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# Numerical Evaluation of Pin-Bearing Strength for the Design of Bolted Connections of Pultruded FRP material

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**Abstract:** Presented in this paper are finite element (Abaqus) predictions for the strength of a pultruded fibre reinforced polymer material subjected to pin-bearing loading with hole clearance. One of the distinct modes of failure in steel bolted connections is bearing. It is caused by the compression action from the shaft pressing into the laminate, and when there is no lateral restraint the mechanism observed at maximum load shows ‘brooming’ for delamination failure.

Each lamina in the glass fibre polyester matrix material is modelled as a homogeneous, anisotropic continuum and a relative very thin resin layer is assumed to contain any delamination cracking between stacked layers. A cohesive zone model is implemented to predict the size and location of the initial delamination, as well as the load-carrying capacity in a pin-bearing specimen. Finite element simulations (as virtual tests) are performed at the mesoscale level to validate the modelling methodology against experimental strength test results with delamination failure, and to show how pin-bearing strength varies with parameter changes. For an example of the knowledge to be gained for the design of bolted connections, the parameteric study where the mat reinforcement is either continuous strand or triaxial (+45°/90°/-45°/chopped strand) shows the latter does not provide an increase in pin-bearing strength.

**Author keywords:** Damage mechanics; Bearing failure; Finite element modelling; Pultruded material.

## Introduction

Shapes made by the pultrusion composite processing method consist of thin-walled laminated panels of glass Fibre Reinforced Polymer (FRP) matrix connected to form open or closed cross-sections. Because I, wide flange, channel, box and leg-angle shapes mimic steel sections it is natural that frame construction follows what is seen in conventional steelwork (Creative Pultrusions 2016; Strongwell 2016; Turvey 2000). Being lightweight and resistant to corrosion, and having expected low life cycle costing, pultruded structures are increasingly used where these attributes meet the requirements of the construction project, such as for pedestrian bridges (Anonymous 2016).

Introduced in Mottram and Zafari (2011) is the rationale for steel bolting to be a main method of connection (Creative Pultrusions 2016; Strongwell 2016), and the requirement that, for strength design, a reliable test method is needed to determine bearing strengths. The *bearing strength* is an important material parameter for static strength design (Bank 2006; Mottram and Turvey 2003), and can be characterized as the resistance of the material to a fastener loaded hole. Strength depends on a number of parameters including: FRP thickness ( $t$ ); FRP mechanical properties; FRP material orientation to the bearing force; bolt material; bolt diameter ( $d$ ); clearance hole size. In bolted lap-joints the end distance ( $e_1$ ), which is the distance from the centre of the hole to the free edge perpendicular to the loading direction has to exceed a limiting value for the bearing failure mode to govern.

An advantage for bolted connection design failing in the bearing mode is that it might provide the connection/joint with a degree of damage tolerance and structural integrity. Bearing failure can be enhanced by: (i) correctly sizing the connection, usually end distance to fastener diameter ratio  $e_1/d \geq 3$  providing the width to fastener ratio is 4 or higher, see Bank (2006), and Girão Coelho and Mottram (2015a); (ii) the choice of the fibre reinforcements in the lamination stacking sequence.

Mottram and Zafari (2011) justify why the measurement of bearing strength for design calculations must be for the lower bound pin-bearing value. The pin-bearing condition assumes the

61 steel bolt shaft has no thread in bearing and that there is no lateral restraint, which will not initially  
62 exist in practice because the washer, nut, bolt combination should be tightened. The paper explains  
63 the reasons for having a strength test method for pin-bearing strength with a plate-shaped coupon  
64 that is no larger in size than, say  $80 \times 80 \times t$  mm. Using an in-house test approach, detailed in Mottram  
65 and Zafari (2011) or Matharu (2014), numerous test results have been reported from programmes of  
66 characterization work (see also Zafari and Mottram 2012) to understand the variations in pin-  
67 bearing strength and characteristic values for pultruded materials. Two different materials were  
68 studied, and in the test matrix parameters were: four bolt diameters, e.g. 9.53 mm (3/8 in.), 12.7 mm  
69 (1/2 in.), 15.8 (5/8 in.) and 25.4 mm (1 in.) (Mottram and Zafari 2011) or M10, M12, M16 and M20  
70 (Matharu 2014); different material orientations (e.g.  $0^\circ$ ,  $90^\circ$  and  $45^\circ$ ); non-aged and hot-wet aged  
71 materials. One aim of the characterization work was to obtain data on how pin-bearing strengths  
72 might change in the field due to the long-term effect of exposure to a site's environment.

73 Both materials were from the American pultruder Creative Pultrusions (2016) having glass  
74 fibre reinforcement in the form of alternative layers of unidirectional rovings and a mat, and a fire-  
75 retardant matrix of an isophthalic polyester polymer. For the Mottram Zafari (2011) and Zafari and  
76 Mottram (2012) tests the pultruded material was from the standard 1525 series having mat  
77 reinforcement of a Continuous Strand Mat (CSM) having a random arrangement of continuous  
78 fibres. Matharu's (2014) work was with material from the Pultex<sup>®</sup> SuperStructural 1525 series, and  
79 the main difference is that the mat reinforcement is triaxial, having unidirectional fibres at  $90^\circ$ , and  
80  $\pm 45^\circ$  and a 'thick' backing of a chopped strand fibres. Another difference in the test matrix with  
81 Matharu is that testing was with and without bolt thread in bearing; only the latter connection  
82 condition can be linked to the Finite Element (FE) work presented in this paper.

83 The FE modelling methodology applied by the authors uses options solely available from the  
84 general purpose software Abaqus with its implicit solver (2016). Girão Coelho *et al.* (2016)  
85 introduce, and show how the approach is modelling the various failure modes, and, in particular, for  
86 the parametric studies given later the critical mode of progressive delamination using cohesive zone

87 models. A similar approach using Abaqus/Explicit has been reported by Du *et al.* (2016) with their  
88 progressive damage analysis implemented via a user subroutine VUMAT. Because no customized  
89 user-subroutines are required with the author's modelling methodology the computational results  
90 using Abaqus implicit can be universally reproduced. Simulation outputs for pin-bearing strengths  
91 will be validated using non-aged test results from Matharu (2014), and in a strength comparison  
92 using data from Mottram and Zafari (2011) for changing the mat reinforcement. The main  
93 contribution of this paper is the predictions from a series of parametric studies to investigate how  
94 pin-bearing strength might vary with changes in the fibre reinforcement.

95

96

## 97 **Fundamental Behaviour and Analysis Approach**

98

99 Bearing failure is a mode showing local crushing and delamination of the laminated material in  
100 direct contact with the steel bolt shaft (Bank 2006; Mottram and Turvey 2003). The bearing strength  
101 of monolithic materials (e.g. steel) is generally evaluated without lateral restraint or through  
102 thickness constraint from a bolt tightening. Application of this test condition with FRP composites  
103 allows delamination fracturing to occur. In this paper the word *delamination* has the meaning for the  
104 formation and growth of a flat flaw in an initially flawless interface (between two previously  
105 'bonded' laminae) that grows into a sizeable delamination crack. With PFRP materials this failure  
106 mechanism leads to the lowest bearing strength.

107 The presence of composite material viscoelasticity (Mottram 2005), and the influence of  
108 structural actions on the bolted connection ensure that it will be unreliable to assume there can  
109 always be lateral restraint at the end of the design working life, which can for a structure be, 50 or  
110 100 years. It is for this reason (Mottram and Zafari 2011) that the *pin-bearing strength* has to be the  
111 strength used in design calculations for bolted connection strength design.

112 The FE study presented next to model the bearing behaviour includes a *continuum damage*  
113 *model* with a *cohesive zone approach*. Continuum damage models address the *intralaminar failure*  
114 *mechanisms* from a global standpoint, whereas if the analyst used individual damage mechanisms  
115 they would be homogenized and constructed around a failure criterion (Knops 2008). This approach  
116 is the least complex and uses the composite lay-up modeller tool within the Abaqus (2016) pre-  
117 processor to define the individual laminae through a laminate thickness. Each layer is cohesively  
118 bonded together to form the *lamination with interfaces* assumed to have a thickness of  $10^{-3}t_{lay}$ ,  
119 where  $t_{lay}$  is the thickness of the thinnest layer either side of the interface. Values for  $t_{lay}$  are  
120 introduced in the section for the *Description of the Model*. This modelling dimension plays the role  
121 of a length scale, and it has been shown by Girão Coelho (2016) that the thickness of the cohesive  
122 interface does not affect the computational performance provided its thickness is small enough  
123 compared to  $t_{lay}$ .

124

125

## 126 ***Modelling of Progressive Delamination Using Cohesive Zone Models***

127

128 Delamination failure is the separation of reinforcing layers from each other, as a consequence  
129 of shear stresses acting in planes parallel to the layers' interfaces and/or tensile stresses acting in the  
130 through-thickness direction. This phenomenon is a typical *crack growth* problem and is treated in  
131 the framework of fracture mechanics (Girão Coelho 2016). Today, the most popular computational  
132 method for the prediction of delamination failure is based on *cohesive zone models* that provide a  
133 natural bridge between strength-based models and energy-based models for fracture. This allows  
134 delamination to be described by a single framework that covers a range of applications for which,  
135 on their own, neither a strength nor an energy criterion might not be sufficient.

