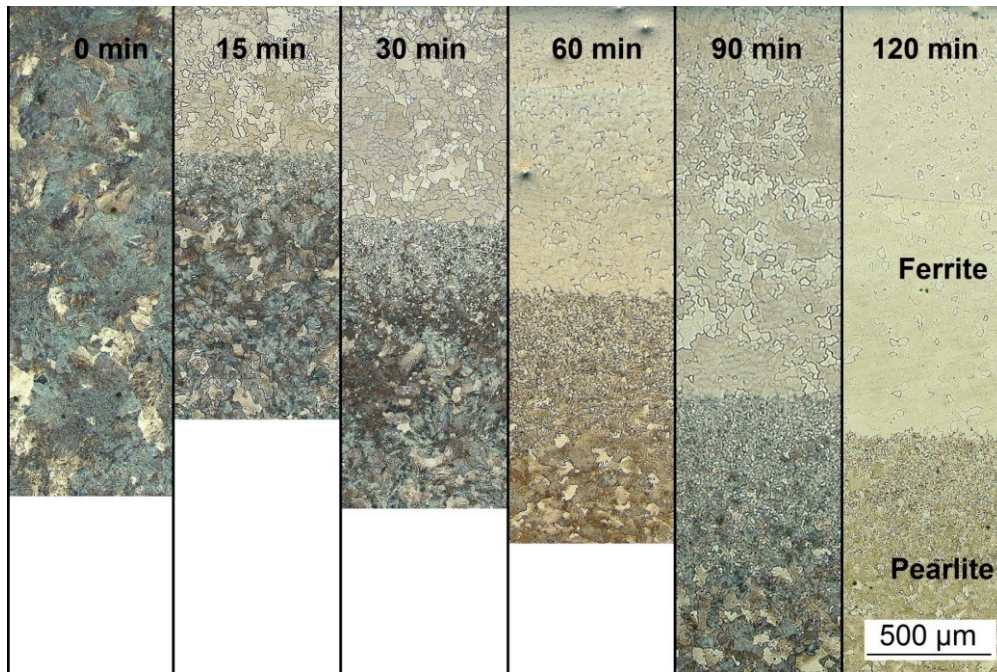


Figure: Angled cross section (7°) of Fe, 0.8 wt-% C after heat treatment at 800°C in Ar / 2.5 vol-% H_2 / 1.5 vol-% H_2O and etching with 1% Nital for 15s.

Properties of Oxygen



InnovMath

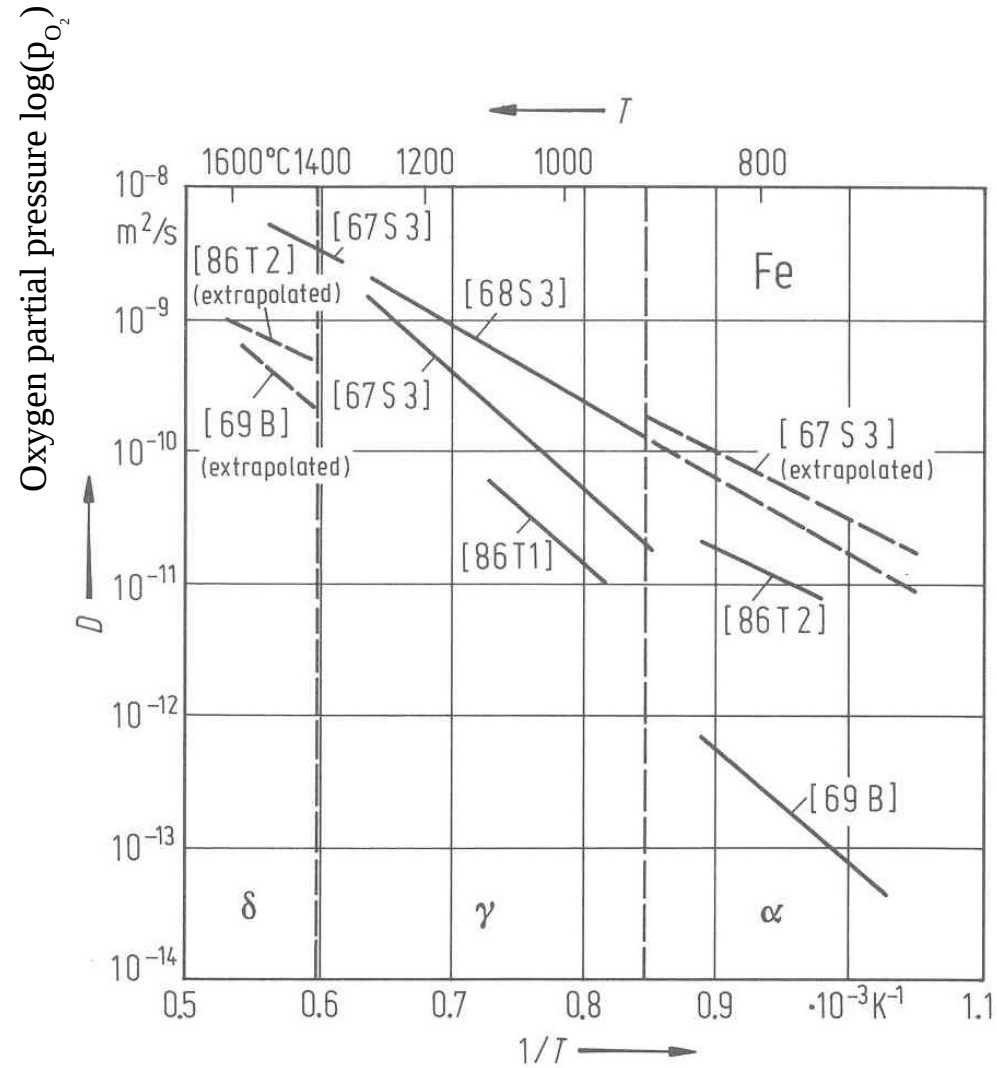
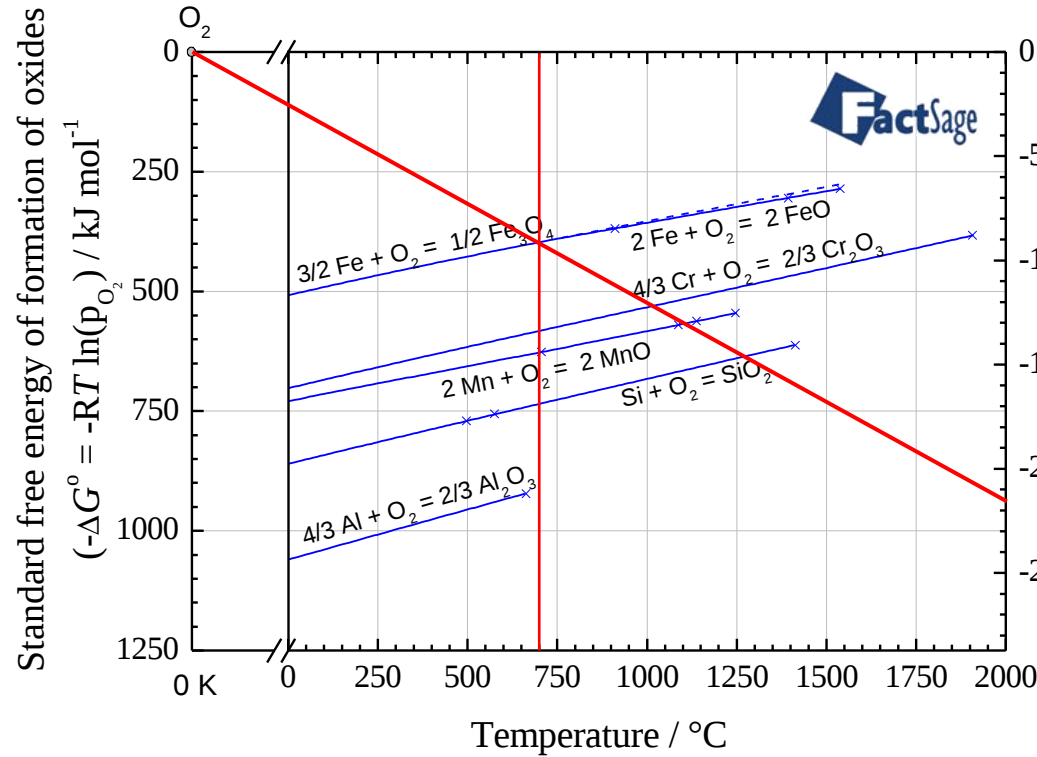


Fig. 23. Fe. Diffusion coefficient for O diffusion in α , γ and δ -phase Fe vs. (reciprocal) temperature.

Theoretical Background

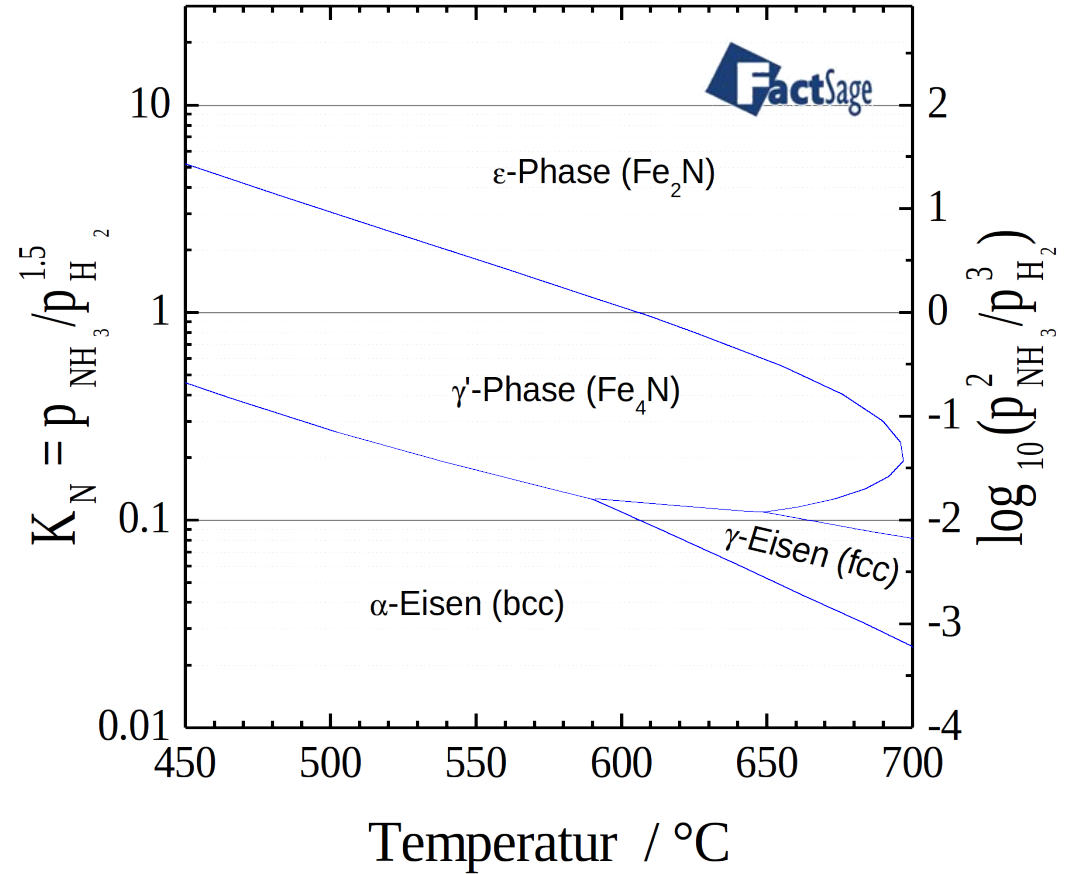
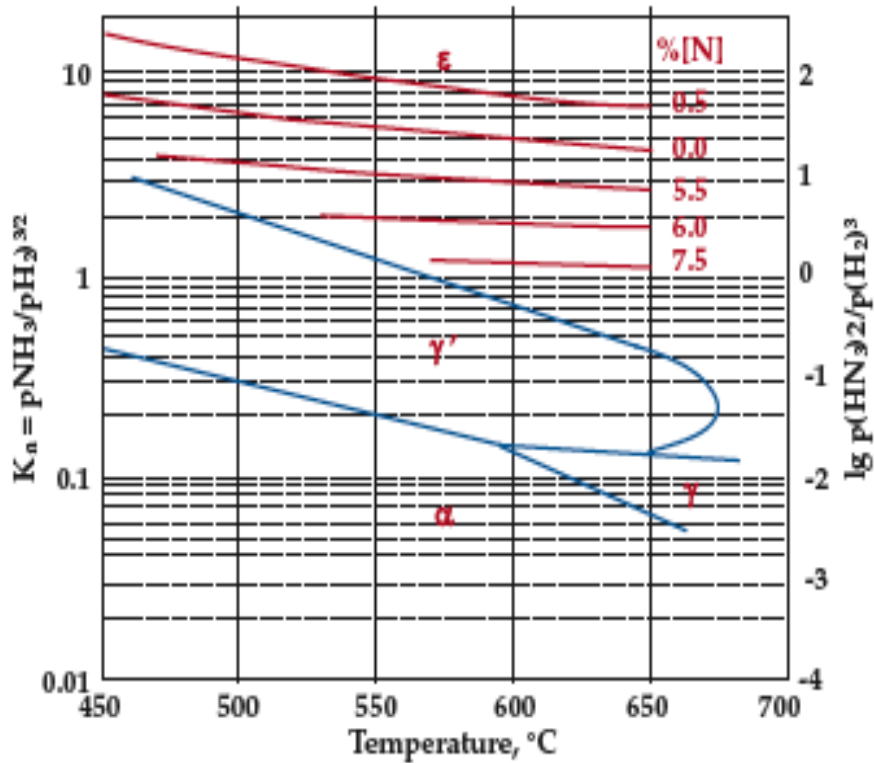


