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COMMUNICATION

The intercalation of 1,10-phenanthroline into layered NiPS₃ via iron dopant seeding

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Using 2% percent of iron dopants as reaction active sites yields a series of single crystals of 1,10-phenanthroline intercalated NiPS₃, via solution reaction with aniline chloride, not possible by direct reaction. Experimental magnetic susceptibility measurements demonstrate that 1,10-phenanthroline intercalation suppresses the anti-ferromagnetism ordering at around 150 K in Fe_{0.02}Ni_{0.98}PS₃, and gives rise to a ferrimagnetic phase transition at the temperature around 75 K. An intercalation mechanism is proposed for the reaction, and this dopant seeding method provides a new approach for intercalation into layered materials.

The intercalation chemistry of ternary transition-metal phosphorus tri-chalcogenides MPX₃ (M=3d transition metal; X=chalcogenide) has attracted extensive attention due to their numerous properties for application in batteries, high performance catalysis and optoelectronics.¹⁻³ In recent years, interest has been paid to their electronic and magnetic properties. For example, MPX₃ shows diverse AF ordering states; MnPS₃ largely behaves like a Heisenberg antiferromagnet, while FePS₃ has the properties of an Ising antiferromagnet. Furthermore, superconductivity has been observed in FePSe₃ with the highest T_c of about 5.5 K achieved by suppressing AF order with external pressure, showing resemblance to high T_c cuprates and iron-based superconductors.⁵ *Ab initio* study calculations also suggests that for single-layer APX₃, competition between anti-ferromagnetic and ferromagnetic states can be substantially influenced by gating and by strain engineering; the sensitive interdependence between magnetic, structural, and electronic properties suggests the potential of this 2D materials class for applications in spintronics.^{5,6} Very recently, Hu *et al.*, proposed that Ni-

based transition-metal tri-chalcogenides are a promising platform to study exotic topological phenomena emerging from electronic correlation effects and that the emergence of topological superconductivity may be possible in Ni-based trichalcogenides.

Intercalation of organic species into layered inorganic solids provides a useful approach to creating ordered organic-inorganic nanocomposite materials with novel properties distinct from the parent compounds, which has attracted considerable attention. For example, the T_c of iron-based superconductor FeSe is only 8 K, but dramatically increases up to 44 K by the intercalation of alkaline and alkaline-earth metals; moreover, and lithium and pyridine-intercalated Li_x(C₅H₅N)_yFe_{2-z}Se₂ exhibits superconductivity with $T_c=45$ K.⁶ For lamellar MPX₃ compounds, two types of intercalation reaction have been reported: lithium intercalation in NiPS₃ and FePS₃ involves electron donation from guest to host, which causes valence state change of the transition metal,^{2,7,8} while for MPS₃ (M=Mn, Fe, Cd, Zn), some of the M cations may be removed during intercalation by strong complexation with a ligand and the charged vacancies are later balanced by protonated guest cations, therefore the valence state of the transition metal remains unchanged.^{2-4,7,9} But so far there are no detailed intercalation investigations with NiPS₃ reported with organic species. In this communication, we report the intercalation of complexing agent 1,10-phenanthroline into layered NiPS₃ in solution with aniline chloride by using iron dopants as reaction active sites, and a series of single crystals of the solid solution Fe_xNi_{1-x}PS₃ and *N,N*-dimethylformamide (DMF) intercalated Fe_xNi_{1-x}PS₃ has also been synthesised (see S1).

Powder X-ray diffraction measurements were used to characterise the structure of Fe_xNi_{1-x}PS₃ samples and Fig. 1a shows the experimental data for samples with nominal compositions of x from 0 to 1. It was previously reported that the family all have monoclinic symmetry with space group $C2/m$; the lattice parameters are $a=5.9681$ Å, $b=10.2925$ Å, $c=6.7229$ Å and $\beta =107.374^\circ$ for FePS₃, and $a=5.7978$ Å, $b=10.1074$ Å, $c=6.6265$ Å and $\beta =107.050^\circ$, for NiPS₃, which is consistent with the larger ionic radius of Fe²⁺ compared to Ni²⁺. Profile refinement of the powder XRD for the Fe_xNi_{1-x}PS₃ samples shows that the lattice parameters increase progressively with