136 Cohesive zone models consider fracture as a gradual phenomenon in which separation takes  
137 place across an extended crack tip, or cohesive zone, and is resisted by cohesive tractions (Ortiz and

138 Pandolfi 1999). Thus cohesive zone elements do not represent any physical material, but describe  
 139 the cohesive forces which occur when layers in the lamination are being pulled apart. In FE  
 140 modelling, cohesive zone elements are placed between the continuum elements used to model the  
 141 individual laminae. A key analysis feature of these elements is that they include the effect of first  
 142 delamination failure, and the subsequent crack propagation by means of critical strain energy  
 143 release rates.

144 In our modelling methodology, see Girão Coelho *et al.* (2016), the simulation of interlaminar  
 145 damage is based on the cohesive zone approach using the Abaqus three-dimensional cohesive  
 146 element COH3D8. The study is performed in the quasi-static regime. The traction-separation law  
 147 formulation assumes a non-zero elastic stiffness of the cohesive zone, which is physically motivated  
 148 by the reduced stiffness of the matrix-rich very thin interface layer ( $t_{lay}$ ) as compared to a perfect  
 149 bond assumed to exist between the fibres and the surrounding polymer matrix. From a numerical  
 150 point of view, this elastic stiffness can be understood as a penalty-type enforcement of displacement  
 151 continuity in the elastic range.

152 A quadratic stress criterion is used for the *damage initiation criterion*. To specify the  
 153 conditions for separation in the cohesive zone model the following expression is chosen (Brewer  
 154 and Lagace 1988; Camanho *et al.* 2003):

$$155 \quad \left( \frac{\langle \sigma_n \rangle}{f_I} \right)^2 + \left( \frac{\sigma_s}{f_{II}} \right)^2 + \left( \frac{\sigma_t}{f_{III}} \right)^2 = 1 \quad (1)$$

156 where  $\sigma_n$  is the stress in pure opening mode,  $\sigma_s$  is the stress in the first shear direction,  $\sigma_t$  is the  
 157 stress in the second shear direction,  $f_I$  (opening),  $f_{II}$  (sliding)) and  $f_{III}$  (tearing) are the peak strength  
 158 values in the same directions, and:

$$159 \quad \langle \sigma_n \rangle = \sigma_n \quad \text{for} \quad \sigma_n > 0 \quad \text{and} \quad \langle \sigma_n \rangle = 0 \quad \text{for} \quad \sigma_n \leq 0. \quad (2)$$

160 The latter modelling constraint in Eq. (2) is because compressive normal stresses cannot open a  
 161 delamination crack.

162        *Progression of damage at the interfaces* is modelled using a linear softening law and a critical  
163 mixed mode energy behavior based on the Benzeggagh-Kenane criterion (1996), which is described  
164 by the following expression:

$$165 \qquad G_c = G_{I,c} + (G_{II,c} - G_{I,c}) \left[ G_{II} / (G_I + G_{II}) \right]^\eta. \qquad (3)$$

166        In Eq. (3)  $G_{m,c}$  (with  $m = I, II, III$ ) is for the total critical strain energy release rate associated  
167 with delamination mode  $m$ , and  $\eta$  is for the semi-empirical criterion exponent applied to  
168 delamination initiation and growth. Based on the argument given by Girão Coelho (2016) exponent  
169  $\eta$  is assumed to be 1.5. Illustrated in Figs. 1(a) to 1(c) are the three distinct opening modes that can  
170 occur singularly or interact together to cause the initiation and formation of a delamination failure.

171        The use of cohesive zone models requires that a very fine mesh specification is used to ensure  
172 that sufficient interface elements exist within the cohesive zone length where the crack tip is  
173 moving. If the mesh design happens to be too coarse, the cohesive stress at the discontinuity may  
174 not even reach the interfacial strength and, as a result, the required failure mode is missed. Falk *et*  
175 *al.* (2001) suggest a minimum of two to five elements in this cohesive zone length in order to  
176 perform a reliable simulation. Turon *et al.* (2000) indicate that for typical graphite-epoxy or glass-  
177 epoxy FRP materials, the length of the cohesive zone should be smaller than one or two millimetres.  
178 As a consequence, the mesh size required in order to have more than two elements in the cohesive  
179 zone should be smaller than half a millimetre. For full-sized structural models of pultruded shapes  
180 and structures this has the obvious consequence of a computationally expensive solution. Current  
181 numerical simulations for the 80×80×9.5 mm pin-bearing specimen comply with the cohesive zone  
182 element mesh size requirements specified by the first author (Girão Coelho 2016) to allow stable  
183 numerical simulations of interface delamination.

184

185

## 186    **Finite Element Validation of Strength Behaviour**

187



188 A general FE model for the pin-bearing problem is developed and validated in this section using  
189 experimental strength results from the PhD work by Matharu (2014). Note that the objective of the  
190 FE work is to obtain predictions for the pin-bearing strength, which is established by the  
191 maximum test load (or stress, which is this load divided by the projected bearing area given by pin  
192 diameter times specimen thickness). There are engineering/scientific reasons why the FE  
193 simulations might not be numerically reliable in predicting the onset of damage and all of the modes  
194 of failure that occur prior to peak load. There is a likelihood that prior to loading there are already  
195 matrix cracks (running parallel to the unidirectional fibre) because of residual thermal strain owing  
196 to the cooling down from a temperature of 150°C in the pultrusion composite processing. What is  
197 known from the extensive programme of static strength tests by Matharu (2014), and by Mottram  
198 and Zafari (2011), is that when the bearing load is aligned in the direction of pultrusion the load-  
199 stroke curve (testing machine displacement) is virtually linear elastic until to ultimate failure, which  
200 always occurs with a noticeable load reduction and audible acoustic emissions. This signal of  
201 ultimate failure is too pronounced to be for initial/new matrix cracking, and when the post-failure  
202 load is released inspection of the bearing surface shows there to be interfacial delamination  
203 fractures. This is the authors' evidence-based justification for how the FE investigation was carried  
204 out and reported in this paper.

205 Matharu (2014) tests are for pultruded material taken from the web of a Pultrex®  
206 SuperStructural wide flange shape with nominal thickness of 9.53 mm (3/8 in.) (Creative  
207 Pultrusions 2016); the measured thickness is closer to 9.6 mm. The test matrix has plain steel pins  
208 of four metric diameters from 10 to 20 mm diameter with a clearance hole size that introduced the  
209 maximum allowable fabrication tolerance using the guidance in Anonymous (2012) from the  
210 Pultrusion Industry Council, USA. The technical reason for the maximum is that the bigger the hole  
211 clearance is, the lower is the pin-bearing strength (Yuan 1996), and for a safe design the lowest  
212 characteristic strength is required (Mottram and Zafari 2011). Without the fabrication tolerance the  
213 recommended clearance size is 1.6 mm (or 1/16 in.) (Anonymous 2012; Creative Pultrusions 2106;

214 Strongwell 2012). The nominal clearance ( $d_0 - d$ ) is 2.2 mm (given by 1.6 mm clearance +0.4 mm  
215 tolerance) for M10 and 2.4 mm (1.6 mm clearance +0.8 mm tolerance) for M12, M16 and M20 bolt  
216 sizes. After drilling a pilot hole of diameter less than required the final hole diameter to  $\pm 0.02$  mm  
217 was prepared using a Cincinnati Arrow 450 Milling machine. Batches of ten nominally identical  
218 specimens were tested to determine the pin-bearing strength that is statistically analysed and  
219 critically evaluated by Matharu (2014).

220 The geometry was taken comparable to the experimental set-up, with the pin diameter as FE  
221 modelling parameter. Fig. 2 defines the specimen geometry and shows the layered structure of the  
222 SuperStructural web material.

223 Specific characteristics and attributes that were incorporated into the modelling are:

- 224 1. Based on the relatively high stiffness of the steel pin with respect to the longitudinal web  
225 material, *c.f.* 210 kN/mm<sup>2</sup> (BS EN 1993-1-1:2005) to 21.3 kN/mm<sup>2</sup> (Creative Pultrusions  
226 2016)), the bearing load is exerted by a rigid (circular smooth) pin, this is shown in Fig. 2.
- 227 2. Interaction between pin and hole is not only modelled with a normal bearing load, but also  
228 with a through-thickness friction having a friction coefficient taken equal to 0.25, based on  
229 the testing by Mottram (2005) to account for the FRP/steel contact.
- 230 3. Residual thermal stresses resulting from the pultrusion thermal-induced and cure-induced  
231 shrinkage processes are not considered. Chen *et al.* (2001) and Zhang *et al.* (2004) developed  
232 micro-mechanical models that have been successfully employed in computational analysis  
233 and showed that initial residual stresses within pultruded structural shapes dissipate over time  
234 owing to viscoelastic creep relaxation (Bank 2006).

235

236

## 237 **Material Properties**

238 This section describes the physical and mechanical properties of the original material used in  
239 the extensive testing programme by Matharu (2014). The pultruded material is from the Pultex®

240 SuperStructural 1525 series of ‘off-the-shelf’ shapes. Identifier 1525 means the thermoset  
241 polyester matrix is Class 1 Flame Retardant. The E-glass FRP fibre architecture consists of mat  
242 layers interspersed with nominally constant thickness layers of UniDirectional (UD) rovings having  
243 a fibre volume fraction of 60%. The mat reinforcement is type E-TTXM 4008, a Triaxial Stitched  
244 Fabric Mat (TSFM) from Vectorply Corporation ([http://vectorply.com/wp-](http://vectorply.com/wp-content/uploads/2015/06/E-TTXM-4008.pdf)  
245 [content/uploads/2015/06/E-TTXM-4008.pdf](http://vectorply.com/wp-content/uploads/2015/06/E-TTXM-4008.pdf)). The pultruded shape has an outer (relatively thin at  
246 0.03 mm (see Table 1)) surface veil (non-structural) for UV protection and a barrier to moisture  
247 diffusion. The NEXUS<sup>®</sup> veil layer is resin-rich, non-structural and consists of randomly orientated  
248 short 100% melt polyester fibres, which has a mass per unit area  $< 100 \text{ g/m}^2$ .