Figure: Lehrer diagram of iron nitride phases from literature (left) and calculated with FactSage (right).

Iron-Chromium-Carbon Alloy

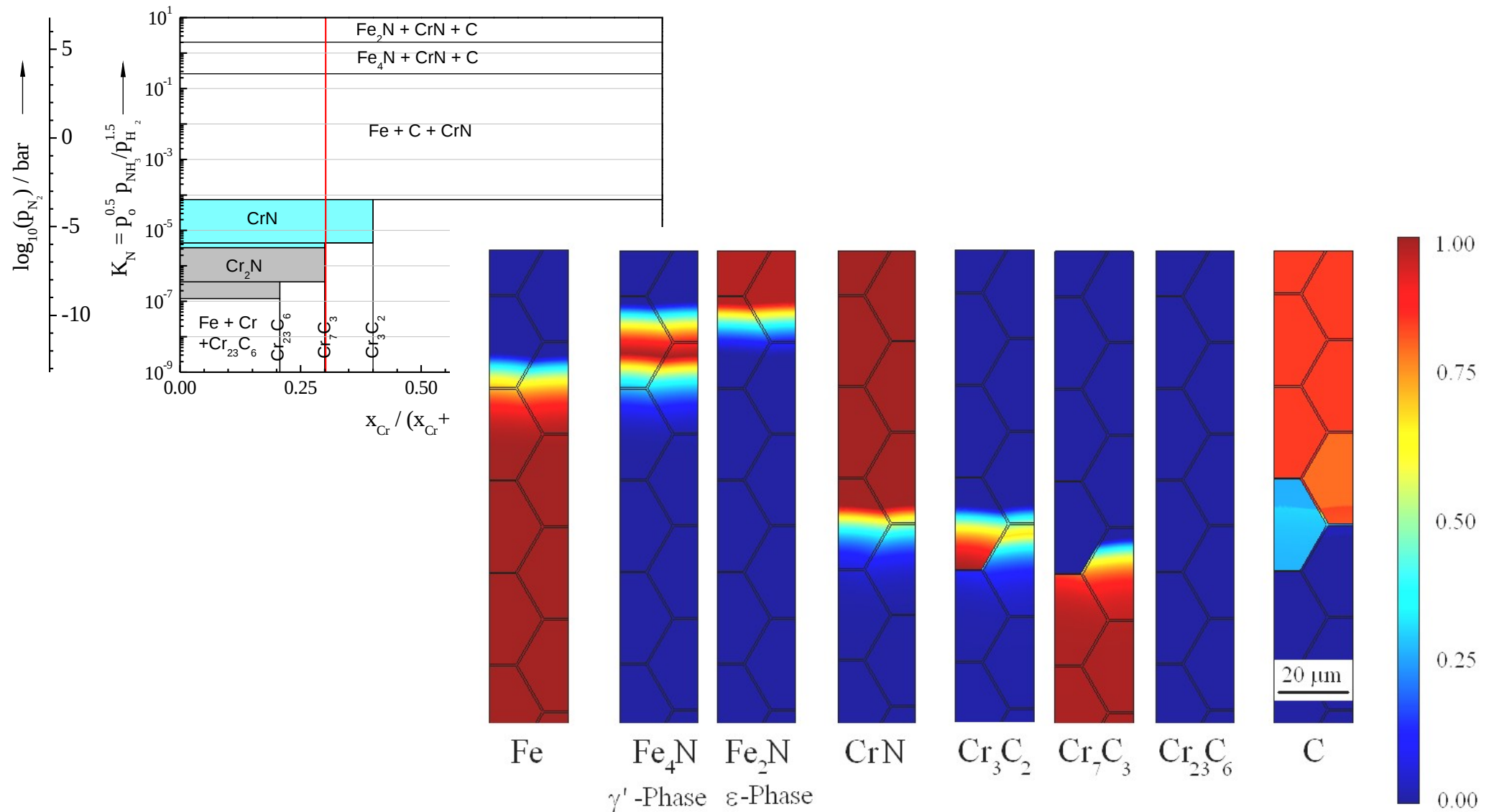


Figure: Spatial phase distribution in an Fe, 1 wt-% Cr, 0.1 wt-% C alloy after gas nitriding at $K_N = 2.4$ and 500 °C for 48 h and phase stability diagram.

Nitriding of AISI H11 (5%Cr, 0.35%C)

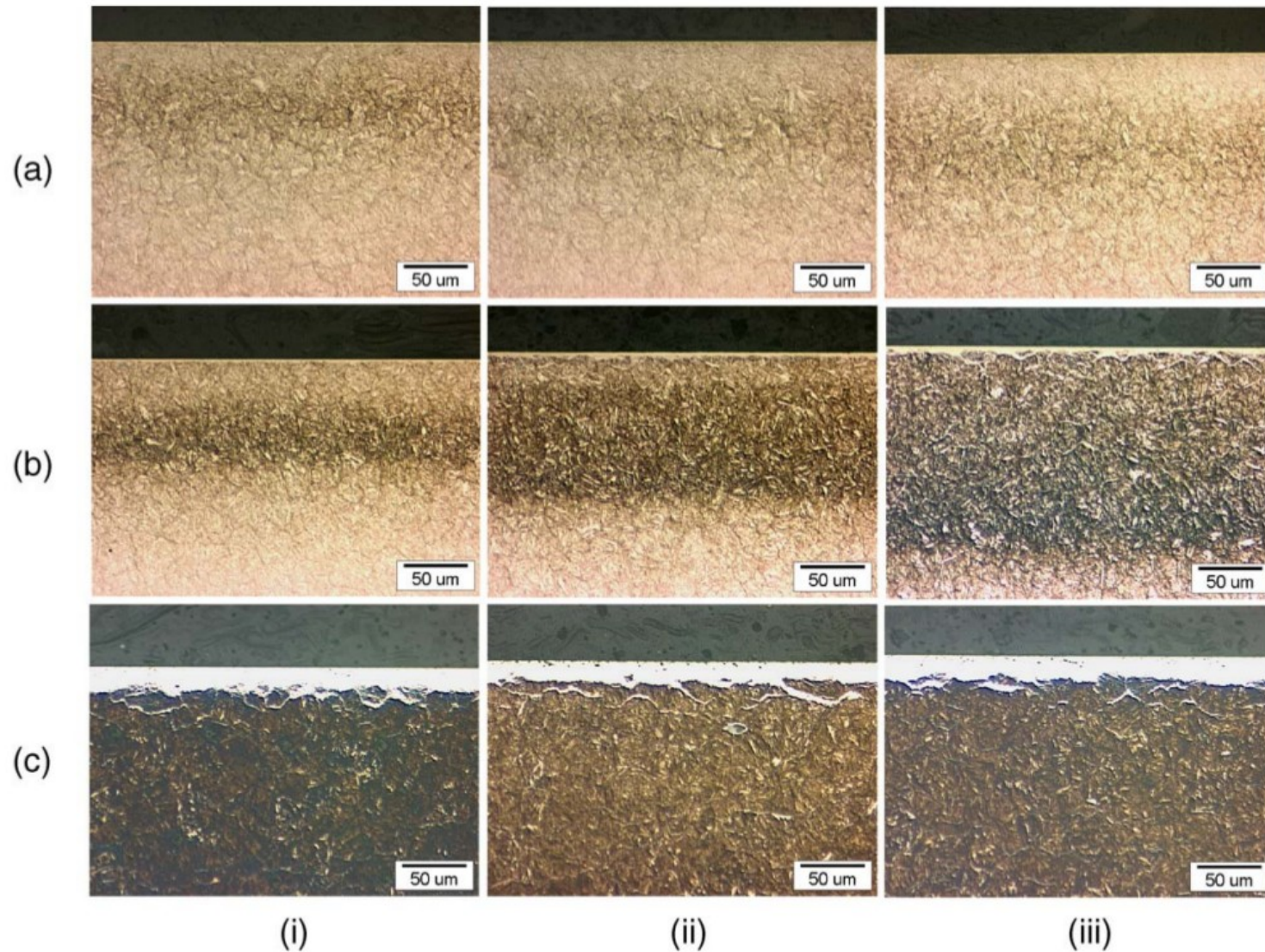


Fig. 4. Optical micrographs showing the cross-sectional microstructure for processing temperatures of (a) 480 °C, (b) 540 °C and (c) 580 °C. For each temperature, the processing times are (i) 8 h, (ii) 16 h and (iii) 32 h.