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the increase of the Fe content (Table S1), confirming the successful substitution of Ni by Fe in $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$. The inset of Fig. 1a shows the (001) Bragg peak, showing a shift to lower angle with increased Fe content. Fig. 1b presents a typical example of EDX spectrum from a piece of single crystal, which confirms the presence of the Ni, Fe, P and S, and the composition of the material is estimated as $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$. The inset of Fig. 1b shows a SEM image illustrating the morphological features of $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ crystal grains, where layered structural features can be easily observed. All the $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ materials show similar morphological features and a chemical composition very close to the expected nominal composition. The crystal structure of $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ was tested by electron-diffraction patterns taken along the [001] zone axis Fig. 1c, in which all primary diffraction spots can be good agreement with the XRD data.

We find that no intercalation is observed for pure NiPS_3 using complexing agent 1,10-phenanthroline with aniline chloride in DMF even with prolonged reaction time. Fig. 2a shows the XRD patterns of the $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ samples taken out at six different reaction times, which is typical for all $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ precursors. At the early stage, only the patterns of pure $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ were observed. After 1 day of reaction, there are two phases, pristine $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ and a new phase, and as the period of intercalation is prolonged, the XRD reflections of pristine $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ phase gradually disappear and the new phase predominates. Finally, the pure intercalation phase is obtained after 9 days. Detailed analysis suggests the intercalated $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ product, the lattice parameters are $a=5.7821 \text{ \AA}$, $b=10.0902 \text{ \AA}$, $c=15.6254 \text{ \AA}$ and $\beta=107.115^\circ$, the increase of lattice spacing is about $\Delta c=8.9979 \text{ \AA}$ compared with the $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ precursor. Table S1 lists the lattice parameters obtained from our XRD experiments. It is clear that for the intercalates, a and c increase with the increase of Fe substitution level, x . The increase of the lattice parameters can be explained by the co-contribution of the increase of the lattice parameters of the $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ starting materials and the slight increase of the inserted 1,10-phenanthroline guest species. Fig. 2b shows the EDX spectrum of the intercalated $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ product, where no Fe was detected, and Ni, P and S with a 0.98:1:3 atomic ratio, indicating that the 1,10-phenanthroline intercalation process extracts all of

the Fe. An SEM image of a typical intercalation product with evident layered structure shown in the inset of Fig. 2b, and a diffraction pattern taken from the [001] zone axis is shown in Fig. 2c, demonstrating that despite the significant increase of the interlayer space due the intercalation of 1,10-phenanthroline, the intralayer structure is similar to the $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ precursors.

A possible intercalation mechanism of intercalation reaction of 1,10-phenanthroline with layered FePS_3 in ethanol and in the presence of anilinium chloride was previously proposed by Chen *et al.*⁹ who suggested the intercalation is a non-redox reaction, and as a result of intercalation, the increase of interlayer lattice spacing of 9.0 \AA with respect to pure FePS_3 indicates that the 1,10-phenanthroline molecular ring is oriented perpendicular to the layer of the host when excess 1,10-phenanthroline is used in the reaction. A smaller increase of interlayer lattice spacing of 3.5 \AA would show that the guest is parallel to the layer when a suitable amount of phenanthroline is used.³

From the results we propose a possible intercalation mechanism for 1,10-phenanthroline inserted into $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ (see Steps. 1 to 3) in Fig. 3, in which 1,10-phenanthroline acts as a complexing agent to remove intralaminar iron ion into solution during intercalation and leads to the presence of the Fe^{2+} vacancies in the Ni-P-S layers, and aniline chloride provides a proton to 1,10-phenanthroline. The intercalation of $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ is then achieved by charge attraction between the charged vacancy defective point and the protonated 1,10-phenanthroline, so that the charge neutrality of the intercalation is maintained.^{3, 7,10-13} Our results also suggest that the intercalation is triggered by the doped Fe as active centre and that no Ni is involved in the whole process, consistent with the fact that no intercalation is observed on pure NiPS_3 , no matter how long the time and how much 1,10-phenanthroline and aniline chloride is used.