249 As seen in Fig. 2 there are five alternating layers of UD and TSFM, with two TSFM layers at  
250 mid-thickness having no UD layer between them. TSFM is of a stitched continuous fabric having  
251 four layers with the lay-up sequence of  $+45^\circ/90^\circ/-45^\circ/\text{random}$  chopped strand. Assuming the  
252 volume fraction of fibres in the continuous layers is 60% the chopped strand backing layer in TSFM  
253 has a volume fraction of 16%. The thicknesses of the layers are 0.35 mm for the  $90^\circ$ , 0.51mm for  
254 the  $\pm 45^\circ$  and 0.54 mm for the chopped strand. It is known from Creative Pultrusions Inc. that the  
255 UD layers comprise of 56 yield rovings.

256 The thermoset resin is the Reichhold 31031 unsaturated isophthalic polyester resin. To produce  
257 a matrix for the pultrusion composite processing there are additives and fillers. The largest  
258 proportions in the formulation, which consist of 13 constituent parts, is for the polyester resin, at  
259 approximately 80% of the bulk matrix.

260 A series of resin burn-off tests were conducted by Matharu (2014) to estimate the volumetric  
261 proportions of UD and TSFM fibres, and matrix, as well as to establish the nominal thicknesses of  
262 each layer. The resin burn-off procedure was adapted from the method described in Appendix B of  
263 the PhD thesis by Lane (2002), and with reference made to the testing procedure given in Ye *et al.*  
264 (1995) and ASTM D2584-11 (2011). Table 1 summarizes the results from the resin burn-off tests.  
265 Because the reinforcing effect of the Surface Veil (SV) layers is minimal it can be seen that the total

amount of glass fibre with the two main reinforcing laminae of UD and TSFM are equal at 34% volume fraction. The volume percentages of UD, TSFM and matrix are in the range 30 to 34. The final row in Table 1 reports the nominal thicknesses of the layers (the  $t_{\text{layS}}$ ) on the assumption that they are constants for a particular reinforcement type in the original material. Note that both the veil and TSFM thicknesses are predetermined by their construction. Only the UD layers have a freedom to have a processing variation for layer thickness.

The mechanical properties presented in Table 2 are defined with respect to the local co-ordinate system, with: 1 for the pultrusion direction; 2 for the in-plane direction normal to the direction of pultrusion; 3 for the through-thickness direction. In the literature the direction of pultrusion is often referred to as the longitudinal (or length-wise) direction and the perpendicular (in-plane) direction as the transverse (cross-wise) direction. Notation  $f_1$  is therefore for the in-plane longitudinal strength,  $f_2$  for the in-plane transverse strength,  $f_{1,s}$  for the in-plane shear strength, and  $f_{2,s}$  for the transverse shear strength. Values in the table for Tensile (T) and Compressive (C) actions are for in-plane strengths.

The determination of the elastic constants for the three different laminae are reported in Table 2 was carried out using micromechanical modelling with volume fractions of the constituents established by using the resin burn-off method, as described in Lane (2002). In making the micromechanical modelling calculations, it is assumed that the densities of matrix and glass fibres are 1.1 g/cm<sup>3</sup> (Reichhold, 2006) and 2.56 g/cm<sup>3</sup> (Hancox and Mayer, 1994), respectively. The modulus of elasticity for the matrix constituent is assumed to be 3.2 kN/mm<sup>2</sup> (Reichhold, 2006) and for the fibres it is taken as 72 kN/mm<sup>2</sup> (Hancox and Mayer, 1994). The constituent Poisson's ratios are assumed to be 0.36 and 0.22, respectively. Using the rule of mixtures approach with the two lamina stiffnesses in Table 2 predictions for the elastic constants of the 9.6 mm thick panel are  $E_1 = 25.0$  kN/mm<sup>2</sup>,  $E_2 = 16.7$  kN/mm<sup>2</sup>,  $G_{12} = 4.5$  kN/mm<sup>2</sup> and  $\nu_{12} = 0.36$ . These elastic constants are seen to be different to the tabulated properties of  $E_1 = 21.3$  kN/mm<sup>2</sup> and  $E_2 = 9.6$  kN/mm<sup>2</sup> in the Design Manual from Creative Pultrusions (2016). The Design Manual stiffnesses are for a range of

292 pultruded shape sizes and the larger the size is, the lower can be the volume proportion (and maybe  
293 fibre volume fraction) of the unidirectional reinforcement. As a consequence of this fact the  
294 tabulated moduli of elasticities are for minimum measured from all the shapes in the series range.  
295 Because the published stiffnesses are for the whole 1525 series they are less reliable, and this is the  
296 engineering reason why the micromechanical modelling laminae stiffnesses in Table 2 are input  
297 data for the FE analyses. The web material consists of alternative reinforcing layer of UD and the  
298 TSFM having a balanced lay-up, comprising TSFM1 layer at 1.38 mm thickness, UD1 at 1.93 mm,  
299 TSFM2 at 1.42 mm, TSFM3 at 1.39 mm, UD2 at 2.01 mm and TSFM4 at 1.47 mm. Note that mid-  
300 thickness layers TSFM2 and TSFM3 do not have UD reinforcement between them. Accounting for  
301 experimental errors in the resin burn-off testing the thickness of the four TSFM layer can be taken  
302 to be constant at 1.4 mm. The UD layers have a nominal constant thickness of 2.0 mm.

303 The strengths listed in the middle column in Table 2 are for the laminate. PFRP materials  
304 having the individual fibre reinforcement types (and the same matrix) are not pultruded and so their  
305 strength (and stiffness) properties cannot be measured. Matharu (2104) conducted a series of  
306 tension and compression coupon tests (in batches of 10) to determine the panel's in-plane strengths.  
307 The longitudinal (compression) strength ( $f_{1,C}$ ) for the batch ranges from 280 to 328 N/mm<sup>2</sup>, whereas  
308  $f_{2,C}$  is found to range from 99 to 153 N/mm<sup>2</sup>. The mean longitudinal strength in tension ( $f_{1,T}$ ) at 294  
309 N/mm<sup>2</sup> is 4% lower than the mean compression, whereas the transverse tensile strength ( $f_{2,T}$ ) is, as  
310 expected, much lower, at 63% of the compression mean of 133 N/mm<sup>2</sup>. The mean in-plane shear  
311 strength  $f_{1,s}$  is 91 N/mm<sup>2</sup> was taken from D'Alessandro (2009). In the FE work the transverse shear  
312 strength ( $f_{2,s}$ ) is taken to be  $f_{1,s}$ .

313 Each (continuum) damage evolution law includes a corresponding fracture energy (or fracture  
314 toughness),  $G_c$ , that governs crack growth for the modes illustrated in Fig. 1. Fracture energy is  
315 defined as the work needed to create a unit area of a fully developed crack. Guidelines for the  
316 evaluation of these fracture properties can be found in Pinho *et al.* (2006) and Maimi *et al.* (2007).  
317 The four values adopted in this study are listed in the last column in Table 2. They were not

318 experimentally determined, but estimated based on research expertise and data available from the  
319 literature (Kelly and Zweben 2000; Girão Coelho *et al.* 2015).

320 The assumed modelling inputs for interfacial mechanical properties are summarized in Table 3.  
321 To establish the peak strengths  $f_I$  and  $f_{II} = f_{III}$  for Eq. (1), in the absence of experimental  
322 measurements, the in-plane strengths  $f_{2,T}$  and  $f_{1,S} = f_{2,S}$  were factored down by adopting a  
323 weakening factor  $f_w$ , according to guidance from the failure criterion work by Puck and Schürmann  
324 (2002), see Knops (2008), and from the computational modelling work by Wimmer *et al.* (2009).  
325 The critical normal interface traction (for opening mode) of the cohesive zone elements is therefore  
326 assumed to be equal to the transverse tensile strength,  $f_{2,T}$ , times the weakening factor. The critical  
327 shear interface tractions (for sliding and tearing modes) are both assumed to be equal to the shear  
328 strength,  $f_{2,S}$ , times the same weakening factor.

329 It was found in a sensitivity analysis from a series of FE simulations that the overall response  
330 for  $f_w = 0.25$  gives the one closest to the pin-bearing experimental results from Matharu (2014).  
331 With  $f_I = 21 \text{ N/mm}^2$  and  $f_{II} = f_{III} = 23 \text{ N/mm}^2$  the analysis indicates a relatively weak interface  
332 between the laminae, as defined by the FE modelling data given in Fig. 2 and Table 2. A similar  $f_w$   
333 has already been proposed by Girão Coelho *et al.* (2016) for this same pultruded material when  
334 analysing the different structural engineering problem of delamination failure in a pultruded leg-  
335 angle shape subjected to tying force, as would be found in a web clip connection between beam and  
336 column members.