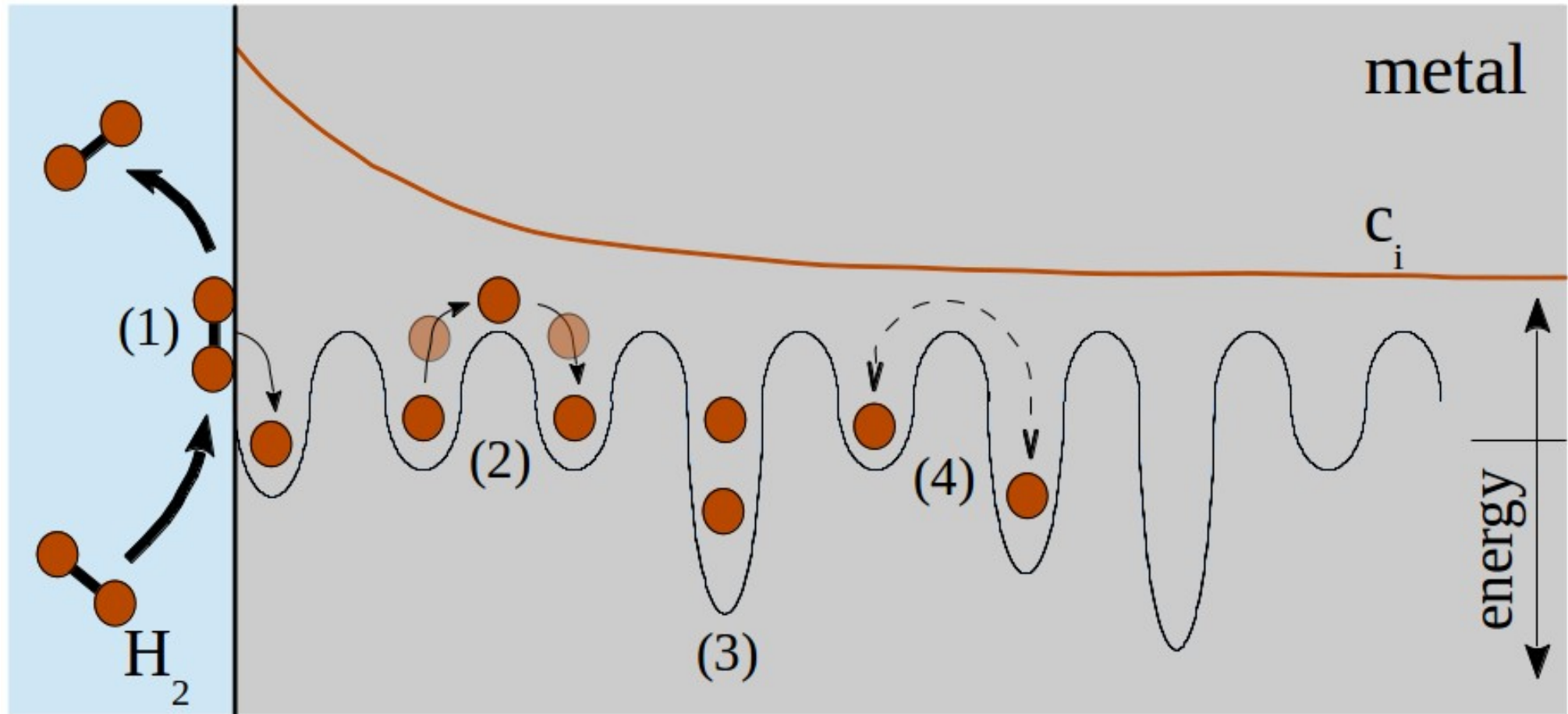
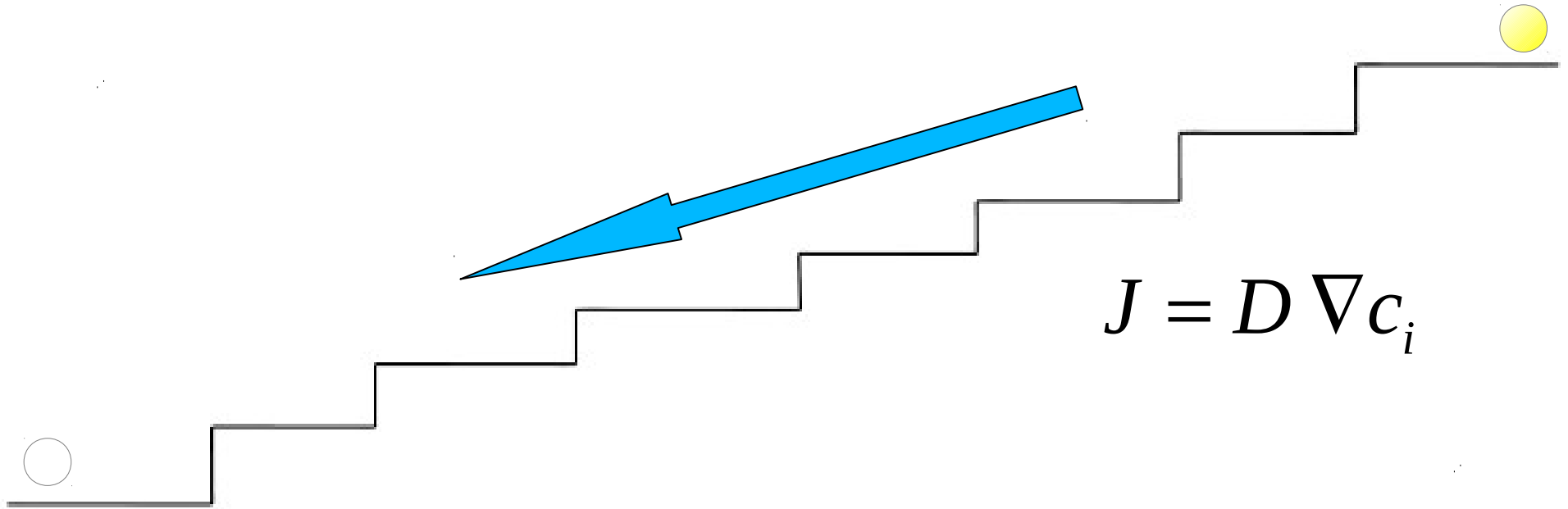


Figure: Illustration of hydrogen transport in crystalline materials. Effects of surface (1), diffusion (2), solubility (3) and trapping (4).

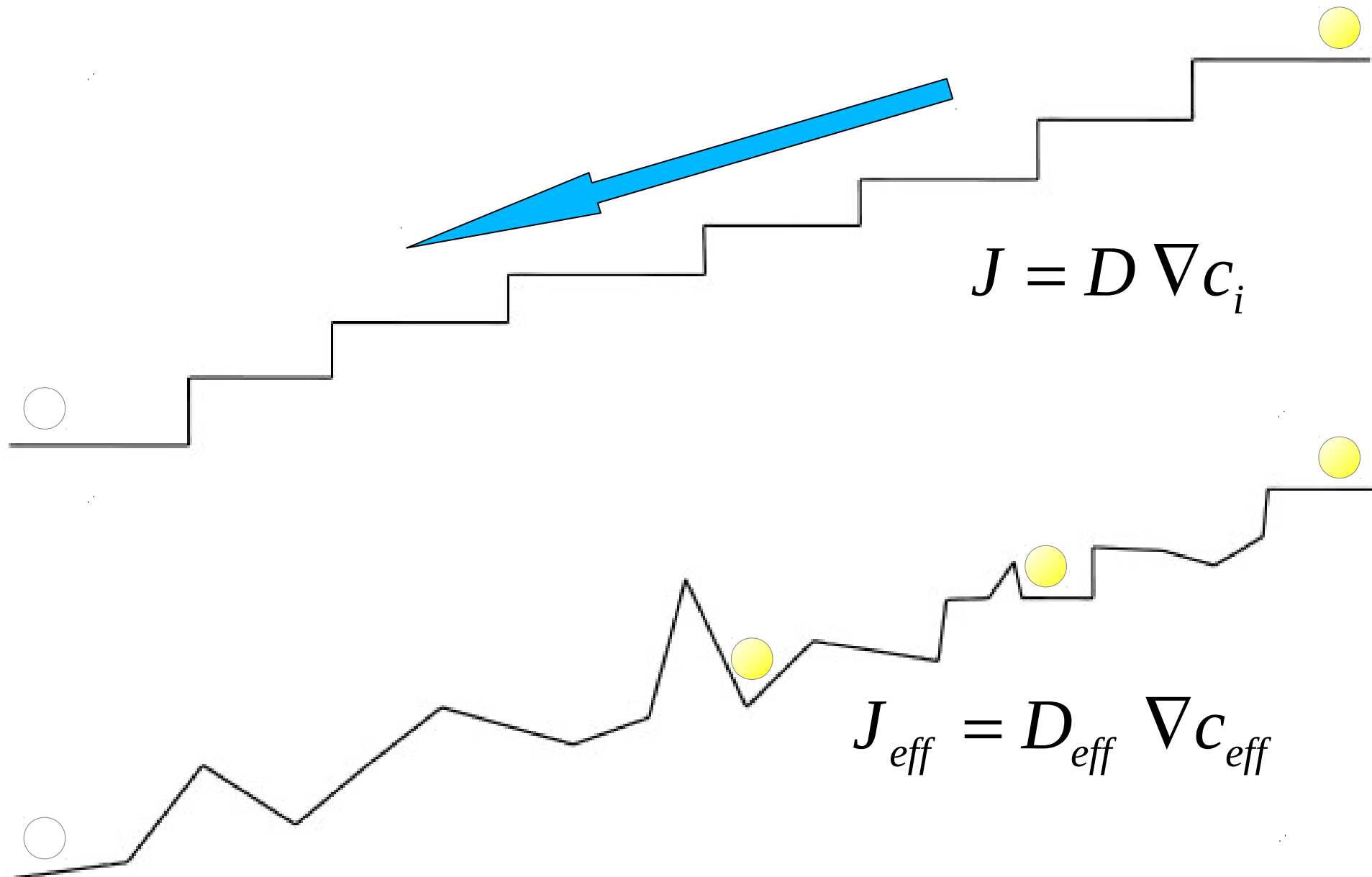
„Effective“ Diffusion



„Effective“ Diffusion



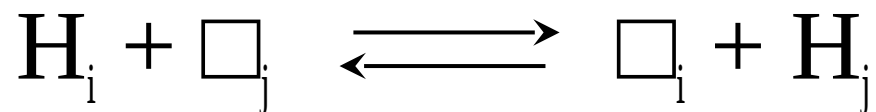
InnovMath



Theoretical Background



InnovMath



$$c_j = c_j^{\max} \frac{c_i K_j}{c_i^{\max} + c_i (K_j - 1)}$$

$$c_{ges} = c_i + \sum_j c_j$$

$$\frac{\partial c_{ges}}{\partial t} = \frac{\partial}{\partial x} \left\{ D \frac{\partial c_{ges}}{\partial x} \right\}$$

$$\left\{ 1 + \sum_j \frac{c_j^{\max} c_i^{\max} K_j}{[c_i^{\max} + c_i (K_j - 1)]^2} \right\} \frac{\partial c_i}{\partial t} = D \frac{\partial^2 c_i}{\partial x^2}$$

$$\frac{\partial c_i}{\partial t} = \frac{D}{\left\{ 1 + \sum_j \frac{c_j^{\max} c_i^{\max} K_j}{[c_i^{\max} + c_i (K_j - 1)]^2} \right\}} \frac{\partial^2 c_i}{\partial x^2}$$