Fig. 4a shows the IR spectra of pure $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$, intercalated $\text{Ni}_{0.98}\text{PS}_3$, neutral phenanthroline and anilinium chloride. In general, 1,10-phenanthroline intercalates have three sharp absorption peaks at around 614 , 576 , and 553 cm^{-1} in the range 550 to 650 cm^{-1} , indicating the presence of intralaminar transition metal. It has been

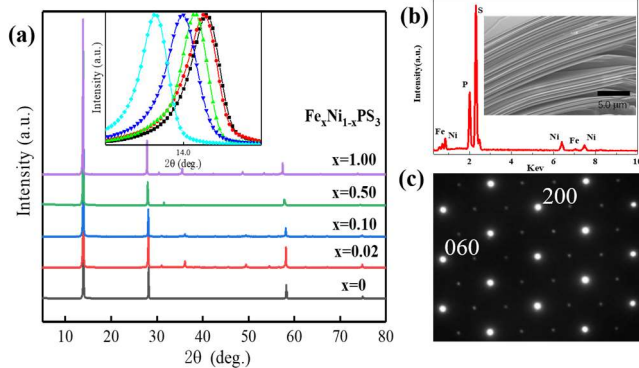


Fig. 1(a) XRD patterns of $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$, showing the increase of the interlayer distance with the increase of Fe content. (b) EDX spectrum and a typical SEM image of a final intercalated product. (c) Selected-area electron diffraction patterns taken along the [001] zone-axis direction of the final intercalated product $\text{Ni}_{0.98}\text{PS}_3$.

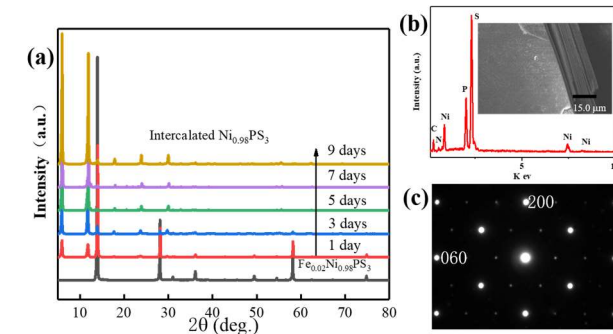


Fig. 2(a) XRD patterns used to monitor the intercalation of $\text{Ni}_{0.98}\text{PS}_3$ with 1,10-phenanthroline. (b) The EDX spectrum and a typical SEM image of a final intercalated product. (c) Selected-area electron diffraction patterns taken along the [001] zone-axis direction of the final intercalated product $\text{Ni}_{0.98}\text{PS}_3$ product.

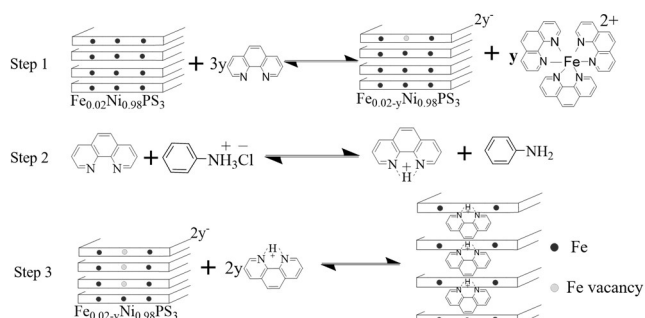


Fig.3 Intercalation mechanism of 1,10-phenanthroline with layered $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$. Step 1: doped Fe^{2+} reacted as active centre removed initially by 1,10-phenanthroline. Step 2: proton transfer between 1,10-phenanthroline and aniline chloride. Step 3: the protonated 1,10-phenanthroline inserted into the layer to balance charge neutrality is perpendicular to the layer of the host.