337 The piecewise linear traction-separation law of the cohesive elements is described according to  
338 the lamina strength presented in the last column in Table 2. Using the guidance from Camanho *et*  
339 *al.* (2003) the elastic stiffnesses (per unit area) of the interface ( $K_I$ ,  $K_{II}$  and  $K_{III}$  associated with  $f_I$ ,  $f_{II}$   
340 and  $f_{III}$ ) are assumed to be equal to  $10^6 \text{ N/mm}^3$ . For modes I, II and III illustrated in Fig. 1 the  
341 critical energy release rates of  $G_{I,c}$  and  $G_{II,c} = G_{III,c}$ , were taken as 0.2 (200) and 0.5 (500) N/mm  
342 ( $\text{J/m}^2$ ), respectively (Girão Coelho 2016).

343

344

## 345 **Description of the Model**

346 Defined in Fig. 2 are the dimensions for the pin-bearing specimen that is to be modelled and  
347 analysed. Each plate of 9.6 mm thickness ( $t$ ) is 80 mm long ( $e_1$ ) and 80 mm wide ( $w$ ). The pin  
348 diameter ( $d$ ) is set equal to 11.8 mm for M12 and to 19.8 mm for M20 steel bolting, while the hole  
349 diameter ( $d_0$ ) was taken 2.4 mm larger, based on practical dimensions. This allowed for the  
350 maximum clearance hole tolerance size (Anonymous 2012) and the shaft diameter to be identical to  
351 the bolt diameter specification. The pin was placed centrally in the hole, which itself is mid-  
352 positioned in the top surface. The UD layers are oriented with the fibres aligned to the direction of  
353 the bearing load.

354 The FE model is constructed from stacked continuum shell elements with the individual UD  
355 and TSFM layers being modelled separately. Although the element SC8R possesses the geometry  
356 of a brick, its kinematic and constitutive behaviour is similar to those of conventional shell elements  
357 (Abaqus 2016). The continuum shell elements are able to reproduce reliable results in simulations  
358 of thin-walled laminated structures by means of only one element in a lamina thickness, owing to a  
359 higher-order displacement field (Parisich 1995; Remmers *et al.* 2003). Cohesive interface elements  
360 are used to connect together an TSFM and an UD layer to allow for (multi-)delamination failures to  
361 occur. As shown in the diagram in Fig. 2 four cohesive interfaces I1 to I4 are modelled. Justification  
362 for no cohesive interface between the two touching TSFM layers at mid-thickness is that Matharu  
363 (2014) did not observe, after tests, delamination failure over this interface.

364 The in-plane mesh geometry for the interfaces had to be the same as that for the laminae. A  
365 typical mesh with its very refined mesh specification is illustrated in Fig. 3, where adjacent to the  
366 hole a finer mesh is used to correctly capture delamination and the important contact situation. Note  
367 that using mirror symmetry about the two TSFM layers at mid-thickness the mesh is for a thickness  
368 of 4.8 mm (or  $t/2$ ). The thickness of interface layers I1 and I2 are  $10^{-3}t_{lay}$ , based on the 1.4 mm  
369 TSFM layer thickness. Element size was increased towards the plate edges to reduce calculation

time. The FE mesh specification in our study of the original material has a total of 27834 (M12) continuum shell SC8R elements (or 32520 elements for the M20 specimen), and 18556 COH3D8 cohesive elements (M12 and two interfaces) (or 21680 elements for M20 specimen and two interfaces). Interfaces I1 and I2 (or I3 and I4) are included in the FE model.

As mentioned earlier the (steel, smooth shafted) pin is modelled as a three-dimensional analytical rigid body revolved shell.

Loading of the specimen is simulated by displacing the bottom edge against the fixed pin as indicated by the ‘load’ arrows in Fig. 2. Vertical deformation is resisted by the contact surface for pin and hole perimeter at the top of the specimen. Contact behavior is modelled with the ‘hard’ surface-to-surface contact formulation in the normal direction and the friction is modelled using an isotropic friction model with a coefficient of 0.25. Note that the coefficient of friction was not measured (Mottram 2005) for the steel-FRP contact in the through-thickness direction. This represent one of several modelling data assumptions made to complete a successful FE simulation and they have been reported in this paper.

The Abaqus implicit analysis is run with the laminae properties listed in Tables 2 assigned to the appropriate UD or TSFM layers and with the interfacial properties in Table 3. For each lamina the Hashin failure criterion (Hashin 1983) is used to predict damage onset (Girão Coelho 2016), and because interfacial delamination failure always occurs first this criterion is of secondary importance. To account for damage progression after crack initiation there is a fracture energy calculated for each of four failure modes using the fracture energies in Table 2 (Girão Coelho *et al.* 2015). To assist with numerical convergence of the Abaqus solver the authors used previous simulation outputs (Girão Coelho *et al.* 2015) to set a global stabilization factor to  $2 \times 10^{-4}$ .

## **Modelling Strategy Validation**



395 Results from static analyses are compared with maximum pin-bearing loads from experiments.  
396 In the series of tests a specimen's stiffness was not measured because it had no meaning to the aim  
397 of the strength investigations in Matharu's PhD work (2014), which were to establish characteristic  
398 values for pin-bearing strengths. In this paper the web material studied by Matharu is referred to as  
399 the original (non-aged) material.

400 One finding from the FE simulations is that delamination failure always governs the magnitude  
401 of the load for pin-bearing strength and this matches experimental observations. As a direct  
402 consequence the relevant Hashin parameters were found to be always below 1.0. This shows that it  
403 has been acceptable for the authors to have made the approximate assumption that the laminate  
404 strengths in Table 2 could be taken to be the individual laminae strengths. The pin-bearing strength,  
405 here expressed in terms of applied load, is reached when there is extensive delamination growth at  
406 the two interfaces (I1 and I2), and there is a sudden load drop-off during the displacement  
407 controlled loading. Fig. 4 is for plots constructed from Abaqus outputs of load (kN) versus the axial  
408 displacement (mm) in the bearing force direction. The grey shaded circle symbols are for numerical  
409 results with the larger black filled circle at the load when initiation of delamination fracturing is  
410 numerically predicted. In the legend there is (I1) to identify that the critical delamination is at  
411 Interface 1, its location is defined in Fig. 2. The batch test results from Matharu (2104) are reported  
412 by three representative loads. The batch mean is given by the solid horizontal line, and the  
413 maximum and minimum specimen loads are given by the upper and lower dashed lines. The load  
414 range between the two dashed lines gives the measured variation from testing a batch of 10  
415 nominally identical specimens. The coefficient of variation for the pin-bearing strength (Matharu  
416 2014) for the M12 batch is 10.4 and for the M20 batch is 8.8%.

417 Fig. 4(a) is for the M12 pin, and using the same axis scales the equivalent computational results  
418 for the M20 pin are reported in Fig. 4(b). The predicted load-displacement response is seen to start  
419 off linear, and progress with non-linearity above 15 kN for M12 and 22 kN for M20. The  
420 computational response is seen to differ from the virtually linear load-stroke relationship recorded

421 by the testing machine. Above the loading (defined by filled circle symbol) for delamination  
422 initiation there is a reduction in structural stiffness and the load-displacement response softens with  
423 progressive delamination growth, before ultimate failure. Stiffness degradation after delamination  
424 initiation and progressive growth is not an outcome in testing since delamination onset is believed  
425 to occur at the same instance the peak load is reached. Importantly, should the FE modelling  
426 methodology presented herein be used to aid the design of bolted connections the predicted load for  
427 delamination initiation is useful because as Figs. 4(a) and 4(b) show it is below the lower bound test  
428 result.

429 Returning to the information in Fig. 4, the peak (Abaqus) loads of 23 kN (Fig. 4(a) for M12)  
430 and of 35 kN (Fig. 4(b) for M20) compare favourably with the mean experimental loads (Matharu  
431 2014) of 22 kN and 34 kN, respectively. The predictions are in very good agreement with computed  
432 strengths only 4.5% and 6% higher. The positive correlation shown by the results in Figs. 4(a) and  
433 4(b) for two independent batches provides good validation for applying the FE modelling approach  
434 to determine pin-bearing strengths.

435 In Fig. 5 the deformed shape is shown at peak load (magnification factor is 2.5). Fig. 5(a) is for  
436 the M12 pin and the maximum load of 23 kN, whereas Fig. 5(b) is for the M12 pin at maximum  
437 load of 36 kN. In both parts the deformations clearly shows the typical ‘brooming’ for compression  
438 induced delamination failure at interfaces I1 (nearest) and I2 (farthest).

439 Inspection of the localised through-thickness deformation in Fig. 5 helps to explain why  
440 delamination failure is the mechanism that governs the pin-bearing strength. Under the increasing  
441 pressure from the bearing (rigid) pin the existence of Poisson’s ratio effect is for the laminate to  
442 need to expand freely in the through-thickness direction. This physically cannot happen because of  
443 the friction restraint over the steel pin/FRP contact area and the volume of the surrounding FRP  
444 material resisting the bearing load. As seen in Figs. 5(a) and 5(b) the deformation is in the form of a  
445 bulge with a short wavelength ( $\cong 0.44d_0$ ). A complex, through-thickness stress field with tension (in  
446 the  $z$ -direction) is generated local to the hole perimeter, and within the influence of the bulge zone.

447 It can be speculated that when the maximum through-thickness tensile stress reaches  $f_I$ , or the limit  
448 given by Eq. (1) when  $f_I$  interacts with  $f_{II}$  and  $f_{III}$  there will be cohesive failure for initiation of  
449 delamination failure. Propagation of a delamination crack to the final size shown, for example, in  
450 Fig. 6 is numerically controlled by fracture mechanics (Girão Coelho 2016) and the energies listed  
451 in Table 3.