$$K_{j(T)} = e^{\frac{Q_j}{RT}}$$

$$D_{(T)} = D^0 e^{\frac{Q}{RT}}$$

D_{eff}

Desorption at Constant Heating Rate

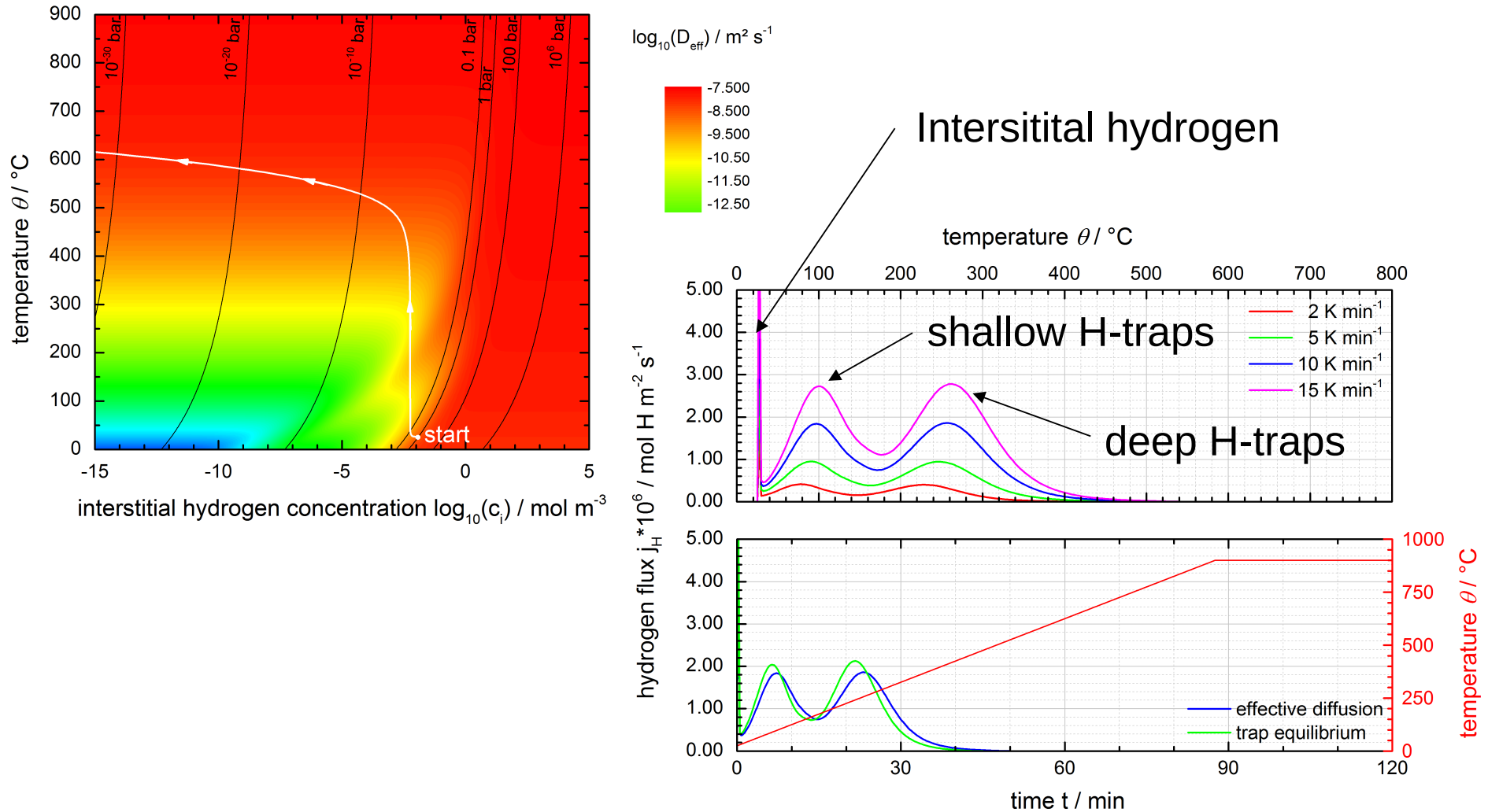


Figure: Effective diffusion coefficient in an idealized Fe-C alloy. Hydrogen traps are calculated as $H_1 = -58.6 \text{ kJ}$ (10 mol H m^{-3}), $H_2 = -84 \text{ kJ}$ (15 mol H m^{-3})

Isothermal Hydrogen Diffusion

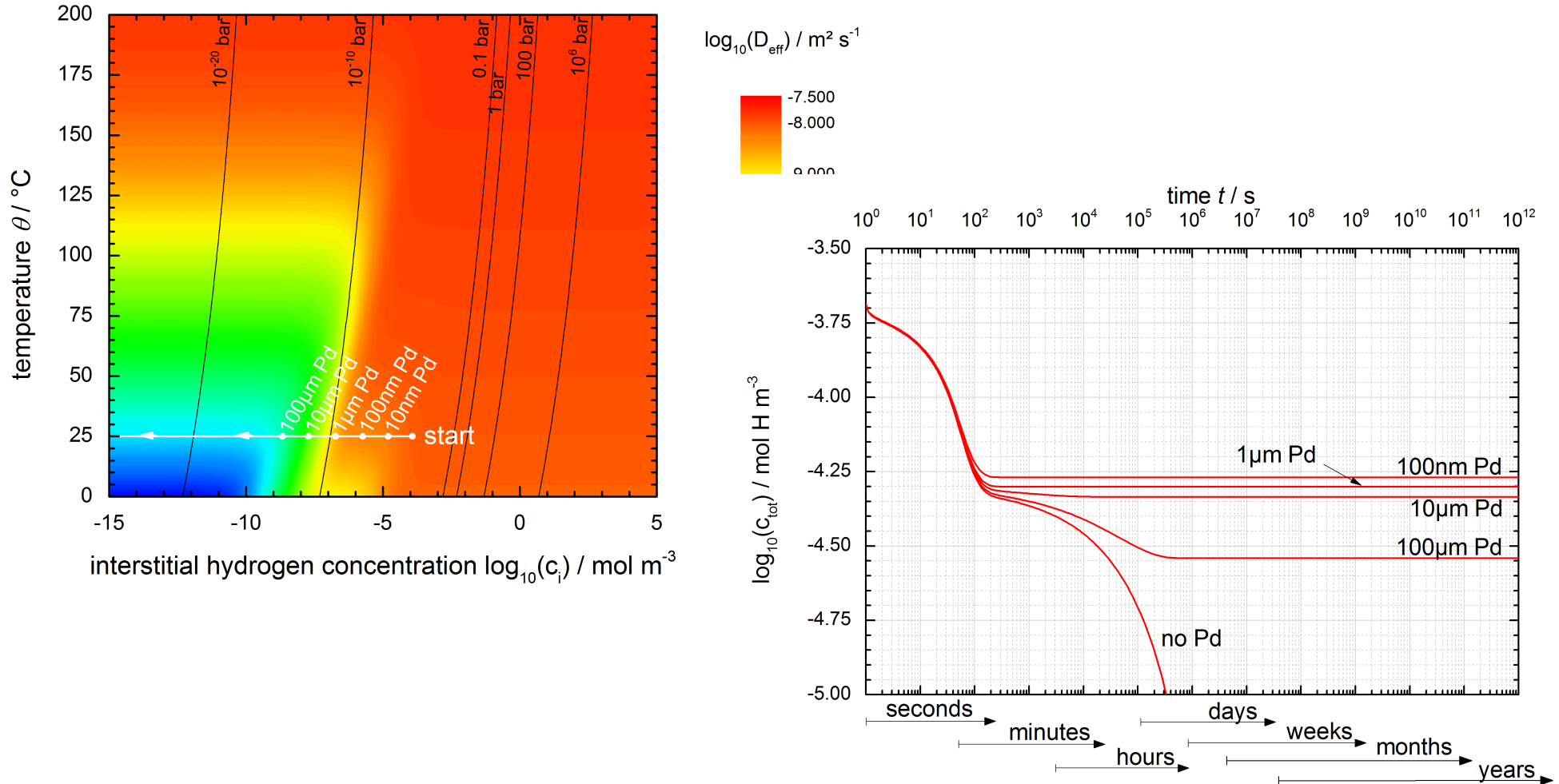


Figure: Effective diffusion coefficient in an idealized Fe-C alloy. Hydrogen traps are calculated as $H_1 = -58.6 \text{ kJ}$ ($0.05 \text{ mol H m}^{-3}$), $H_2 = -84 \text{ kJ}$ ($0.05 \text{ mol H m}^{-3}$)

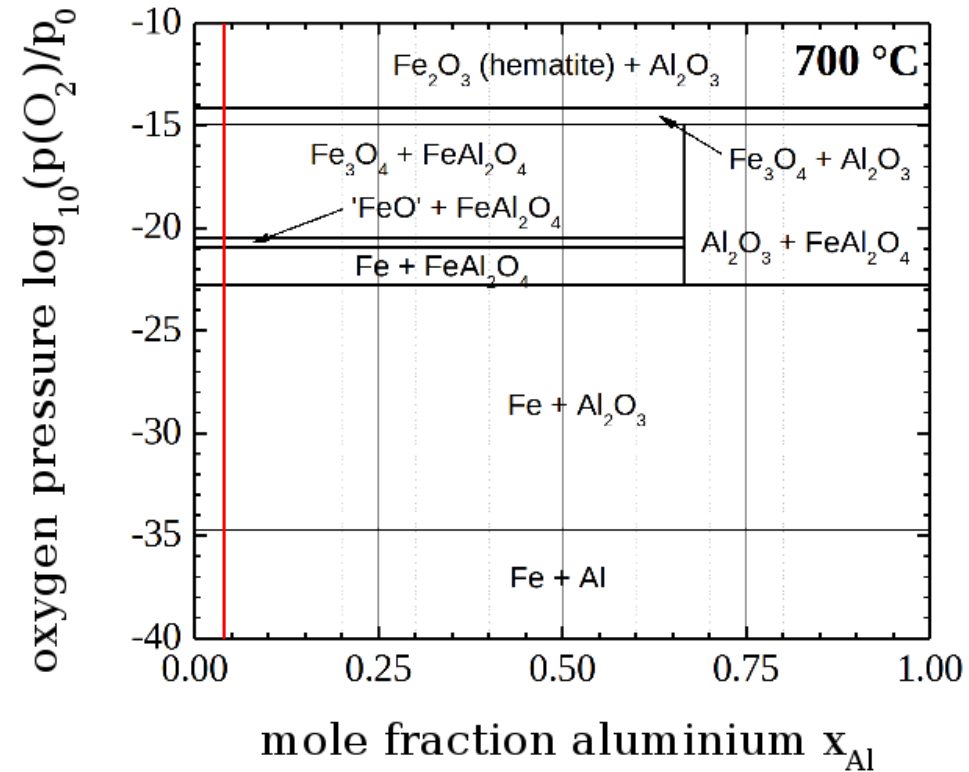
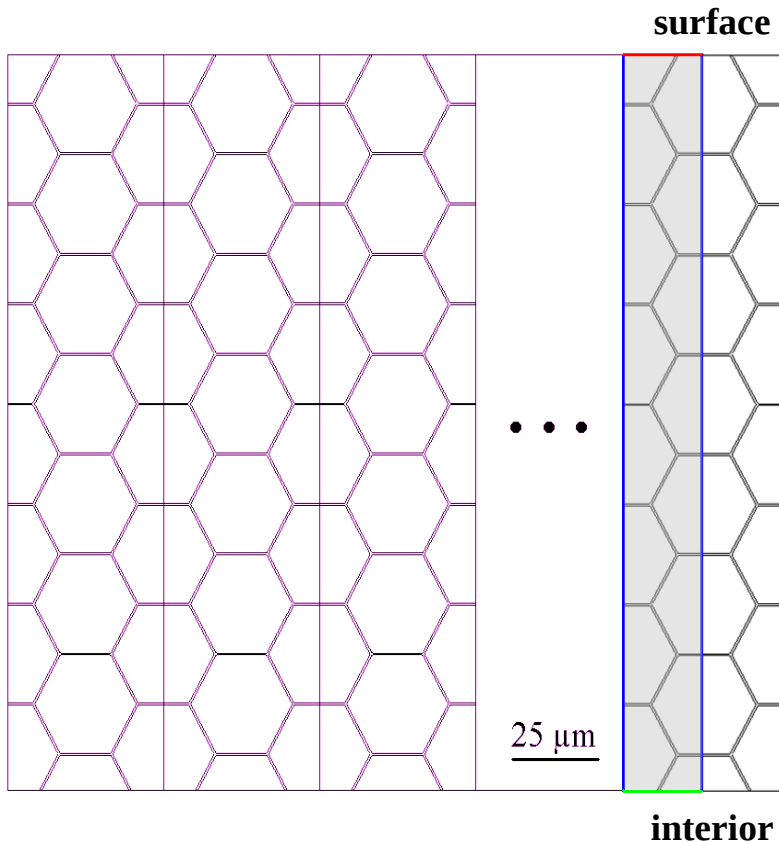
Program Structure