previously shown that the occurrence of the intralamellar M^{2+} vacancies in MPS_3 makes the P-S bonds in the PS_3 groups unequivalent, which causes asymmetric $\nu(\text{PS}_3)$ stretching vibrations of the host and gives rise to the splitting of the 575 cm^{-1} peaks.^{2,3,14} Fig. 4b shows the UV spectra of the filtrate of reaction mixture after intercalation, where an absorption peak around 510 nm is seen, which is characteristic for $\text{Fe}(\text{phen})_3^{2+}$, indicating the 1,10-phenanthroline complexes the doped iron in $\text{Fe}_{0.02}\text{Fe}_{0.98}\text{PS}_3$, removing it into solution as $\text{Fe}(\text{phen})_3^{2+}$ during the intercalation reaction.² The infrared spectra of the intercalated compounds show similar characteristic absorption bands (at around 1616, 1593, 1421, 1094, 843, and 723 cm^{-1}) to those of molecular phenanthroline, which proves the existence of phenanthroline based species as a guest.^{2,3} In order to prove that the intercalated guest species is protonated phenanthroline, we added acetic acid instead of anilinium chloride as proton provider in the reaction mixture, which can also successfully trigger the intercalation reaction. However, without adding acid in the reaction mixture, no intercalation occurs for $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$, even though the $\text{Fe}(\text{phen})_3^{2+}$ can be detected in the filtrate. These experiment results support the conclusion that the protonation of phenanthroline is essential for intercalation and the guest intercalated into the layer is protonated phenanthroline. Further evidence is provided by IR spectroscopy (see Fig.2S).

Our model is further supported by the thermogravimetric (TG) analysis and differential thermal analysis (DTA) shown in Fig. 5a. The first mass loss can be assigned to the deintercalation of protonated 1,10-phenanthroline. The mass loss derived from

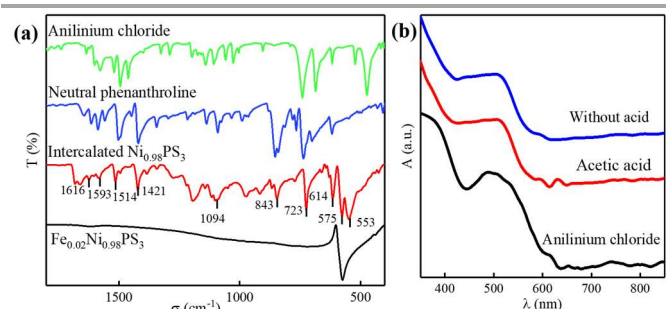


Fig. 4(a) IR spectra of pure $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$, 1,10-phenanthroline intercalated $\text{Ni}_{0.98}\text{PS}_3$, neutral phenanthroline and anilinium chloride. (b) The UV spectra of the filtrates of reaction mixture after intercalation reaction.

the measurement is 3.80%, consistent with the mass of protonated 1,10-phenanthroline that compensates for the vacancies left by Fe^{2+} : if the doped Fe is fully reacted, there should be 3.77% protonated 1,10-phenanthroline present. This confirms that the 1,10-phenanthroline only extracts Fe and therefore that the protonated 1,10-phenanthroline achieves intercalation into $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$ by charge attraction so that charge neutrality is maintained. When the doped Fe content, x , increases to $x=0.1, 0.5$ and 1 , the mass loss of protonated 1,10-phenanthroline from TG measurement, is 7.55, 10.46 and 13.41%, respectively, which corresponds to 4.0%, 5.5%, and 7.1% intralamellar vacancy concentration in $\text{Fe}_x\text{Ni}_{1-x}\text{PS}_3$, respectively. A further mass loss occurring from 600 to $700\text{ }^\circ\text{C}$ corresponds to the decomposition of the compound.

The materials, FePS_3 and NiPS_3 , show ordering anti-ferromagnetically at low temperatures, with the Néel temperature being 123 and 153 K, respectively.^{4,15} The right hand inset of Fig. 5b shows the temperature dependent susceptibility for $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$, which is similar to NiPS_3 with an antiferromagnetic phase transition at around 150 K. Fig. 5b shows the temperature dependent susceptibility curves of intercalation $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ with 1,10-phenanthroline in different magnetic fields, and the Curie-Weiss fitting curve. Fig. 5c shows $\chi \times T$ -T and $1/\chi$ -T curve of magnetic field $H=0.1\text{ T}$. The magnetism of the intercalated sample is different to the precursor $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$ that the anti-ferromagnetic phase at around 150 K no longer exists. As shown in Fig. 5c the $1/\chi$ -T curve above 100 K is in good agreement with the Curie-Weiss law with a Weiss constant of $\theta = -396.56\text{ K}$. The negative Weiss constant indicates the existence of the antiferromagnetic interactions between the localized Ni^{2+} ions in the paramagnetic range. It can be seen clearly that the $\chi \times T$ -T curve at 75 K exhibits a sudden upward turn, suggesting the existence of a ferrimagnetic phase transition.^{3,15,16} Fig. 5d shows hysteresis loop measurements at temperatures, $T=80, 40$ and 1.80 K . The magnetic hysteresis

