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Energy loss rates of two-dimensional hole gases in inverted Si/Si_{0.8}Ge_{0.2} heterostructures

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We have investigated the energy loss rate of hot holes as a function of carrier temperature T_C in p-type inverted modulation-doped (MD) Si/SiGe heterostructures over the carrier sheet density range $(3.5-13)\times 10^{11}\,\mathrm{cm^{-2}}$, at lattice temperatures of 0.34 and 1.8 K. It is found that the energy loss rate (ELR) depends significantly upon the carrier sheet density, $n_{\rm 2D}$. Such an $n_{\rm 2D}$ dependence of ELR has not been observed previously in p-type SiGe MD structures. The extracted effective mass decreases as $n_{\rm 2D}$ increases, which is in agreement with recent measurements on a gated inverted sample. It is shown that the energy relaxation of the two-dimensional hole gases is dominated by unscreened acoustic phonon scattering and a deformation potential of $3.0\pm0.4\,\mathrm{eV}$ is deduced. © 2000 American Institute of Physics. [S0003-6951(00)02809-6]

A detailed quantitative understanding of hot carrier energy loss in two-dimensional (2D) carrier gas systems has yet to be realized. Si/SiGe and Ge/SiGe structures provide new vehicles for the study of this important topic and have revealed several unusual features. For example, Song et al.² claimed to have observed 2D phonon behavior in a strained Ge sample. In the present article, we address the question of energy relaxation by emission of 3D acoustic phonons and the controversy concerning the role of screening. Stöger et al.3 have found that they can explain their energy loss rates (ELR) for the 2D electron gas in strained silicon by taking static screening into account. Xie et al., 4 on the other hand, find evidence for weak screening in a strained Si_{0.8}Ge_{0.2} 2D hole gas, and explain other discrepancies between theory and experiment in terms of an additional piezoelectric coupling. Braithwaite et al.5 have made further measurements on Si_{0.8}Ge_{0.2}, confirm the need for a weak screening approach and point out that static screening gives much too large a value for the deformation potential.

The present experiments have been carried out on Si/Si_{0.8}Ge_{0.2}/Si heterostructures which modulation-doped (MD) by growing the Si:B doping slab before the alloy (inverted structure) as opposed to after alloy growth (normal structure), in contrast to previous work.^{4,5} Furthermore, we have investigated the carrier sheet density dependence of the ELR for the first time in an inverted SiGe heterostructure. Sample growth and preparation have been described previously. 6(a) In the inverted structure, the surface states on the silicon cap strongly affect the hole sheet density in the channel, which can be changed by reducing the silicon cap thickness using chemical etching. Assuming that the surface states pin the Fermi level in, say, midgap in Si, a reduction in cap thickness will pull the valence band maximum down at the bottom Si/SiGe interface, thereby reducing the carrier density.

Hall effect measurements at 0.4 T and 4 K give carrier

mobilities between 3200 cm 2 V $^{-1}$ s $^{-1}$ at a carrier sheet density of 3.5×10^{11} cm $^{-2}$ and 1500 cm 2 V $^{-1}$ s $^{-1}$ at 13 $\times 10^{11}$ cm⁻². The carrier densities agree with those obtained from measurements of Shubnikov-de Haas (SdH) oscillations in the resistance R as the magnetic field B is increased. For example, from the Hall effect, we obtain 5.3 ± 0.3 $\times 10^{11}$ cm⁻², while the resistivity obtained from the SdH period in B⁻¹ is $5.6\pm0.4\times10^{11}$ cm⁻². The results are consistent with the absence of parallel conduction, i.e., conduction at a single heterointerface in a single 2D subband. Poisson-Schrödinger calculations have confirmed that the position of the Fermi level is in the lowest subband. The SdH oscillations were examined in fields B < 4 T for which there is only one peak in the fast Fourier transform, Fig. 1, corresponding to one frequency of oscillation, and for which the condition $\Delta R/R < 0.1$ applies. Electric field (F) heating of carriers, at a lattice temperature T_L of 0.34 K, was investigated by SdH thermometry.⁵ The temperature dependence of the amplitude of the SdH oscillations was measured for infinitesimal electric fields (<0.2 V m⁻¹) and then, holding the lattice temperature at 0.34 K, different currents were passed through the sample to provide carrier heating in the electric field (F) range of 0.2-140 V m⁻¹. The carrier temperature was deduced by measuring the amplitude of the SdH oscillation,

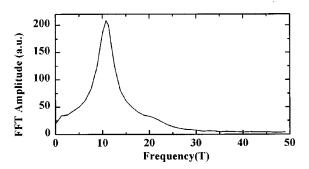


FIG. 1. Typical fast Fourier transform of the Shubnikov-de Haas oscillations observed in the present inverted MD samples, taken up to a field of 5 T. For fields below 4 T, the shoulder at a frequency of 22 T (and associated with spin splitting) is absent.