element transport

chemical reaction

$$\frac{dc_{i(x,t)}}{dt} = \text{div}(D_{i(x,T)} \cdot \nabla c_{i(x,t)})$$

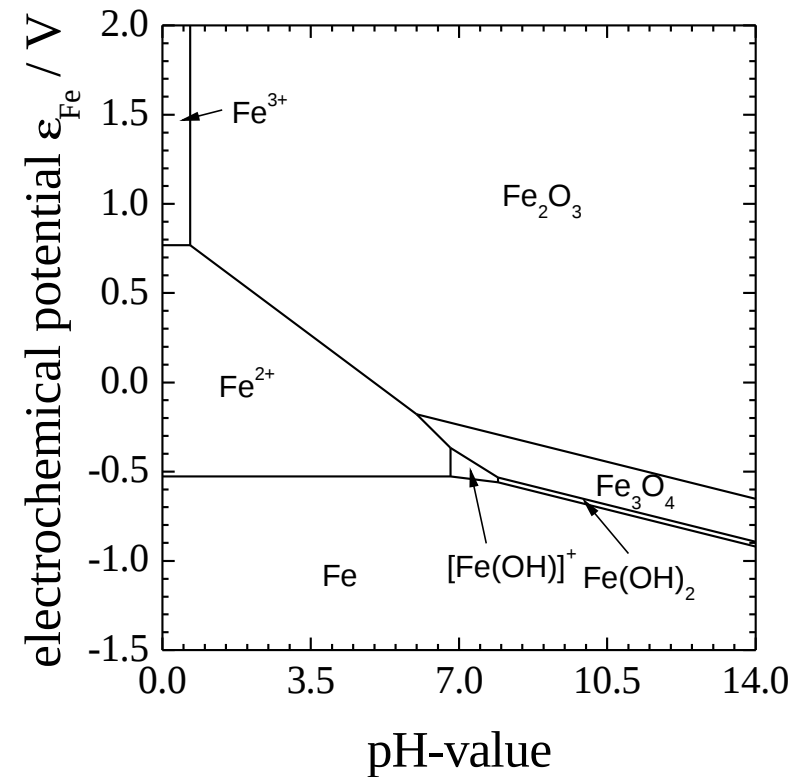
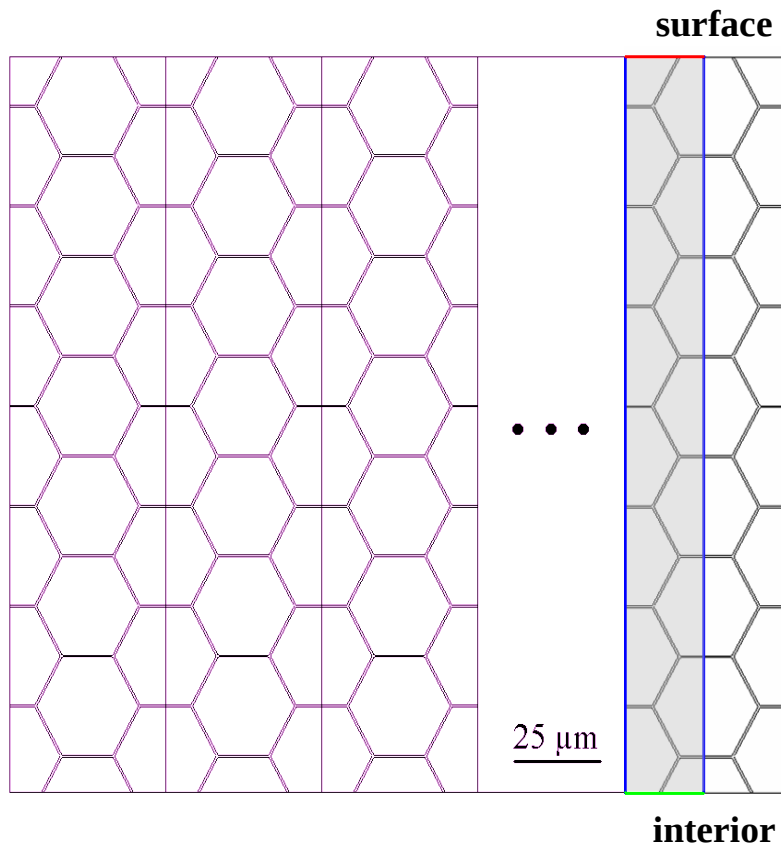


Program Structure

element transport

chemical reaction

$$\frac{dc_{i(x,t)}}{dt} = \text{div}(D_{i(x,T)} \cdot \nabla c_{i(x,t)})$$

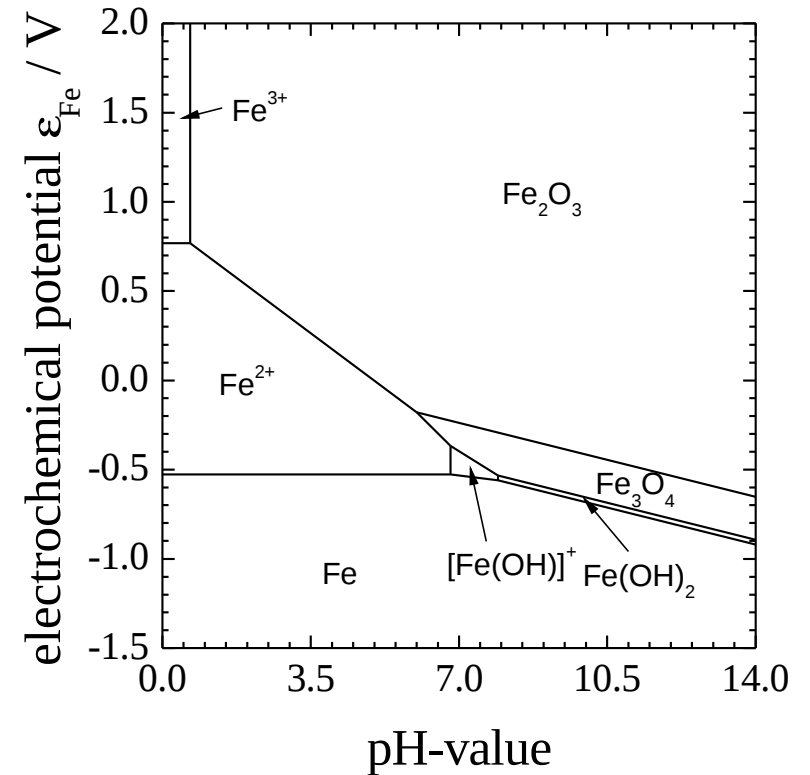
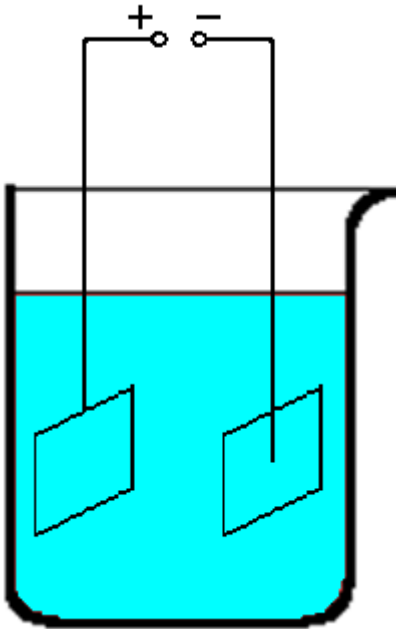


Program Structure

element transport

chemical reaction

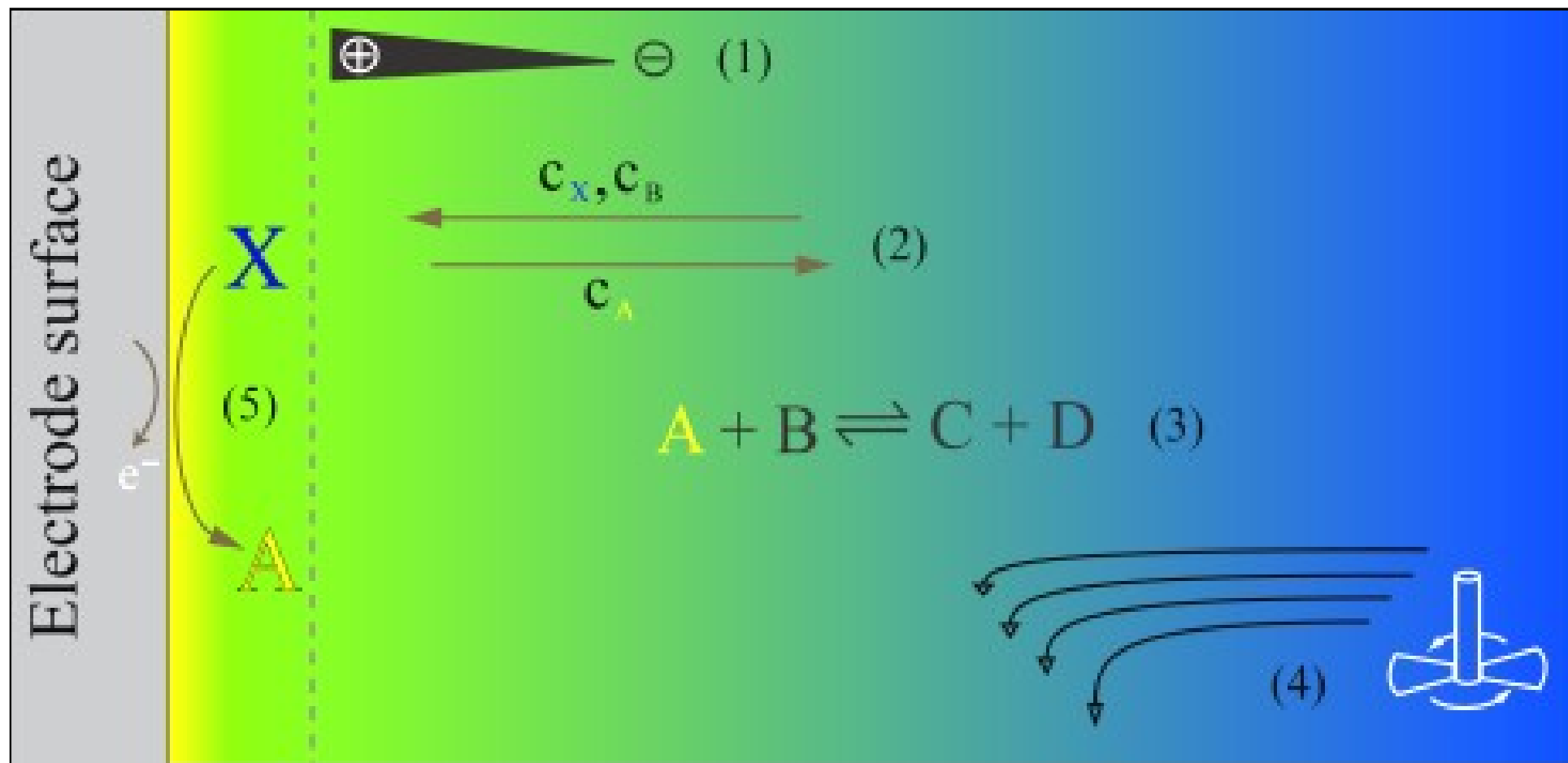
$$\frac{dc_{i(x,t)}}{dt} = \text{div}(D_{i(x,T)} \cdot \nabla c_{i(x,t)} + z_i \cdot \mu_{i(x,T)} \cdot c_{i(x,t)} \cdot \nabla \phi_{(x,t)})$$