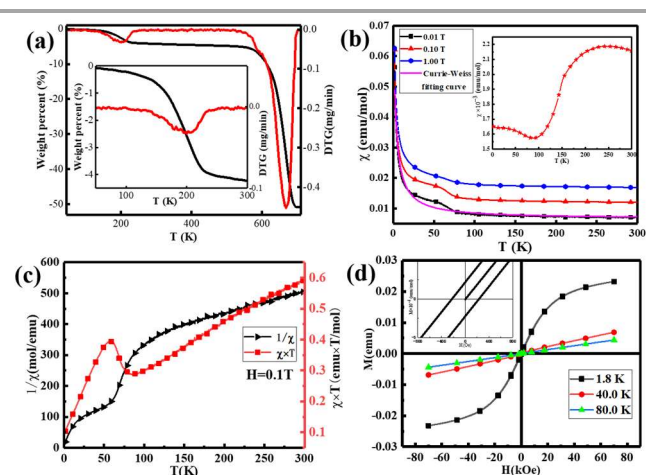


Fig. 5(a) The TG and DTG curves for the final intercalated $\text{Ni}_{0.98}\text{PS}_3$ product from 30 to $700\text{ }^\circ\text{C}$, and the inset is the zoomed figure in the range of $40\text{--}300\text{ }^\circ\text{C}$. (b) Temperature dependence of the magnetic susceptibility of intercalated $\text{Ni}_{0.98}\text{PS}_3$, the right hand inset shows the χ -T curve of pristine $\text{Fe}_{0.02}\text{Ni}_{0.98}\text{PS}_3$. (c) The $1/\chi$ -T and $\chi \times T$ -T curves of intercalated $\text{Ni}_{0.98}\text{PS}_3$ in $H=0.1\text{ T}$. (d) The hysteresis loop measurements at temperature, $T=1.8, 40$ and 80 K , respectively, and the inset shows the hysteresis loop zoomed figure in range of from -800 to 800 Oe at temperature, $T=1.8\text{ K}$.

loop with coercivity (320 Oe) found at 1.8 K further suggests that the intercalation compound is a low-temperature ferrimagnet. In NiPS₃, the magnetic order is a zig-zag stripe type of anti-ferromagnetism (AFM). The main AFM interactions stem from the two-third nearest neighbour Ni atoms while the first- and second-nearest neighbour magnetic interactions are ferromagnetic.¹⁶ In the 1,10-phenanthroline intercalated Fe_xNi_{1-x}PS₃ samples, the local lattice distortion caused by the vacancy significantly increases those ferromagnetic interactions locally to cause the observed ferromagnetic signal because they are very sensitive to the change of the local angles of Ni-S-Ni.

Further experiments show that Co and Mn doped NiPS₃ have a very similar 'seeding' effect. XRD patterns of Co_{0.02}Ni_{0.98}PS₃ and Mn_{0.02}Ni_{0.98}PS₃ and their corresponding 1,10-phenanthroline intercalations are shown in Fig. 6S. In summary, a series of Fe_xNi_{1-x}PS₃ (0 ≤ x ≤ 1) single crystals and their corresponding 1,10-phenanthroline intercalated compounds have been synthesised. As a result of intercalation, the lattice parameter of the c axis is expanded by 8.9979 Å with respect to pure Fe_{0.02}Ni_{0.98}PS₃. It is found that the 1,10-phenanthroline only extracts Fe and the protonated 1,10-phenanthroline achieves intercalation into Fe_xNi_{1-x}PS₃ by charge attraction so that the charge neutrality of the intercalation is maintained. Fe_xNi_{1-x}PS₃ materials show anti-ferromagnetic ordering at low temperature, and the anti-ferromagnetic ordering in pure Fe_xNi_{1-x}PS₃ was suppressed after intercalation and instead gives rise to a ferrimagnetic phase transition at the temperature of about 75 K. The dopant seeding method could be applicable to other intercalation reaction to allow access to phases not accessible by other methods.

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Conflicts of interest

There are no conflicts to declare.

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