452 The contour plots presented on interface surfaces in Fig. 6 show that, at maximum load,  
453 delamination fracturing at the two interfaces has progressed from the hole into the body of the  
454 specimen. Fig. 6 parts (a) and (b) are for the M12 and M20 bolt diameters, respectively. Damage  
455 progression can be tracked, from 0 (none) to 1 (complete), using the Abaqus output parameter  
456 Stiffness DEGradation (SDEG), which indicates the state of damage in the cohesive elements and  
457 thereby provides insight into damage initiation and propagation. The red coloured area (SDEG is  
458 0.7 to 1.0) is for complete interface separation and the thin (green coloured) zone around the  
459 damage zone's perimeter is for partial damage (SDEG is 0.3 to 0.7). Where the material on the  
460 interface surface is coloured dark blue there is no damage (for SDEG < 0.3), with full continuity  
461 across the interface. Interface I1 is on the left-side and interface I2 on the right-side. One  
462 observation from the contouring presented in Fig. 6 is that the delamination damage has progressed  
463 from an initial state with a stable crack front; this is evidence for bearing failure offering the bolted  
464 connection a level of damage tolerance.

465 Overall, we find that the predicted delamination cracks are fairly repeatable and overall are  
466 representing the visual observations of bearing surface cracking made by Matharu (2014). The  
467 numerical results presented in Figs. 4 to 6 demonstrate the ability of the FE modelling methodology  
468 to appropriately predict the pin-bearing mode of failure where interaction between intralaminar and  
469 interlaminar modes is strongly coupled.

470

471

## Changing the Bearing Strength for Bolted Connections

It is well-known that by increasing the size of the clearance hole the bearing strength decreases (Yuan *et al.* 1996). Figs. 7 and 8 are equivalent to Figs. 4 and 6. In Fig. 7 there are two further load-displacement results for clearance holes of 0 mm (tight fitting) and 1.6 mm (without fabrication tolerance (Anonymous 2012)). Although there is a gain in delamination initiation and maximum loads on reducing clearance from 2.6 to 1.6 mm, there is a significant increase in their values, at 48% (M12) and 39% (M20), for the no clearance situation. Even the change in load on having the clearance equal to the nominal design guideline of 1.6 mm (Creative Pultrusions 2016; Strongwell 2016) justifies why testing for the characteristic strength (Mottram and Zafari 2011; Matharu 2014) has to be with the maximum practical clearance. Illustrated in Fig. 8 are the delamination zones for no clearance and the maximum loads of 25 kN for M12 (Fig. 8(a)) and of 37 kN for M20 (Fig. 8(b)). By inspection it can be seen that the shape of the delamination cracks at interfaces I1 and I2 are broadly similar to those in Figs. 6(a) and 6(b) when the clearance hole is 2.6 mm. Because the tight fitting situation generates pressure over the whole semi-circular notch there is interfacial failure in the laminate beyond the boundary of the hole. This finding shows that the test method applied by Matharu (2014) will not give the actual pin-bearing strength if it is practical on- or off-site to assemble a bolted lap-joint with no hole clearance. By applying our FE modelling approach it is feasible to establish the clearance size at which the delamination growth at peak load first goes beyond the confines of the bolt hole perimeter.

Keeping all other parameters constant to those in Matharu's testing, the next change to be evaluated by a sensitivity analysis is the relative thicknesses of the UD and TSFM layers (for a constant specimen thickness ( $t$ ) of 9.6 mm). For both bolt diameters Table 4 reports the numerical results of: elastic stiffness ( $S_{el}$ ); pin-bearing load for initiation of delamination ( $R_{br,ID}$ ); maximum pin-bearing load ( $R_{br,max}$ ); the two load ratios of  $R_{br,ID}/R_{br,max}$  and  $R_{br,max}/R_{br,max,baseline}$ , where  $R_{br,max,baseline}$  is the maximum pin-bearing load for the original material (i.e. 23 and 36 kN for M12 and M20, respectively). The original material's predictions are presented in the first row of

498 numbers, with three parametric variations given in the next three rows. The percentage of TSTM in  
499  $t$  is 58% in the original material and ranges from 42 to 67% in the three sensitivity analysis  
500 materials. As expected,  $S_{el}$  increases as the UD percentage increases. Combining the M12 and M20  
501 pin results it is found that neither  $R_{br,ID}$  nor  $R_{br,max}$  show any definite trends.  $R_{br,max}$  is seen to remain  
502 virtually constant at 23-24 kN (M12) or 35-36 kN (M20). These observations are highlighted by the  
503 virtually constant load ratios for the two bolt sizes.

504 The next study investigates the effect of changing the thickness of the mid-thickness TSFM.  
505 One reason for this parametric study is that the maximum load is governed by delamination failure  
506 at interface I1, not I2 that is for the outer interfaces of the mid-thickness mat layer. The presentation  
507 of information and computed predictions in Table 5 is the same as in Table 4. The parameter change  
508 is given in the second column where the thickness of the mid-thickness TSFM reduces in four  
509 increments to  $0t_{orig}$  from its original specification of 2.8 mm ( $1.0t_{orig}$ ). The loss in the thickness of  
510 the middle TSFM is replaced by UD reinforcement, shared equally by the two equal-thickness UD  
511 layers. One observation is that the material with  $0t_{orig}$  and the M12 pin has the highest  $R_{br,ID}$  at 21  
512 kN. With the M20 bolt size the highest  $R_{br,ID}$  of 33 kN is for  $0.15t_{orig}$ . There is, again, not a  
513 significant change in  $R_{br,max}$  with the parametric variation, which for M12 is either 23 or 25 kN, and  
514 for M20 is 36, 37 or 38 kN.

515 Note that the delamination contours for the specimens covered in Tables 4 and 5 were found to  
516 be similar to those shown in Fig. 6. They are not reported in this paper because of lack of space.

517 Returning to the original study parameters (Matharu 2014), the next variable to be examined is  
518 reversing, in the laminate, the stacking sequence of layer TSFM1 or layer TSFM2 or of both. In the  
519 original material interfaces I1 and I2 had UD fibres on one side and  $+45^\circ$  fibres from TSFM on the  
520 other side. By reversing the TSFM layer the interface has UD and chopped strand fibres in contact.  
521 Information in Table 6 follows Tables 4 and 5 with the stacking sequence changes given in the  
522 second column. The results are difficult to interpret with little change in predictions with the M12

pin and a significant increase to 1.16 in ratio  $R_{br,max}/R_{br,max,baseline}$  when the larger M20 pin is employed.

In the 1990s the wide flange shapes (Mottram and Zafari 2011) from Creative Pultrusions Inc. (2016) did not have a triaxial mat reinforcement. Prior to introducing the Pultrex® SuperStructurals product range the mat reinforcement was an CSM, having a local glass fibre volume fraction of  $\cong 23\%$ . For the next study the material has either TSFM or CSM for the mat reinforcement. For these simulations the model did not have the mid-thickness mat reinforcement. Predictions are presented in Figs. 9 and 10 in the usual way. In the plots in Fig. 9 the load-displacement response for the original laminate (Matharu 2014) is given by the shaded filled circle symbols. Superimposed on this curve are the predicted results when the TSFM is fully replaced with CSM. The higher pair of curves in Figs. 9(a) and 9(b) is for the situation where the mid-thickness mat is replaced by UD reinforcement, thereby increasing  $S_{el}$ . A difference observed is that the maximum load is slightly higher with the ‘older’ mat reinforcement and it occurs at a higher axial displacement. The positions of the open unfilled circle for the load at delamination initiation show that another difference on having CSM reinforcement is that failure is delayed by about 2 kN (M12) or 4 kN (M20). A key finding is that the maximum loads are the same at 25 kN (M12) and only slightly different at 38 and 40 kN (M20) after replacing the TSFM with CSM. An important conclusion from this study is that the presence of  $\pm 45^\circ$  fibres in the TSFM layers has not increased the pin-bearing strength.

What is observed from looking at the contour plots in Figs. 10(a) and 10(b) for delamination growth at maximum load is that the shape of the fractured zone is much more centrally located and goes further into the specimen when the mat reinforcement is CSM. The area associated with partially damaged material is seen to be larger too. These figures show that the size and shape of a delamination crack will depend on the orientations of the fibres in the two layers adjacent to the failed interface.

The final two FE simulations were performed to find out if pin-bearing strength increases when the chopped strand backing in the TSFM layer is replaced by proportional increases in the  $90^\circ$  and

549  $\pm 45^\circ$  fibre reinforcements. The rationale for this parametric study is that the chopped strand layer  
550 has a relatively low fibre volume fraction at 16% and occupies 39% of the TSFM's 1.4 mm  
551 thickness. The predicted results are reported in Figs. 11 and 12. As observed from the two curves in  
552 Fig.11 there is little difference in the response up to when delamination failure initiates. Perhaps,  
553 unexpected it is found that the original TSFM with the chopped strand backing has the higher  
554 maximum load for both M12 (i.e. 23 and 18 kN) and M20 pins (i.e. 35 and 29 kN). Comparing the  
555 equivalent contour plots in Fig. 12 with those presented in Fig. 6 it is seen that shape and size is  
556 dependent on the fibre reinforcement local to the failing interface.

557 Results for this, and other parametric studies reported in this paper have shown that pin-bearing  
558 strength can be highest when the mat reinforcement has random continuous or chopped fibres at the  
559 relatively low volume fraction of  $< 25\%$ .