Electrochemistry and Surface pH-values



InnovMath



Local Corrosion Effects



InnovMath

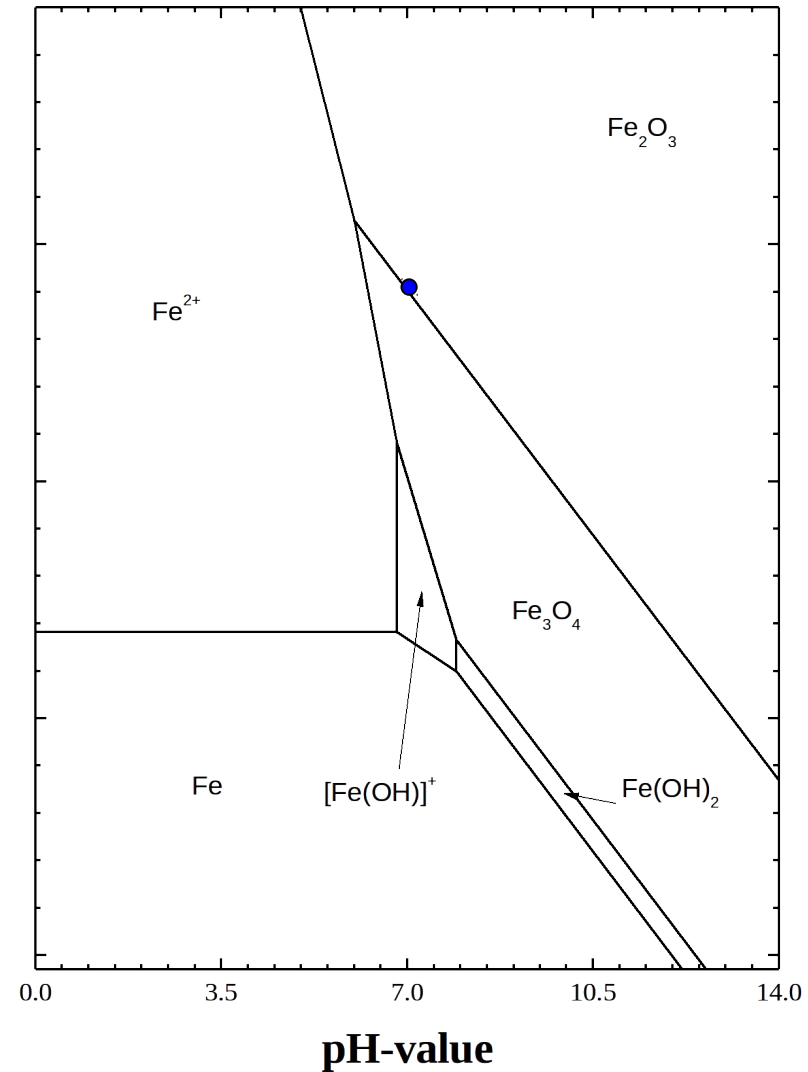
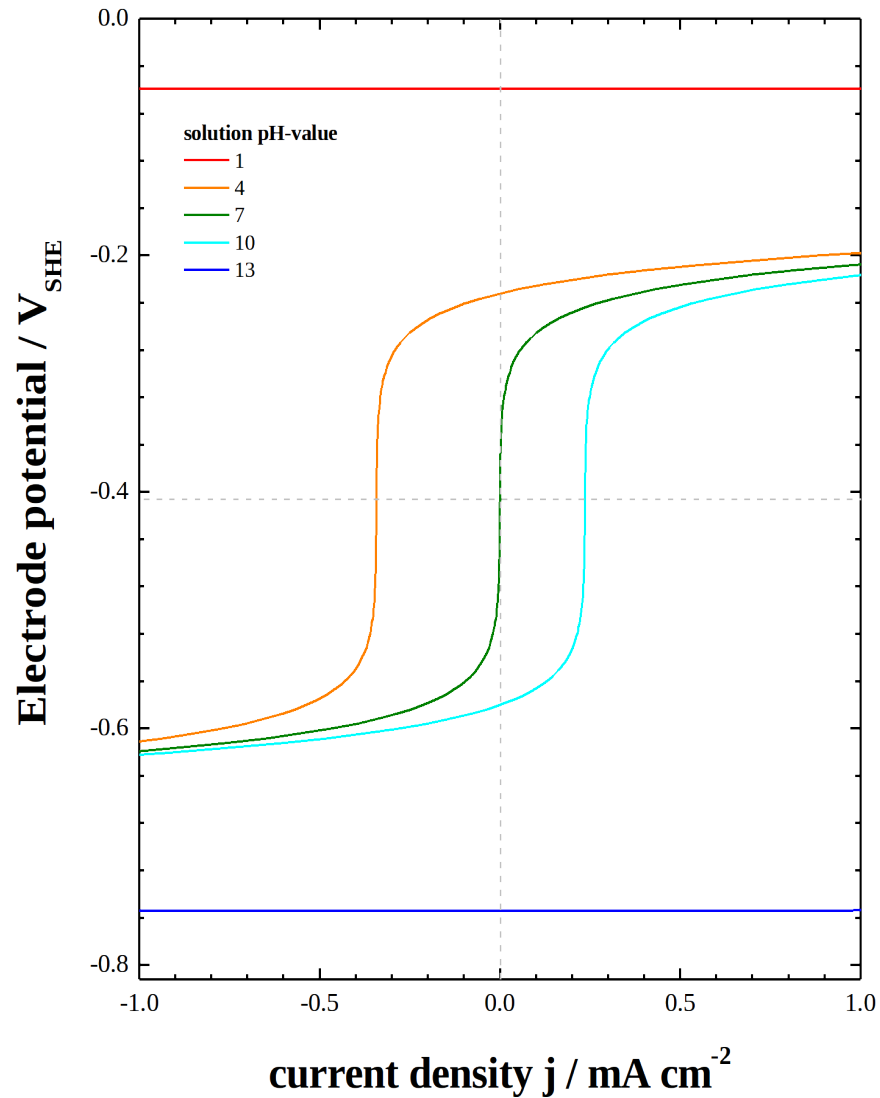


Figure: Cyclic voltammograms in unbuffered solutions of different pH-value.

Local Corrosion Effects

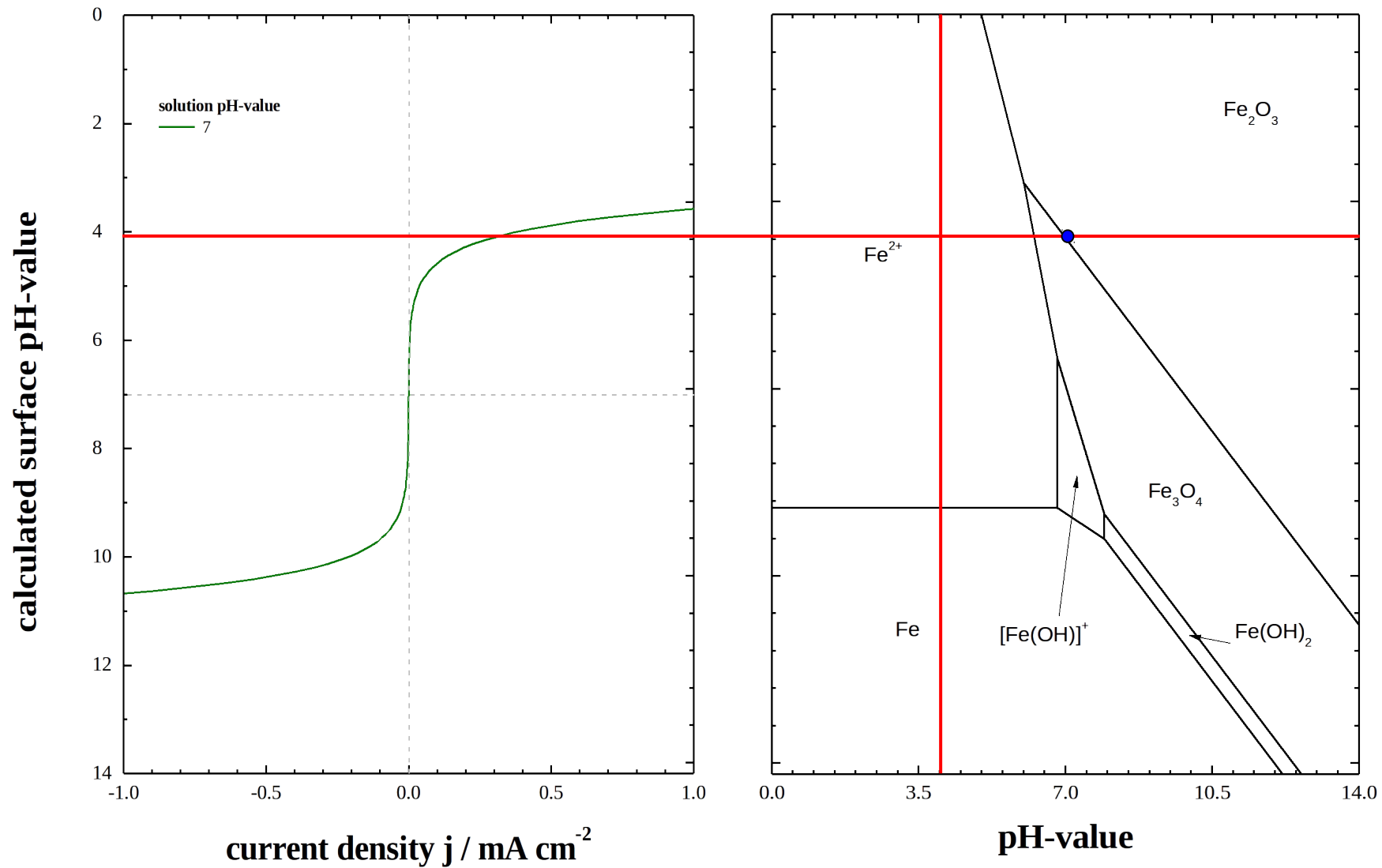
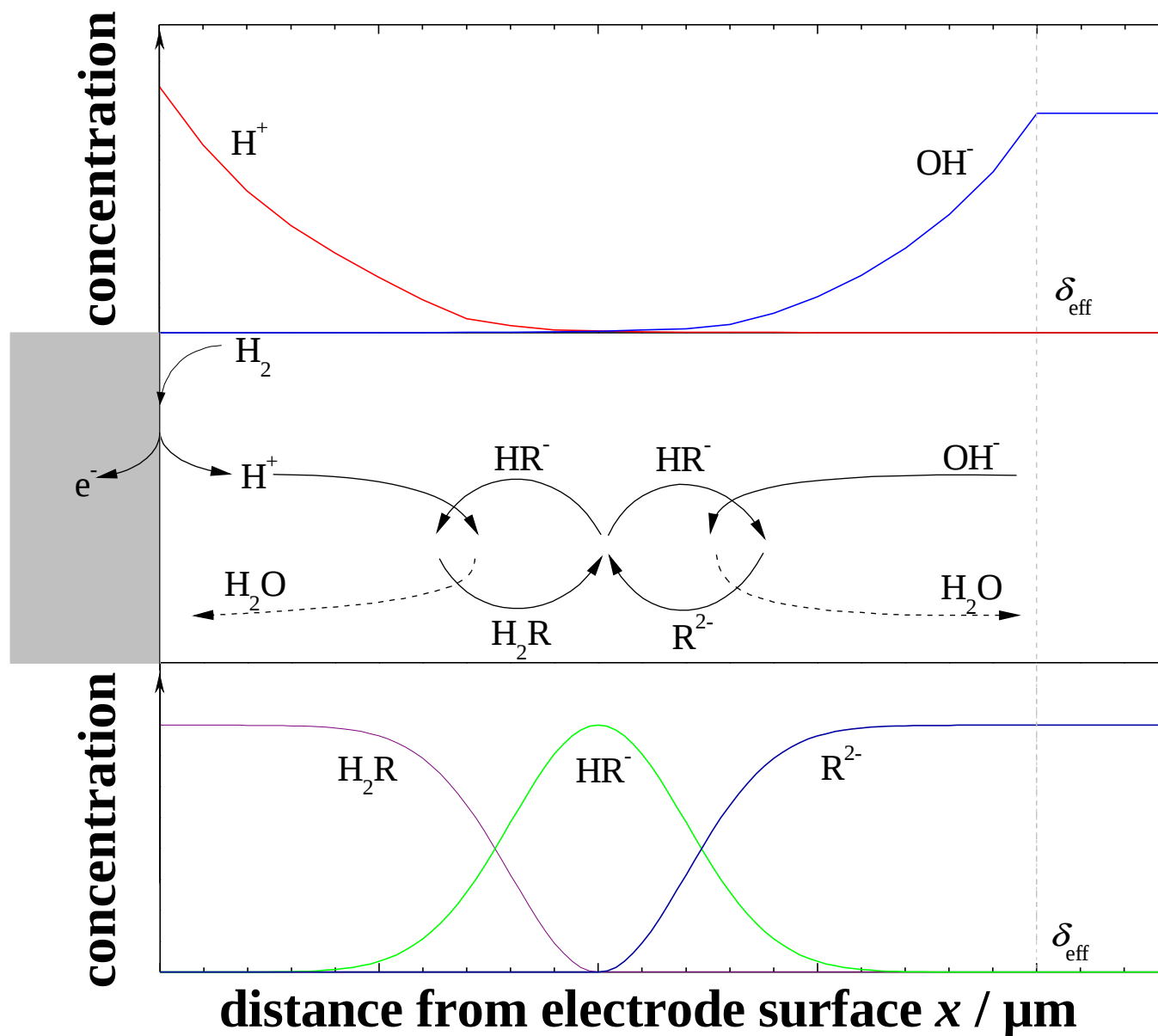


Figure: Cyclic voltammogram in unbuffered solution of pH 7.