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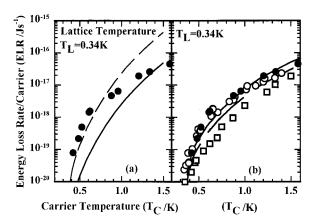


FIG. 2. (a) Experimental energy loss rate per carrier vs carrier temperature for an inverted MD sample with sheet density $5.1\times10^{11}~\rm cm^{-2}$ and Ge content x=0.2 (symbols). Lines represent theoretical calculations for fully screened deformation potential scattering with $D_{\rm ac}=20~\rm eV$ (solid) and $D_{\rm ac}=60~\rm eV$ (dashed). Effective mass used $m^*=0.29m_0$ and (b) ELR vs T_C for the same inverted sample as in (a) and two normal MD SiGe samples along with theoretical calculations for an unscreened potential. Inverted 5.1 $\times10^{11}~\rm cm^{-2}$ (\blacksquare Experiment), normal $5.6\times10^{11}~\rm cm^{-2}$ (\blacksquare Experiment, dashed line—theory, $m^*=0.3m_0$, $D_{\rm ac}=3~\rm eV$), normal $4.8\times10^{11}~\rm cm^{-2}$ (\square Experiment, solid line—theory, $m^*=0.3m_0$, $D_{\rm ac}=3~\rm eV$).

and by comparison with the lattice temperature dependence of the SdH amplitude. Since the SdH oscillation was measured at constant current (i.e., varying I^2R product), an interpolation method was used to determine the carrier temperature under constant power.

Although typical carrier mobilities were ~ 3600 cm² V⁻¹ s⁻¹, at a lattice temperature ≥ 1.8 K, it was found that the SdH oscillations were not as well pronounced when compared to lower temperatures. Therefore, to examine electric field heating at this higher lattice temperature, the analogous method of zero magnetic field conductance thermometry was used, as in our previous work.⁵ Hall effect measurements were used to deduce the carrier mobility μ at a carrier temperature T_C and the ELR per carrier, equal to $e \mu F^2$, was then determined.

The mean ELR due to inelastic scattering by acoustic phonons is given by 8

$$\left\langle \frac{dE}{dt} \right\rangle = \frac{\int_0^\infty \int_0^{2\pi} f(E, T_C) \hbar \, \omega_q [\, \nu_+(\,\theta) - \nu_-(\,\theta)\,] d\,\theta \, dE}{\int_0^\infty E f(E, T_C) dE} \quad (1)$$

with

$$\begin{split} \nu_{\pm}(\theta) &= \int_{-\infty}^{\infty} dq_z |I(q_z)|^2 P(q_{x,y}, q_z) (N_q + \frac{1}{2} \pm \frac{1}{2}) \\ &\times \left[1 - f(E \pm \hbar \, \omega_q, T_C)\right] \Theta\left(E - \frac{\hbar \, \omega_q}{2} \pm \frac{\hbar \, \omega_q}{2}\right), \end{split} \tag{2}$$

where θ is the scattering angle, $\nu_+(\theta)$, refers to phonon emission, $\nu_-(\theta)$ to phonon absorption, q_z and q_{xy} are the wave vector components perpendicular and parallel to the heterointerface, N_q is the phonon occupation number at wave vector $q = (q_{xy}^2 + q_z^2)^{1/2}$ and lattice temperature T_L , Θ is the Heaviside function, $f(E, T_C)$ the Fermi Dirac distribution for a carrier temperature T_C , $I(q_z)$ is the form factor for the Fang–Howard wave function, and $\hbar \omega_q$ is the acoustic phonon energy. The acoustic phonon scattering probability is

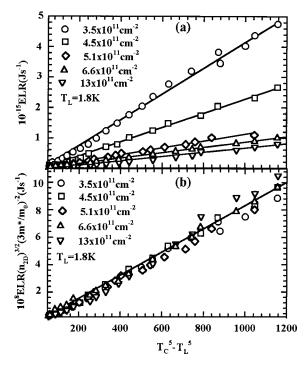


FIG. 3. (a) Measured energy loss rate per carrier (ELR) for samples of carrier sheet densities $(3.5-13)\times 10^{11}\,\mathrm{cm}^{-2}$, plotted vs $T_C^5-T_L^5$, and (b) normalized energy loss rate plotted vs $T_C^5-T_L^5$.

$$P(q_{xy}, q_z) = \frac{m^* D_{ac}^2 u_L(x) q}{8 \pi^2 \hbar c_L(x) S^2(E, T_C, q)},$$
(3)

where $S(E,T_C,q)$ is the static screening factor, ${}^8c_L(x)$ is the longitudinal elastic coefficient, m^* is the in-plane effective-mass, u_L is the longitudinal sound velocity, and $D_{\rm ac}$ is the acoustic phonon deformation potential. The theoretically predicted maximum values of $S^2(E,T_C,q)$ are between 6.5 (for $n_{\rm 2D}=13\times10^{11}\,{\rm cm}^{-2}$) and 16 (for $3.5\times10^{11}\,{\rm cm}^{-2}$).