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561

## 562 **Concluding Remarks**

563

564 A FE modelling methodology using Abaqus software, without customized user-subroutines, has  
565 been formulated that can be employed in parametric and/or sensitivity analyses to determine the  
566 pin-bearing strength of laminated composite materials. Numerical outputs are able to predict the  
567 initiation and growth of delamination failures that governs what a pin-bearing strength is. Using  
568 batch test results for a 9.6 mm thick web from a pultruded structural shape the simulation results for  
569 the two pin diameters of 12 and 20 mm are shown to give good strength predictions. To assist  
570 designers of bolted connections an interesting finding is that initiation of delamination cracking is  
571 not at the peak load, and that this predicted load is close to, yet below the lowest pin-bearing  
572 strength measured from a batch of 10 specimens. There was, however, no observable evidence that  
573 there was stable delamination failure(s) during testing before the pin-bearing strength load was  
574 reached. Although this finding shows that the computational solution differs from the actual pin-

575 bearing response an important finding is that the FE modelling methodology can reliably establish  
576 the ultimate failure load for pin-bearing failure.

577 To demonstrate the potential of applying the modelling approach to investigate how the pin-  
578 bearing strength might vary with parameter changes a number of parametric studies are presented.  
579 While maintaining constant the other parameters in the test series, numerical predictions are  
580 reported for changing the: clearance hole size; relative thickness of the unidirectional and triaxial  
581 mat layers; thickness of the mid-thickness triaxial mat layer; stacking sequence of the triaxial mat  
582 layer; triaxial mat to a continuous strand mat; triaxial mat's construction so that the 39% thickness  
583 of chopped strand backing is replaced proportionally with reinforcement having the orientations of  
584  $\pm 45^\circ$  and  $90^\circ$ .

585 The main findings from the numerical results can be summarized as follows. New  
586 understanding is obtained from contour plots for the Abaqus output Stiffness DEGradaion showing  
587 the size and shape of delamination failure(s) over interfaces. As expected, pin-bearing strength is  
588 found to increase with reduced hole clearance and is significantly higher (at 48% and 39% for the  
589 two pin diameters) when there is no clearance. Under the tight fitting condition it is observed that  
590 the test method (Matharu 2014) becomes invalid because, at peak load, delamination failure is not  
591 contained within the semi-circular hole. A less expected finding is that the strength is not sensitive  
592 to the relative thicknesses of the unidirectional and mat layers, and there is no benefit on having  
593 replaced the continuous strand mat with a triaxial mat (which is more expensive). Furthermore,  
594 tabulated results from parametric studies do not offer obvious trends, and the authors cannot give  
595 any scientific explanation to why computed results are seen to be highly dependent on the pin  
596 diameter being either 12 or 20 mm. The final study shows that strength is reduced on replacing the  
597 chopped strand backing in the triaxial mat layers; the opposite might be expected since the  
598 additional  $\pm 45^\circ$  fibres ought to be effective in resisting a higher bearing load.

599 The satisfactory performance of the computational predictions in determining the response at  
600 peak load encourages the authors to recommend that the Abaqus modelling methodology may be



601 used to: (i) design the laminate (produced by the pultrusion composite processing method) for a  
602 specified pin-bearing strength; (ii) predict the pin-bearing strengths for bolted connections having  
603 parameters to be scoped in a structural design standard.

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607

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705

706 **Fig. 1.** Modes of delamination failure: a) Mode I ( $m = I$ ): opening; b) Mode II ( $m = II$ ): sliding; c)  
707 Mode III ( $m = III$ ): tearing.

708 **Fig. 2.** Geometry: schematic of plate and pin with applied load and composite lay-up showing  
709 locations of four interfaces I1 to I4.

710 **Fig. 3.** Typical plate mesh for the evaluation of the (12 mm) pin-bearing behaviour of a pultruded  
711 specimen, with the symmetry plane facing backwards.

712 **Fig. 4.** Load-displacement plots from implementing the FE model, and comparison with  
713 experimental data from Matharu (2014): a) M12; b) M20.

714 **Fig. 5.** Specimen deformation at maximum predicted load (plots on deformed structure,  
715 magnification factor 2.5): a) M12; b) M20.

716 **Fig. 6.** Delamination at interfaces I1 (left) and I2 (right) at maximum predicted load (contour  
717 plotting on deformed mesh): a) M12 at 23kN; b) M20 at 36 kN.

718 **Fig. 7.** Effect of changing the size of the hole clearance: a) M12; b) M20.

719 **Fig. 8.** Delamination at interfaces I1 (left) and I2 (right) at maximum load (plots on deformed  
720 structure) when specimen has no clearance hole: a) M12 has maximum load of 25 kN; b) M20 has  
721 maximum load of 37 kN.

722 **Fig. 9.** Plots to maximum load having mat reinforcement TSFM replaced with CSM (filled markers  
723 – TSFM; unfilled markers – CSM): a) M12; b) M20.

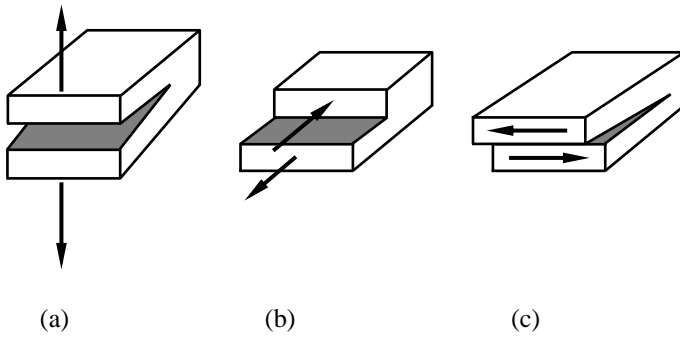
724 **Fig. 10.** Delamination growth at interface I1 at maximum load for mat reinforcement of TSFM  
725 (left-side) or of CSM (right-side): a) M12 has maximum load of 25 kN | 25 kN; b) M20 has  
726 maximum load of 38 kN | 40 kN)

727 **Fig. 11.** Comparison of original material with a new material when the chopped strand layer in the  
728 TSFM is replaced by the continuous fibre reinforcements (filled markers – TSFM; unfilled markers  
729 – TSFM without chopped strand layer): a) M12; b) M20.

730 **Fig. 12.** Delamination at interfaces I1 (left) and I2 (right) at maximum load when chopped strand  
731 layer in the TSFM is not present (plots on deformed structure): a) M12 has maximum load of 18  
732 kN; b) M20 has maximum load of 28 kN

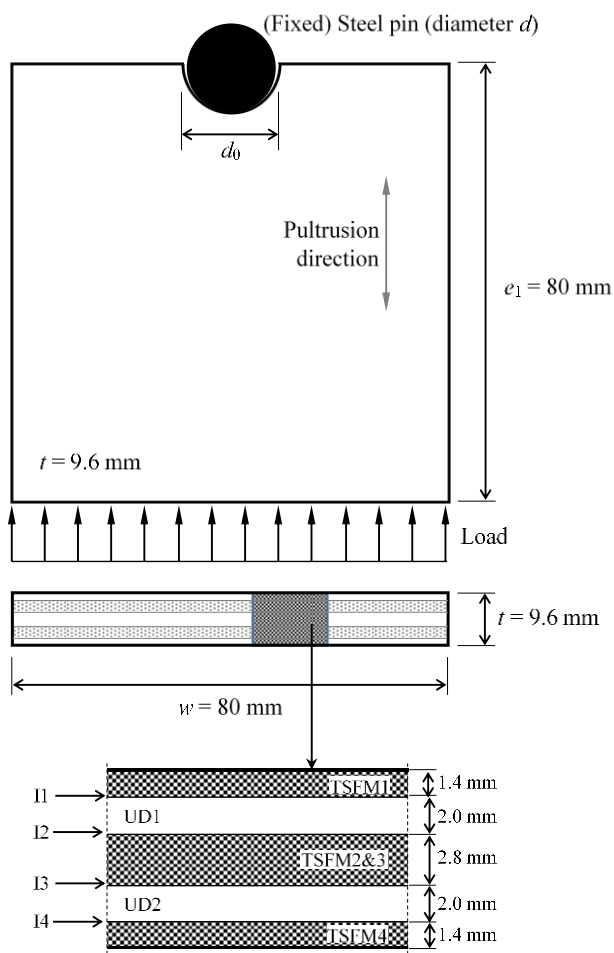
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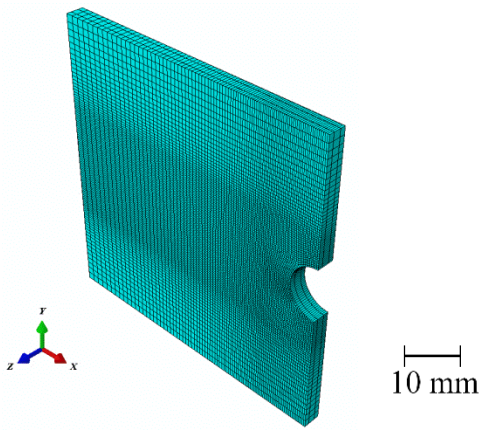
735 Fig. 1

736



737 Fig. 2

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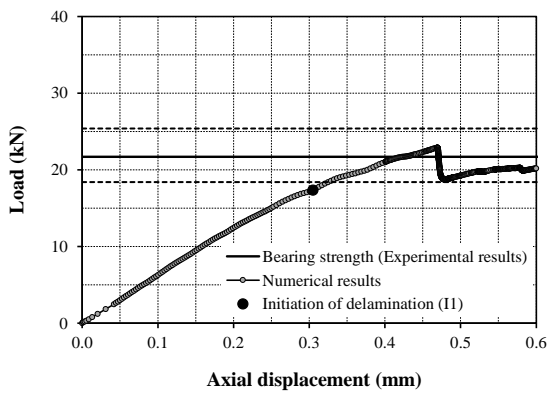