General Transportation Scheme



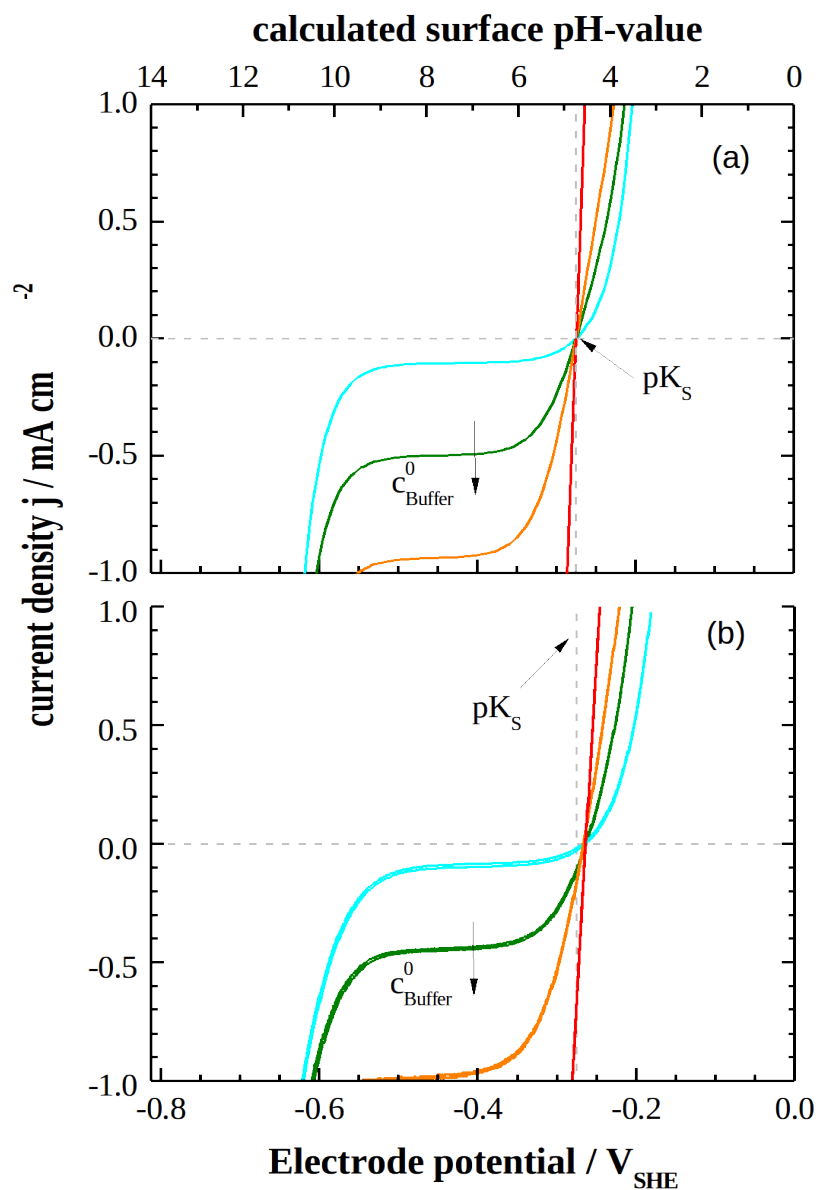
InnovMath



Acetate Buffered Solution



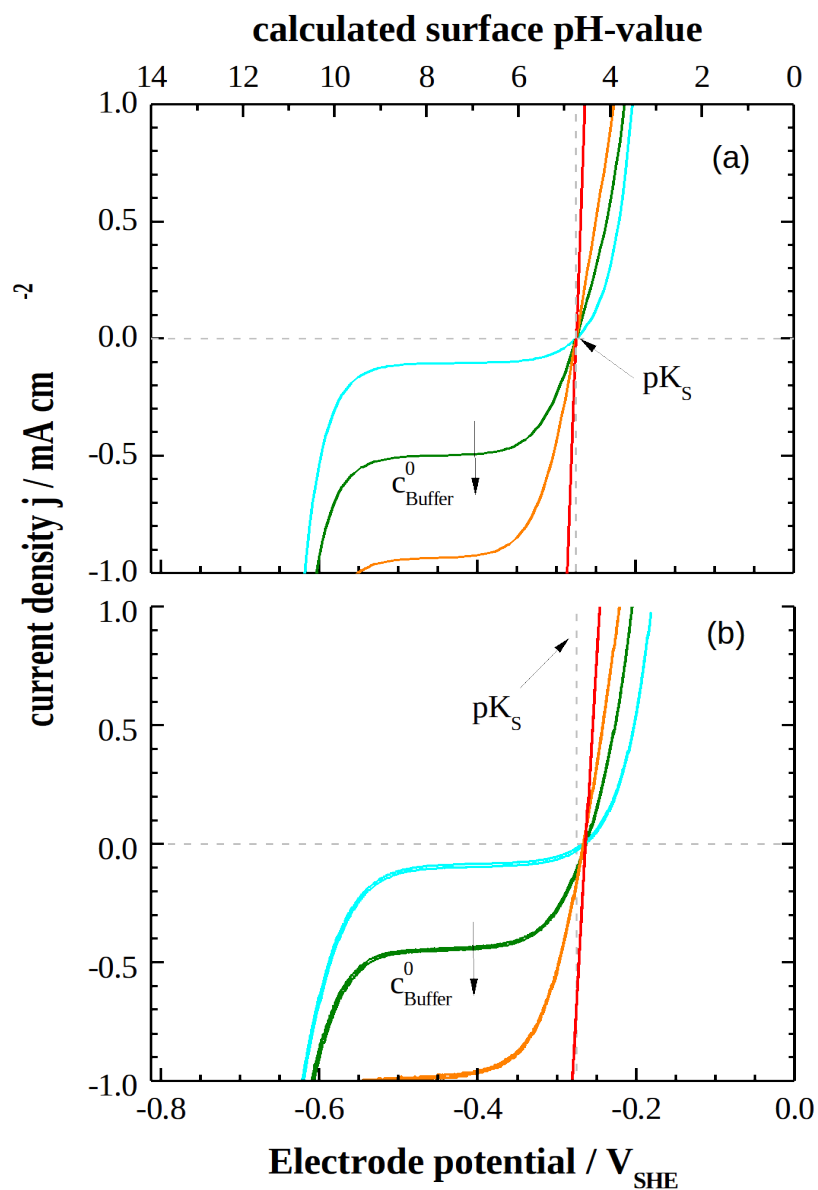
InnovMath



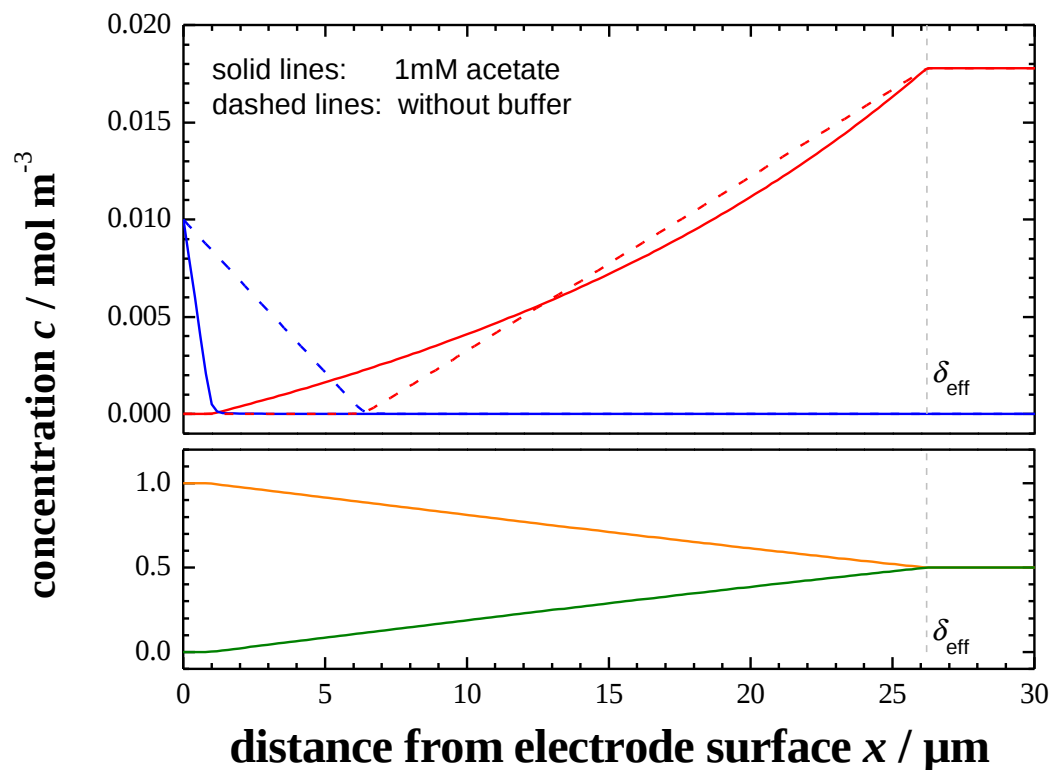
Acetate Buffered Solution



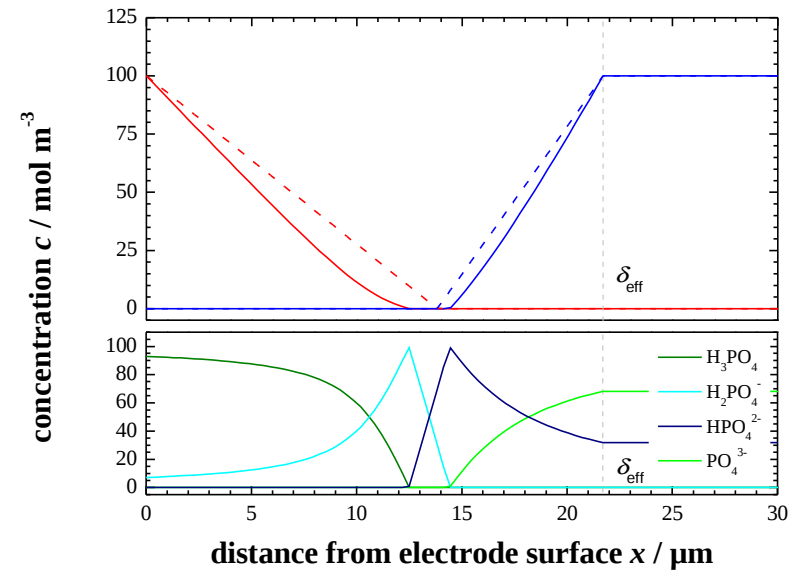
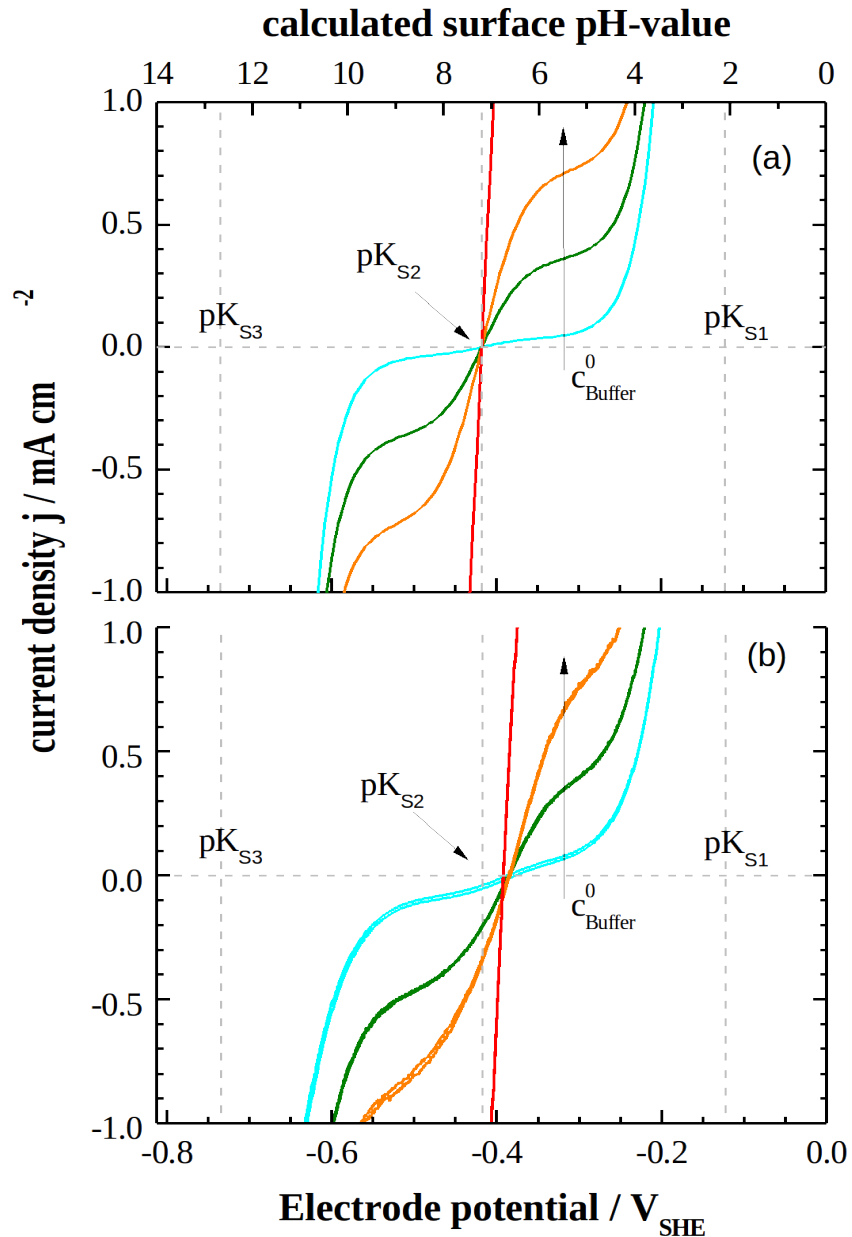
InnovMath