Figure 2(a) shows the measured energy loss rate for an inverted structure of sheet density $5.1\times10^{11}\,\mathrm{cm^{-2}}$, at a constant lattice temperature of 0.34 K. Also shown is an attempt to fit our experimental data using a screened scattering potential with S>1. Depending on the value of D_{ac} chosen, we only obtain a fit at low $T_C(D_{\mathrm{ac}}=60\,\mathrm{eV})$ or at high $T_C(D_{\mathrm{ac}}=20\,\mathrm{eV})$ but are unable to obtain a fit over the whole carrier temperature range. Moreover, the values of D_{ac} are considerably in excess of what is reported for bulk Si and Ge (i.e., 4 and 3.5 eV, respectively), 9 and are therefore regarded as unrealistic.

We next attempt to fit the data by assuming S=1 (weak screening), as shown in Fig. 2(b). Experiment and theory now agree within experimental error, for both the normal and inverted structures, with a very plausible value of $D_{\rm ac}=3$ eV. In order to further test the validity of the weak screening assumption, we now consider the experimental data for four inverted MD SiGe samples of various carrier densities, at a lattice temperature of 1.8 K. At low enough temperatures where the phonon population is small (Bloch–Gruneisen regime), the carrier gas is degenerate and the Pauli exclusion restricts the allowed scattering processes, Eq. (1) with S=1 reduces to 4

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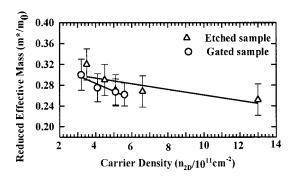


FIG. 4. Comparison between the extracted effective masses of the etched samples and those measured on a gated sample [see Ref. 6(b)].

$$ELR = \frac{m^{*2} D_{ac}^2 k_B^5 (4!) \zeta(5)}{8\sqrt{2} \pi^{5/2} (n_{2D})^{3/2} \rho u_I^4 \hbar^7} (T_C^5 - T_L^5), \tag{4}$$

where ρ is the mass density and $n_{\rm 2D}$ is carrier sheet density. In Fig. 3(a), the experimental ELR are plotted against T_C^5 $-T_I^5$ and a linear dependence is demonstrated. It is also seen that the power loss of the investigated samples depends upon the carrier sheet density. This carrier sheet density dependence of the energy loss rate has not previously been observed in similar normal MD Si/SiGe samples.⁵ This is also in contrast to the results given by Stöger et al.3 who found no dependence of ELR on carrier density for the 2D electron gas in strained silicon. We note from Eq. (4) that the ELR should vary as $(T_C^5 - T_L^5)m^{*2}(n_{2D})^{-3/2}$. The data are therefore replotted in Fig. 3(b) as normalized energy loss rate $[ELR(n_{2D})^{3/2}m^{*-2}]$ vs $T_C^5-T_L^5$ and by adjusting m^* to obtain a universal straight line. The m^* so obtained are compared in Fig. 4 with effective mass measurements on a gated inverted structure^{6(b)} and it is seen that they agree to within experimental error. Having demonstrated that the weak screening approach gives a satisfactory description of the variation of ELR with effective mass and carrier density, we compute the average value of $D_{\rm ac}$ for the complete set of data from the exact theory for an unscreened potential, Eq. (1) with S=1, and as before obtain $3.0\pm0.4\,\mathrm{eV}$.

To summarize, the static screening approach does not seem to work for the present samples, since we need to assume unreasonably small values of the static screening factor S in order to fit our results. A possible explanation is provided by Fischetti and Laux¹⁰ who point out that dynamic

screening of the acoustic phonons should apply at low enough temperatures and/or high carrier densities. They also point out that the theoretical treatment is quite intractable at present, but it is reasonable to expect that such screening might be quite weak. Challis *et al.*¹¹ also question the procedures used to account for screening of a deformation potential and note that the screening might, in any event, be reduced by other scattering mechanisms. In the case of the Si/SiGe system, the question as to why static screening seems to be appropriate for electrons³ but not for holes remains to be answered.

In conclusion, we have investigated the energy loss rate of holes in *p*-type inverted modulation-doped Si/SiGe heterostructures. The results support an interpretation in terms of weak screening of the acoustic phonon deformation potential, in contrast to the behavior seen in the Si/SiGe electron gas.

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