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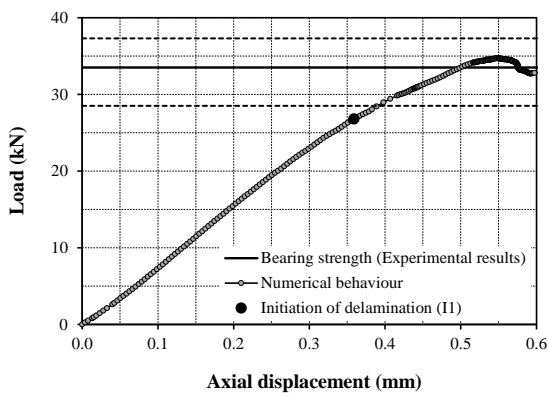
741 Fig. 3

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744 (a)



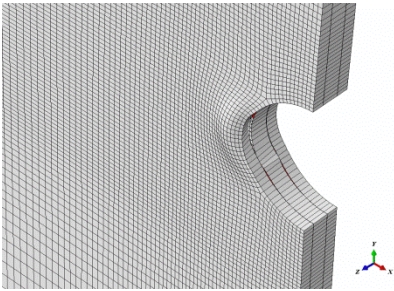
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746 (b)

747 Fig. 4



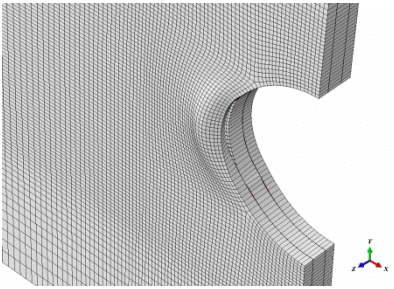
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(a)



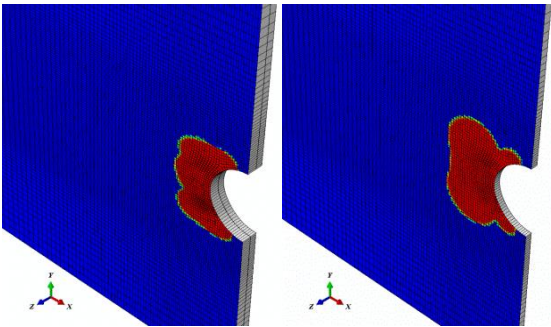
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(b)

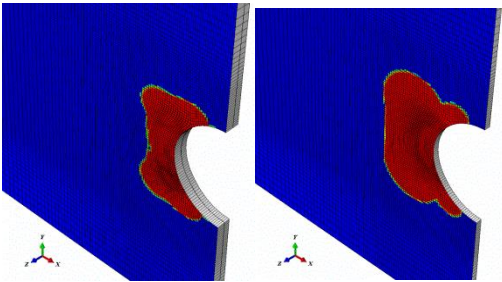
753 Fig. 5

754



755

(a)



756

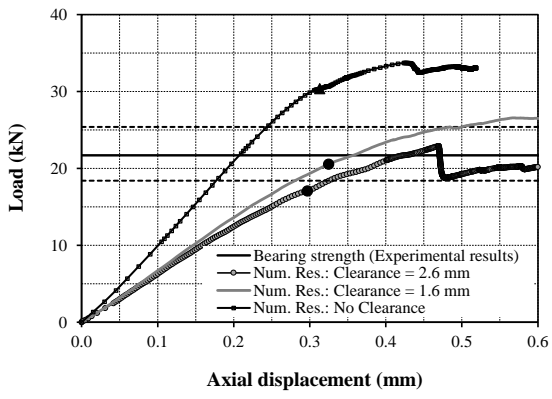
(b)

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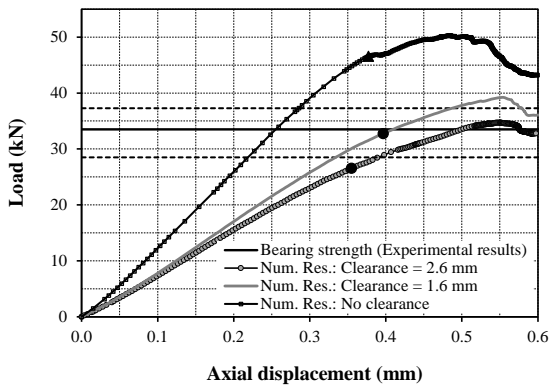
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759 Fig. 6

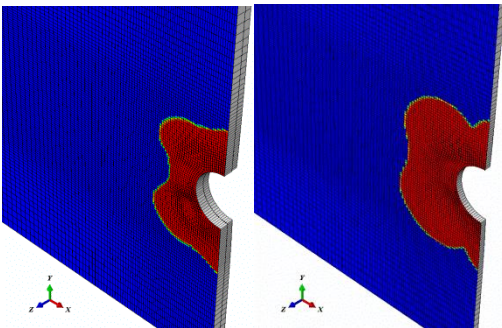
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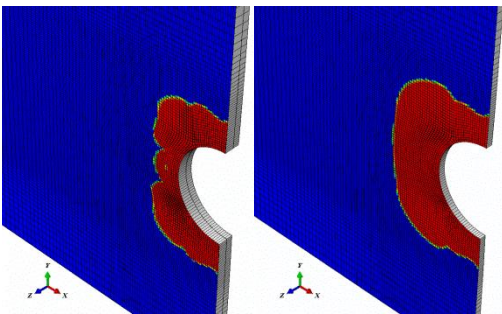
760  
761 (a)



762  
763 (b)  
764 Fig. 7



765 (a)

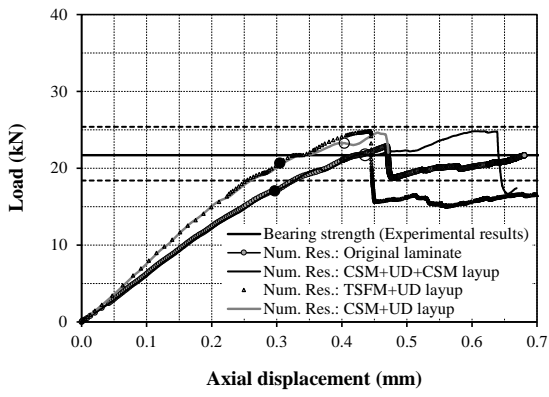


(b)

■ Completely damaged (0.7 to 1.0)
 ■ Partially damaged (0.3 to 0.7)
 ■ Undamaged less than 0.3

766  
767 Fig. 8

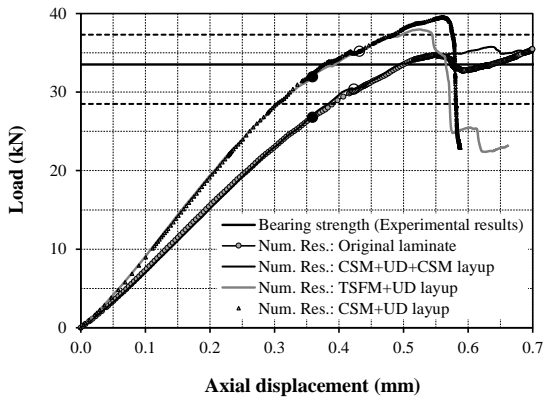
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769

770

(a)



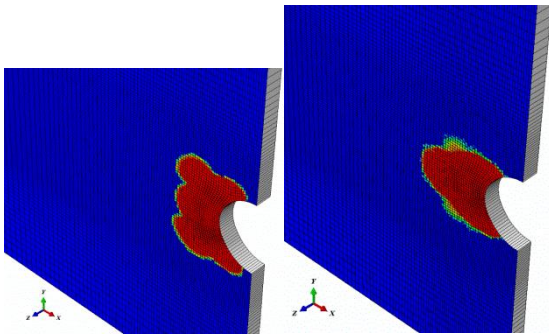
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772

(b)

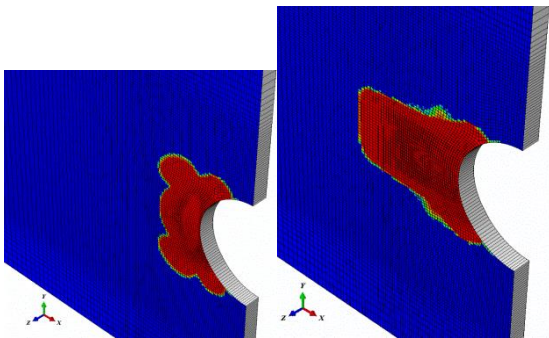
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Fig. 9



774

(a)



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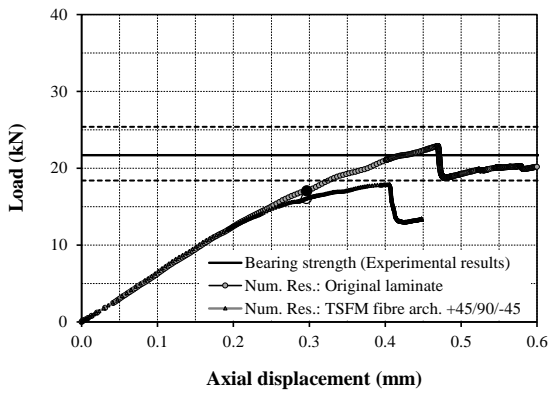
(b)