$$j = j_{\text{H}^+/\text{OH}^-} + \frac{c_{\text{Buffer}}^0 F}{\delta_{\text{eff}}} \left[\frac{D_{\text{HAc}} c_{\text{H}}^{\text{surface}} - D_{\text{Ac}^-} K_S}{K_S + c_{\text{H}}^{\text{surface}}} - \frac{D_{\text{HAc}} c_{\text{H}}^{\text{solution}} - D_{\text{Ac}^-} K_S}{K_S + c_{\text{H}}^{\text{solution}}} \right]$$



Phosphate Buffered Solution



$$j = j_{\text{H}^+/\text{OH}^-} + j_{\text{Buffer}}$$

$$j_{\text{Buffer}} = \frac{c_{\text{Buffer}}^0}{\delta_{\text{eff}}} F \left(\frac{3D_1 + D_2 \frac{K_{\text{S}1}}{c_{(x),\text{H}^+}} - D_3 \frac{K_{\text{S}1} K_{\text{S}2}}{c_{(x),\text{H}^+}^2} - 3D_4 \frac{K_{\text{S}1} K_{\text{S}2} K_{\text{S}3}}{c_{(x),\text{H}^+}^3}}{1 + \frac{K_{\text{S}1}}{c_{(x),\text{H}^+}} + \frac{K_{\text{S}1} K_{\text{S}2}}{c_{(x),\text{H}^+}^2} + \frac{K_{\text{S}1} K_{\text{S}2} K_{\text{S}3}}{c_{(x),\text{H}^+}^3}} \right)_{x=0}^{x=\delta_{\text{eff}}}$$

Summary



Modelling High Temperature Oxidation in Iron-Chromium Systems

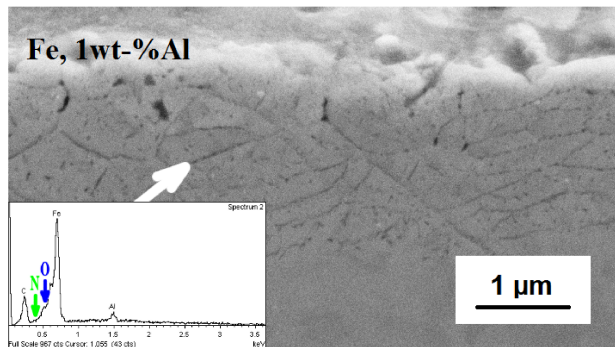
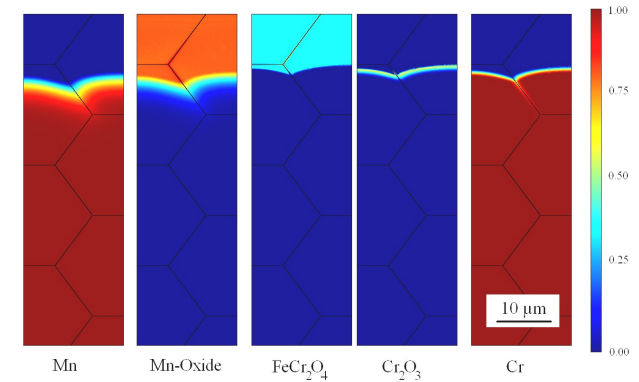
M. Auinger, R. Naraparaju, H.-J. Christ, M. Rohwerder

Oxid. Met. 76 (2011) 247-258

Thermodynamic Stability and Reaction Sequence for High Temperature Oxidation Processes in Steels

M. Auinger, S. Borodin, S. Swaminathan and M. Rohwerder

Mater. Sci. Forum. 696 (2011) 76-81



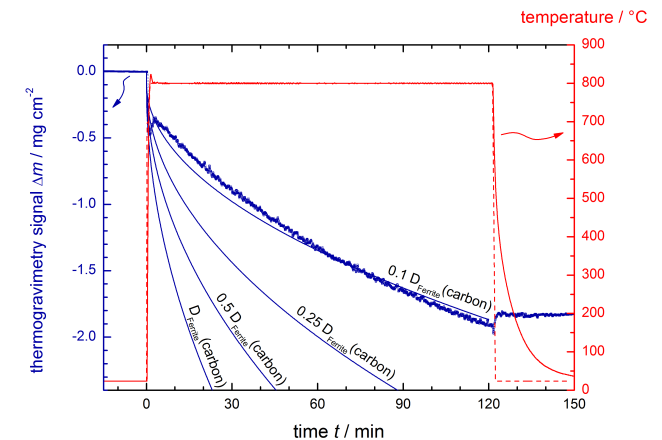
Modelling and Experiment of Selective Oxidation and Nitridation of Binary Model Alloys at 700°C - The Systems Fe, 1 wt-%{Al, Cr, Mn, Si}

M. Auinger, E.-M. Müller-Lorenz, M. Rohwerder, *Corros. Sci.* accepted

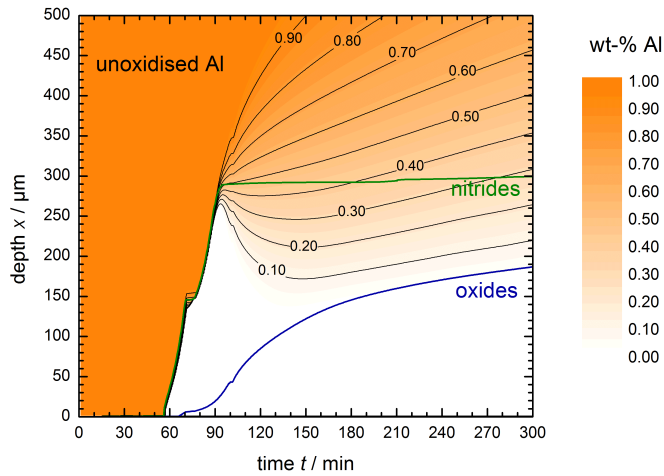
Thermogravimetry and in-situ mass spectrometry at high temperatures compared to theoretical modelling – The weight loss during selective decarburisation at 800 °C

M. Auinger, A. Vogel, V. G. Praig, H. Danninger, M. Rohwerder

Corros. Sci. 78 (2014) 188-192

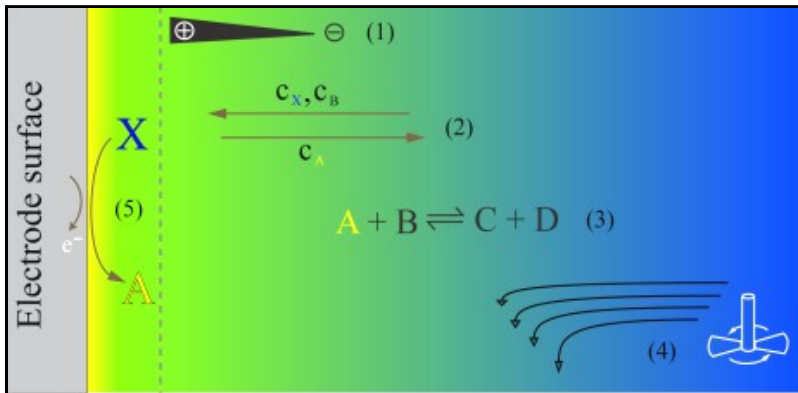
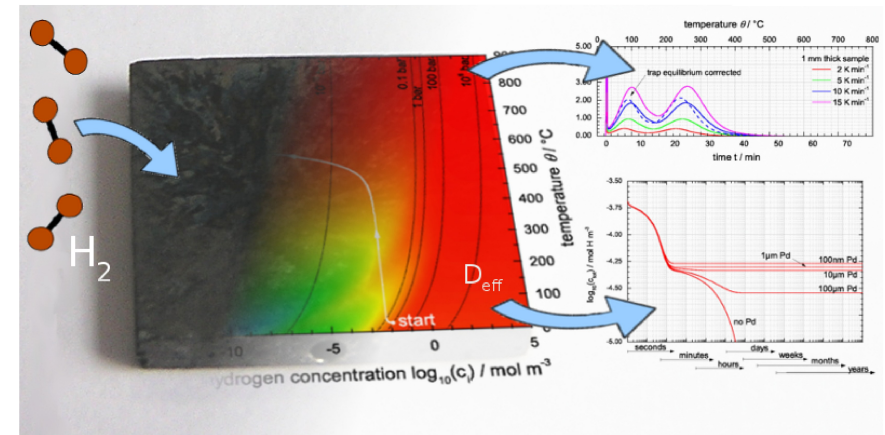


Summary



Selective Precipitation of Oxides and Nitrides in Fe-Al Alloys
 J.H. Bott, H. Yin, S. Sridhar, M. Auinger
submitted

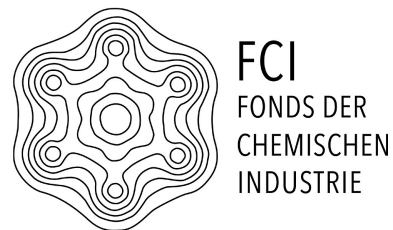
Hydrogen Transport in non-ideal Crystalline Materials
 M. Auinger, *Chem. Phys. Chem.* (2014)
 DOI:10.1002/cphc.201402001



Near-surface ion distribution and buffer effects during electrochemical reaction

M. Auinger, I. Katsounaros, J.C. Meier, S.O. Klemm,
 P.U. Biedermann, A.A. Topalov, M. Rohwerder, K.J.J. Mayrhofer
Phys. Chem. Chem. Phys. 13 (2011) 16384-16394

Financial Support



Acknowledgements



Max-Planck-Institut
für Eisenforschung GmbH

K. Mayrhofer, M. Rohwerder, J. Meier, M. Nellesen,
D. Kurz, I. Katsounaros, A. Mingers



TECHNISCHE
UNIVERSITÄT
WIEN
Vienna University of Technology

Prof. Danninger
V. Praig



Prof. Seetharaman (now Uni Warwick)
J. Bott, J. Zhu

voestalpine

EINEN SCHRITT VORAUS.

Dr. Paesold
B. Linder, K. Rendl, A. Muhr