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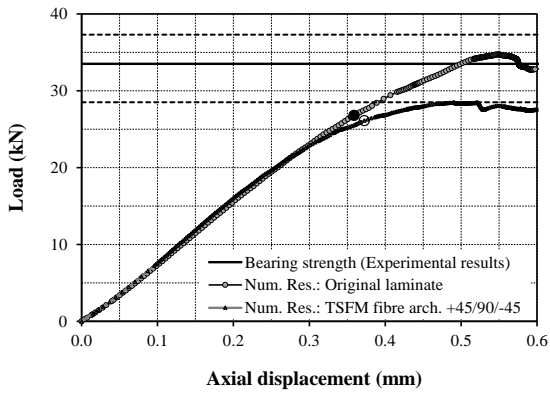
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 ■ Partially damaged (0.3 to 0.7)
 ■ Undamaged less than 0.3

Fig. 10



778

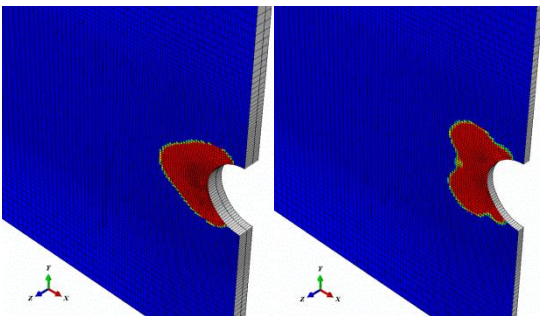
779 (a)



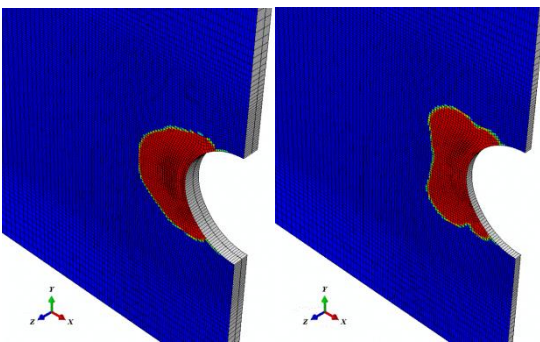
780

781 (b)

782 Fig. 11



783 (a)



(b)

■ Completely damaged (0.7 to 1.0)
 ■ Partially damaged (0.3 to 0.7)
 ■ Undamaged less than 0.3

784

785 Fig. 12

786 **Table 1.** Material constituent properties for the Pultrex® Superstructural material of 9.6 mm  
787 thickness.

	Constituent part			
	SV	TSMF	UD	Matrix
Number of layers in thickness	2	4	2	–
Fibre architecture	Random fibre veil	+45°/90°/–45°/CSM	156 (56 yield)	–
% volume fraction	3	34	33	30
% total of fibre reinforcement	5	48	47	–
Nominal layer thickness, $t_{lay}$ (mm)	0.03	1.4	2.0	–

788

789

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791

792 **Table 2.** Lamina properties for the laminate defined in Fig. 2.

Elastic lamina properties (kN/mm <sup>2</sup> )				Laminate strength properties (N/mm <sup>2</sup> )		Fracture energy (N/mm)	
Constituent part							
UD	TSFM						
		Mat layer (+45°/90°/−45°)	CSM				
$E_1$	44.7	13.1	7.7	$f_{1,T}$	294	$G_{1,T,c}$	100
$E_2 = E_3$	14.0	25.5	7.7	$f_{1,C}$	306	$G_{1,C,c}$	100
$G_{12} = G_{13}$	4.1	6.1	2.7	$f_{2,T}$	84	$G_{2,T,c}$	1.2
$\nu_{12}$	0.28	0.41	0.41	$f_{2,C}$	133	$G_{2,C,c}$	5
				$f_{1,S} = f_{2,S}$	91		

793

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795

796 **Table 3.** Interfacial properties for the pultruded FRP laminate defined in Fig. 2.

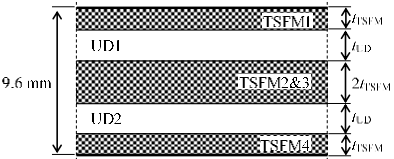
Interfacial properties	
$K_I=K_{II}=K_{III}$ (N/mm <sup>3</sup> )	10 <sup>6</sup>
$f_I$ (N/mm <sup>2</sup> )	21 <sup>1</sup>
$f_{II}=f_{III}$ (N/mm <sup>2</sup> )	23 <sup>1</sup>
$G_{I,c}$ (N/mm)	0.2
$G_{II,c}=G_{III,c}$ (N/mm)	0.5

797 **Note.** These values correspond to a weakening factor of 0.25.

798

799

800 **Table 4.** Modelling results: relative thickness of layers.

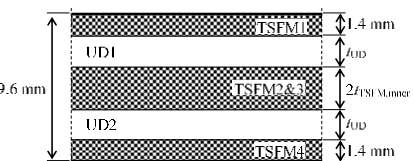
		M12					M20				
		$S_{el}$ (kN/mm)	$R_{br,ID}$ (kN)	$R_{br,max}$ (kN)	$R_{br,ID}/R_{br,max}$	$R_{br,max}/R_{br,max,baseline}$	$S_{el}$ (kN/mm)	$R_{br,ID}$ (kN)	$R_{br,max}$ (kN)	$R_{br,ID}/R_{br,max}$	$R_{br,max}/R_{br,max,baseline}$
<i>Original material</i>											
$t_{TSFM}$ (mm) (% of $t$ )	1.4 (58)	59	17	23	0.75	1.00	82	27	36	0.75	1.00
$t_{UD}$ (mm) (% of $t$ )	2.0 (42)										
<i>Parametric variation</i>											
$t_{TSFM}$ (mm) (% of $t$ )	1.0 (42)	69	19	23	0.82	1.02	95	31	35	0.89	1.00
$t_{UD}$ (mm) (% of $t$ )	2.8 (58)										
$t_{TSFM}$ (mm) (% of $t$ )	1.2 (50)	65	15	23	0.63	1.01	74	25	35	0.73	1.00
$t_{UD}$ (mm) (% of $t$ )	2.4 (50)										
$t_{TSFM}$ (mm) (% of $t$ )	1.6 (67)	58	20	24	0.87	1.05	88	28	35	0.81	1.01
$t_{UD}$ (mm) (% of $t$ )	1.6 (33)										

801

802



803 **Table 5.** Modelling results: thickness of internal TSFM layer.

		M12					M20				
		$S_{el}$ (kN/mm)	$R_{br,ID}$ (kN)	$R_{br,max}$ (kN)	$R_{br,ID}/R_{br,max}$	$R_{br,max}/R_{br,max,baseline}$	$S_{el}$ (kN/mm)	$R_{br,ID}$ (kN)	$R_{br,max}$ (kN)	$R_{br,ID}/R_{br,max}$	$R_{br,max}/R_{br,max,baseline}$
<i>Original material</i>											
$t_{TSFM,inner}$ (mm)	$(1.0t_{orig})$ 1.4	59	17	23	0.75	1.00	82	27	36	0.75	1.00
$t_{UD}$ (mm)	2.0										
<i>Parametric variation</i>											
$t_{TSFM,inner}$ (mm)	$(0.5t_{orig})$ 0.7	72	19	23	0.83	0.99	94	30	37	0.82	1.02
$t_{UD}$ (mm)	2.7										
$t_{TSFM,inner}$ (mm)	$(0.25t_{orig})$ 0.35	77	20	25	0.80	1.11	102	31	37	0.83	1.03
$t_{UD}$ (mm)	3.05										
$t_{TSFM,inner}$ (mm)	$(0.15t_{orig})$ 0.21	80	17	25	0.68	1.10	99	33	36	0.92	1.01
$t_{UD}$ (mm)	3.19										
$t_{TSFM,inner}$ (mm)	$(0t_{orig})$ 0	75	21	25	0.83	1.08	105	32	38	0.84	1.06
$t_{UD}$ (mm)	3.4										

804

805

806 **Table 6.** Modelling results: stacking sequence TSFM layer.

		M12					M20				
		$S_{el}$ (kN/mm)	$R_{br,ID}$ (kN)	$R_{br,max}$ (kN)	$R_{br,ID}/R_{br,max}$	$R_{br,max}/R_{br,max,baseline}$	$S_{el}$ (kN/mm)	$R_{br,ID}$ (kN)	$R_{br,max}$ (kN)	$R_{br,ID}/R_{br,max}$	$R_{br,max}/R_{br,max,baseline}$
<i>Original material</i>											
$t_{TSFM,1}$	+45°/90°/-45°/CSM	59	17	23	0.75	1.00	82	27	36	0.75	1.00
$t_{TSFM,2}$	+45°/90°/-45°/CSM										
<i>Variation</i>											
$t_{TSFM,1}$	+45°/90°/-45°/CSM	59	19	24	0.77	1.05	82	27	35	0.79	0.96
$t_{TSFM,2}$	CSM /-45°/90°/45°										
$t_{TSFM,1}$	CSM /-45°/90°/45°	59	20	23	0.87	1.01	82	33	37	0.88	1.08
$t_{TSFM,2}$	+45°/90°/-45°/CSM										
$t_{TSFM,1}$	CSM /-45°/90°/45°	59	20	23	0.88	1.01	82	33	40	0.82	1.16
$t_{TSFM,2}$	CSM /-45°/90°/45°										

807

808